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Analysis of the Failure of Cracked Biscuits

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- Effect of cracks on biscuit strength was explored.
- Failure by overstressing was compared to failure by crack propagation.
- Tensile strength and fracture toughness were measured experimentally.
- Experimental results were confirmed by theoretical analysis.

1	Analysis of the Failure of Cracked Biscuits
2	
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13	ABSTRACT
14	Cracks or checks in biscuits weaken the material and cause the product to break at low load
15	levels that are perceived as injurious to product quality. In this work, the structural response
16	of circular digestive biscuits, with diameter 72 mm and thickness 7.2 mm, simply supported
17	around the circumference and loaded by a central concentrated force was investigated by
18	experiment and theory. Tests were conducted to quantify the distribution in breakage strength
19	for structurally sound biscuits, biscuits with natural checks and biscuits with a single known
20	part-through crack. For sound biscuits the breakage force is Normally distributed with a mean
21	of 12.5 N and standard deviation of 1.2 N. For biscuits with checks, the corresponding
22	statistics are 9.6 N \pm 2.62 N respectively. The presence of a crack weakens the biscuit and
23	strength, as measured by breakage force falls almost linearly with crack length and crack
24	depth. The orientation of the crack, whether radial or tangential, and its location (i.e. position
25	of the crack mid-point on the biscuit surface) are also important. Deep, radial, cracks located
26	close to the biscuit centre can reduce the strength by up to 50 %. Two separate failure criteria
27	were examined for sound and cracked biscuits respectively. The results from these tests were
28	in good accord with theory. For a biscuit without defects, breakage occurred when maximum
29	biscuit stress reached or exceeded the failure stress of 420 kPa. For a biscuit with cracks,
30	breakage occurred as above or alternatively when its critical stress intensity factor of 18
31	kPam ^{0.5} was reached.
~ ~	

32

33 Keywords: Biscuits, Fracture, Cracks, Stress Intensity Factor

34			
35		NOTATION	
36	a	Crack half length	m
37	b	Biscuit sample width	m
38	g ₁	Crack depth to biscuit thickness parameter	-
39	g ₂	Crack length to biscuit thickness parameter	-
40	K _I	Stress intensity factor	kPa m ^{0.5}
41	K _{IC}	Critical stress intensity factor	kPa m ^{0.5}
42	L	Support span length	m
43	Р	Applied force	Ν
44	R	Biscuit radius	m
45	r	Radial distance	m
46	r _c	Crack radial location (mid-point)	m
47	r ₀	Radius of applied load	m
48	r ₀ '	Equivalent loading radius	m
49	t	Biscuit thickness	m
50	W	Crack depth	m
51	Х	Linear distance	m
52			
53			
54	α	Crack angular orientation	rad
55	σ	Stress	kPa
56	ν	Poisson's ratio	-
57	θ	Angle	rad
58		\mathbf{C}	
59			
60		1. INTRODUCTION	
61	Biscuits are o	ne of the most consumed snack-type products acros	ss the world by all levels of
62	society (Okpa	la and Okoli, 2013). Their popularity is mainly due	to their sweet taste, ready-
63	to-eat nature,	affordable cost, nutritional value and long shelf life ((Sudha et al., 2007; Vujic et
64	al., 2014). On	e of the most important quality features of biscuits	is texture (Mamat and Hill,
65	2012). Textur	e depends on many factors including the structure of	f the biscuit and methods of
66	manufacturing	g and handling during the process. Texture is the	mechanical strength of the

biscuit quantified by the load required to produce failure by fracture. From a physical basis, this load can be taken as equivalent to the critical stress level at which an existing flaw propagates through the material and leads to breakage of the biscuit (Kim et al., 2012). Hence the fracture properties of the biscuit must be understood. Fracture properties relate the loading on a biscuit to its structural response and particularly to the propagation of cracks leading to failure. For biscuits this is also related to the phenomenon of checking.

73

74 For almost a century, biscuit manufacturers have sought to avoid 'checking' This can be 75 defined as the appearance of small hairline cracks in biscuits after baking that affects fragility 76 and product degradation (hence checks are naturally occurring cracks resulting from the 77 baking process). This phenomenon of crack formation can occur during the industrial cooling 78 of the biscuits as a consequence of the stresses generated by dimensional changes associated 79 with equilibration of moisture due to moisture gradients within the freshly baked biscuit 80 (Manley, 2000). These cracks extend from the centre towards the periphery, making the whole structure weak and giving the possibility for the biscuit to break spontaneously (Dunn & 81 82 Bailey 1928). The drying process in the last zones of the baking oven inevitably causes the 83 central and thicker parts of the biscuit to have slightly more moisture than the outer zones. 84 During subsequent cooling and storage, moisture diffuses from regions of high moisture 85 content (centre) to areas with less moisture (edges), which also take up moisture from the 86 surrounding environment. This moisture migration leads to expansion towards the edge of the 87 biscuit and contraction at the centre, causing stresses to build up in the biscuit. Depending on 88 the physical properties of the biscuit structure as it cools, cracks may develop when these 89 stresses exceed a critical value Manley, (1983).

90

91 In addition to the possible presence of checks or cracks, it should be understood that baked 92 biscuits contain a very large number of pores ranging in size from 10 µm up to 300 µm, 93 (Pareyt et al., 2009). The pores are formed as a result of water vapour production and 94 expansion during the baking process. Morphology of these pores can vary from rounded to 95 very angular. Long narrow, notched pores can act as sites of stress concentration and hence as 96 crack initiation points while large rounded pores in the structure offer the possibility of 97 arresting the propagation of cracks. Thus the microstructure of the biscuit has an effect on its 98 physical and sensory properties (Frisullo et al., 2010). The value and functionality of most of 99 the brittle food products rely on their cellular foam structure that is strongly linked to texture 100 (Lim and Barigou, 2004). The complex non-uniformity in the internal structure of the biscuit

101 means the process of crack propagation in such materials, compared to homogenous ones, 102 possesses additional features due to their randomness. The random distribution of pores in 103 location, size and shape makes the fracture of porous materials difficult to predict. 104 Contradictory results are often reported in the fracture of porous materials; strengthening and 105 weakening. No simple relationship is possible as the fracture performance depends on the 106 distribution in pore size, shape and location (Legullion & Piat, 2007). These two effects of 107 non-uniformity of the internal porous structure and randomness associated with the potential 108 presence of checks causes the well-known scatter in strength and fracture parameters.

109

110 Superimposed on the phenomenon of fracture is the fact that the presence of pores also 111 influences the effective elastic constants and a random redistribution in the nominal stress, 112 (Ramakrishnan & Arunachalam, 1990). Thus many researchers who have examined brittle, porous foods have considered them to act as a homogeneous, elastic solid using nominal 113 values for the mechanical properties of the homogenised sections (Rojo & Vincent, 2008). 114 The other approach involves very detailed morphological structural modelling with finite 115 116 element analysis (Guessasma et al., 2011). Most previous work reported in the literature has involved the measurement of the strength and fracture properties of biscuits using the 117 118 standard three point bending tests, (Saleem, 2005). The aim of this work was to examine the 119 dispersion in breakage force and breakage pattern for uncracked and cracked biscuits in an 120 axisymmetric bending load test. In addition, the effect of crack geometry on breakage force was explored and experimental tests used to identify an appropriate theoretical model of 121 122 biscuit failure.

- 123
- 124

2. THEORY

125 2.1 Biscuit Loading

Each biscuit was loaded by applying a point force at its centre while its circumference rested on a smooth circular ring as illustrated in figure 1. This method is not representative of the actual loading of biscuits during manufacture, storage and transportation; however it provided a rational basis to quantify the bending strength of circular biscuits. Based on this arrangement, each biscuit was considered a thin, flat circular plate, simply supported around its perimeter and loaded by a concentrated force, P applied at its centre. The lower surface of the biscuit is in a state of tension due to the induced two-dimensional bending response of the

biscuit with two orthogonal, normal, tensile stresses in the radial and circumferential
directions respectively. These can be predicted as follows (Benham et al., 1996):

136
$$\sigma_r = \frac{3P(1+\upsilon)}{2\pi t^2} \ln\left(\frac{R}{r}\right) \qquad \sigma_\theta = \frac{3P(1+\upsilon)}{2\pi t^2} \left[\ln\left(\frac{R}{r}\right) + \frac{1-\upsilon}{1+\upsilon}\right] \quad \text{for} \quad r \neq 0 \tag{1}$$

137

The circumferential (tangential) stress acts in the perpendicular direction to any radial line 138 139 while the radial stress is perpendicular to a circumferential curve. Both stresses decrease rapidly with radial distance and are considerably lower at the edge than at the centre of the 140 141 biscuit. The circumferential stress is always the larger of the two and the fractional difference 142 between σ_{θ} and σ_{r} increases when moving from the centre of the biscuit to the perimeter. The predictions for stress in equation 1 exhibits a discontinuity at the origin (r = 0) and tend to 143 144 infinite magnitudes at that position. This arises because the load is considered to act at a point 145 whereas in reality the load is applied over a small central area of radius, r₀. It has been shown that the maximum stresses in the plate (at either surface) are limited to the following levels 146 147 (Young, 2001)

148

149
$$\sigma_r = \frac{3P(1+\nu)}{2\pi t^2} \ln\left(\frac{R}{r_0}\right) \qquad \sigma_\theta = \frac{3P(1+\nu)}{2\pi t^2} \left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{1+\nu}\right]$$
(2)

- 150
- 151 where r_0 ' is the equivalent loading radius defined as

152

153
$$r_0' = \sqrt{1.6r_0^2 + t^2} - 0.675t$$
 (3)

154

155 Hence in the central region of the biscuit ($0 < r < r_0$), if the predicted values of σ_{θ} and σ_r 156 from equation 1 are in excess of those estimated using equation 2, they are replaced by the 157 latter values. Figure 2 illustrates the variation of tangential and radial stress with radial 158 distance from the biscuit centre to the edge using data representative for the study.

159

The stress equations require knowledge of the Poisson's Ratio for these biscuits. Kim et al., (2012), suggested a value of 0.2 as appropriate for this material. This is an estimate but a sensitivity analysis revealed that the level of uncertainty in the correct magnitude of Poisson's Ratio does not meaningfully affect the predictions of stress; the fractional variation

in resultant stress is considerably lower (more than 50 %) than the fractional uncertainty in
Poisson's Ratio. It should also be noted that material behaviour was considered to be
isotropic and so only a single value for the Poisson's Ratio was needed.

167

168 2.2 Maximum Tensile Perpendicular Stress

169 For the subsequent fracture mechanics analysis, the maximum tensile stress acting perpendicular to any line segment, σ_{\perp} of length 2a on the biscuit lower surface must be 170 estimated. For a line (crack) acting in the radial direction, this stress will be the tangential 171 stress at the point in the crack closest to the biscuit centre, as illustrated in figure 3a. If the 172 173 line passes through the centre, σ_{\perp} will coincide with the limiting stress of equation 2. For a line acting in the tangential direction, the situation is more complex. At the mid-point of the 174 175 line, the perpendicular stress is the radial stress at that location. At any other point along the 176 line, the perpendicular stress is a function of both the radial and tangential stress at the 177 location in question. At any distance, x, $(x < r_c)$ along the line that subtends an angle θ as 178 shown in figure 3b, the tensile perpendicular stress will be (Benham et al., 1996):

179

180
$$\sigma_{\perp} = \sigma_{\theta} \frac{1 + \cos 2\theta}{2} + \sigma_r \frac{1 - \cos 2\theta}{2}$$

181

182 Using the trigonometric relationships $r = \sqrt{r_c^2 + x^2}$ $\tan \theta = \frac{r_c}{x}$ (5) 183

184 The perpendicular stress along the line can be expressed as

185

186
$$\sigma_{\perp} = \frac{3P(1+\upsilon)}{2\pi t^2} \left\{ \ln R + \frac{1-\upsilon}{1+\upsilon} \frac{x^2}{r_c^2 + x^2} - \ln(r_c^2 + x^2)^{\frac{1}{2}} \right\} \quad 0 \le x \le a$$
(6)

187

Equation 6 gives the tensile perpendicular stress at any point along a tangential line at a distance, x from the midpoint of the crack. Figure 3c illustrates how σ_{\perp} varies with distance along the line. The perpendicular stress rises from a value equal to the radial stress at the midpoint, reaches a maximum value at some distance along the line and then falls off. For a line segment defined by an angle other than 0 (radial line) and 90° (tangential line), the perpendicular stress has a more complicated relationship with distance along the line and wasevaluated numerically in this paper.

195

196 2.3 Biscuit Failure Criteria

To predict failure, the stress state in the biscuit must be combined with a valid failure criterion. A biscuit is linearly elastic and breaks suddenly with minimal plastic deformation indicating that it can be considered as a brittle material once its temperature is below the glass transition temperature for the product. For the digestive-type biscuits of this study, the glass transition temperature, T_g is well above room temperature (T_g = 62.8°C) (Kawai et al., 2014). The following cases were considered; 1] the biscuit contained no cracks or defects and 2] cracks were present.

204

For the former, the appropriate failure theory is the maximum principal stress theory and for the loading situation described above, this corresponds to the maximum circumferential stress at the centre of the biscuit (predicted by equation 2) exceeding the tensile strength of the biscuit at failure, σ_f (Haghighi and Segerling, 1988).

209

210
$$\sigma_{\max} \ge \sigma_f \implies failure$$

In other words the biscuit is predicted to fail when the applied load, P is sufficient to ensure the tangential stress, σ_{θ} at the biscuit centre reaches a maximum value that equals the material strength of the biscuit, σ_{f} .

(7)

214

If the biscuit contained cracks, each crack was considered to have the geometry of a straight, 215 216 part-through crack (crack depth being limited to less than half biscuit thickness) of finite length (crack half-length being limited to less than biscuit radius). A crack is defined by the 217 following four geometric properties; length 2a, depth w, radial distance from biscuit centre 218 219 to the midpoint of the crack, r_c and angle subtended at the mid-point between a radial line and the crack line. The geometry is illustrated in figure 4. A crack or check will propagate if a 220 221 sufficiently large in-plane tensile stress is applied normal to the crack plane (assuming mode 222 1 fracture i.e. the crack opening mode by tension). Specifically if the stress intensity factor, K_{I} exceeds the critical stress intensity factor, K_{IC} for the crack, then failure by fracture is 223 224 predicted, (Van Vliet, 2014).

$$226 K_I \ge K_{IC} \implies failure (8)$$

227

For this paper, the stress intensity factor proposed by Rice & Levy (1972) for a part-through surface crack of finite length in an elastic plate under a bending load was selected as the closest analysis to our case. K_I is shown to be a function of plate (biscuit) thickness, the magnitude of the perpendicular stress, the ratio of crack depth to plate (biscuit) thickness and the ratio of crack length to plate thickness:

233

234
$$K_I = g_1 g_2 \sigma_\perp \sqrt{t}$$
 (9)

235

236 For the analysis it is assumed that the presence of a crack does not significantly change the 237 overall stress distribution and only the local distribution in the crack region. Moreover as 238 biscuit thickness to biscuit diameter ratio is equal to 0.1, the biscuit is treated as a thin 239 cylinder subject to plane stress. The approximations made in discounting the 3D nature of stress in the structure were discussed more fully by Rice & Levy (1972) and by Yang & 240 Shiva (2011). The dimensionless factors g_1 and g_2 are derived from the semi-analytical 241 analysis presented in the work of Rice & Levy (1972). Specifically g_1 is solely a function of 242 243 the ratio of crack depth to plate thickness while g_2 is additionally a function of crack length to 244 plate thickness ratio. The following modified 4th order polynomial was used to express the g₁ 245 parameter in terms of relative crack depth

246

247
$$g_1 = \sqrt{\frac{w}{t}} \left(1.99 - 2.47 \frac{w}{t} + 12.97 \left(\frac{w}{t}\right)^2 - 23.17 \left(\frac{w}{t}\right)^3 + 24.80 \left(\frac{w}{t}\right)^4 \right)$$
(10)

248

while the g_2 parameter can be represented by a fitted empirical equation (for the range of data of interest to this paper) extracted from Rice & Levy (1972) of the form

251

252
$$g_2 = c_1 \ln(2a/t) + c_2$$
 (11)

253

where the constants c_1 and c_2 depend on the relative crack depth (w/t). Table 1 gives the magnitudes of these constants for a number of relative crack depth ratios. The g_1 parameter

increases monotonically from 0 to close to 1.5 as the ratio of crack depth to biscuit thickness increases from 0 (very shallow crack) to 0.5 (deep crack) quantifying the influence of crack depth on the stress intensity factor. For shallow cracks, the g_2 parameter and hence stress intensity factor is relatively insensitive to crack length but for deeper cracks, the stress intensity factor will increase significantly with crack length. Overall, the longer and deeper the crack, the larger is the stress intensity factor and the greater the likelihood of biscuit breakage.

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- 264

265 2.4 Determination of Critical Crack Size

266 The presence of cracks weakens the biscuit by reducing the required breakage force. Critical crack size is the crack dimension at which the biscuit will fail by fracture rather than 267 268 overloading. The seriousness of a crack depends on its size (depth and length) and its radial 269 location and angular orientation on the biscuit surface. Qualitatively from the stress intensity 270 factor approach, the deeper the crack, the longer the crack, the more central the crack and the 271 more radial in inclination, the lower is the required breakage force. Also while all these factors affect biscuit integrity, crack depth is more influential than crack length in 272 273 determining the response and crack radial location is more significant than crack orientation. However because of the complexity of the stress intensity factor model and the non-274 275 uniformity of the stress distribution in the biscuit, it is not possible to produce a simple 276 analytical formula for critical size for a general crack.

277

For the restricted case of a very shallow crack that is long relative to biscuit thickness, an analytical approach can be conducted. In this situation, for the selected SIF (Stress Intensity Factor) of equation 9, the g₁ parameter is approximately equal to $2\sqrt{\frac{w}{t}}$ while the g₂ parameter has a value of almost 1. Hence the SIF is solely dependent on crack depth, w and is given as

$$283 K_I = 2\sigma_\perp \sqrt{w} (12)$$

284

Also if this crack passes through the central region of the biscuit, where both the tangential and radial stress are limited and furthermore in this region the perpendicular stress is almost equal to the average of the radial and tangential stresses and can be approximated as

288

289
$$\sigma_{\perp} = \frac{3P(1+\nu)}{2\pi t^2} \left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{2(1+\nu)} \right]$$
 (13)

290

291 Thus the criterion for biscuit failure by fracture will be

292

293
$$\frac{3P(1+\nu)}{2\pi t^2} \left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{2(1+\nu)} \right] 2\sqrt{w} = K_{IC}$$
(14)

294

While the criterion for failure by overloading is when the maximum stress (equal to the limited tangential stress) equals the failure stress

$$298 \qquad \frac{3P(1+\upsilon)}{2\pi t^2} \left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{1+\upsilon} \right] = \sigma_f \tag{15}$$

299

By combining equations 14 and 15, it was possible to estimate the necessary crack depth thatwill cause failure by fracture rather than overloading to occur

$$302 \qquad w \ge \left\{ \frac{\left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{(1+\upsilon)}\right] K_{IC}}{\left[\ln\left(\frac{R}{r_0}\right) + \frac{1}{2(1+\upsilon)}\right] 2 \sigma_f} \right\}^2 \tag{16}$$

303

Thus when the magnitude of crack depth exceeds the value predicted by the expression in equation 16, failure by fracture is predicted to occur. This formula is applicable to a long, shallow crack that passes through the central zone (where the load is applied) of any angular orientation.

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- 309

310

3. MATERIALS & METHODS

311 3.1 Materials

The biscuits used in this study were obtained from a commercial manufacturer (own brand supermarket variety) and were of the digestive type. Typical composition was 22.3 % fat, 18.8 % sugars, 3.5 % fibre, 6.7 % protein and 1 % salt. The average moisture content was

315 measured by the oven dry test and found to be 1.55 ± 0.19 % wet basis. The average porosity 316 was measured using the Kawas and Moreira (2001) approach of using values of the bulk 317 density and the solid density of the biscuit. Bulk density was obtained using a modified 318 Archimedean method replacing the displacement of a fluid for the displacement of 1 mm 319 solid-glass spheres (Consolmagno and Britt, 1998); solid density was measured by placing a 320 fragment of the biscuit in a compressed helium gas multivolume pycnometer (Micromeritics, 321 Model 1305, USA). These tests were performed in five replicates. Each biscuit tested had its 322 diameter and thickness recorded in which the average of three readings were taken with a 323 Vernier digital caliper (Mitutoyo, model 500-151-30, Japan).

324

325 The biscuits were divided into three classes. The first included all biscuits containing no 326 visible checks these being most of the biscuits. The second class consisted of those biscuits that had visible checking. The checks were all superficial i.e. very shallow on the surface of 327 328 the biscuit. There were many checks (ten or more) on each biscuit. Each path was random in orientation and jagged as opposed to straight. These checks were present on all locations of 329 330 the biscuit surface i.e. close to the edge and near the centre. Typical lengths ranged from 10 331 mm to 30 mm. Finally tests were conducted on biscuits that had pre-defined cracks placed in 332 them to explore the effect of cracks on biscuit structural response. The cracks were made with 333 a thin blade having a thickness of 2 mm. Most of the cracks were either radial or tangential in 334 orientation though a small number of tests were done with other crack angles. In total, 18 different crack types (labelled A to R) were investigated with 15 replicates used for each 335 336 type. Table 2 lists the geometrical parameters (orientation, length, mid-point location and 337 depth) for each crack type. Additionally the geometries of each crack type are graphically 338 displayed in figure 5 where the length and orientation of the crack and the distance from its midpoint to the biscuit centre (when non-zero) are indicated by the arrowed line. 339

340

341 3.2 Three Point Bending Tests

The standard three point bending tests was first carried out to obtain values for the material properties of failure stress, σ_f and the critical stress intensity factor, K_{IC}. Prismatic specimens were cut from the biscuits with rectangular sides of 60 mm by 20 mm and 7.2 mm thick. These were supported on parallel bars, 40 mm apart and loaded by a third bar (line load) equi-distant between the two support bars. The tests were performed on a Texture Analyser (TA.HDplus, Stable Micro Systems, UK). In total, 20 samples were tested. Force versus 348 deflection was recorded until the biscuit specimens broke. The failure stress can be quantified349 from

350

$$351 \qquad \sigma_f = \frac{3PL}{2bt^2} \tag{17}$$

352

where P is the measured load at failure, b is the cross section dimension of the beam (20 mm), t the beam depth (7.2 mm) and L the bar spacing or beam span (40 mm).

355

Experiments were also conducted with these specimens to estimate the fracture toughness. A total of five samples were used. The sample was re-orientated so that equivalent beam depth, t was 20 mm and beam cross section dimension, b 7.2 mm. A notch of 10mm depth and running through the thickness of the sample was made at the bottom face. A line load was applied at its centre of the span and for this work the supports spacings, L were 45 mm apart. Fracture toughness or critical stress intensity factor, K_{IC} was quantified in accordance (ASTM, 2008)

363

$$364 K_{IC} = \frac{PL}{bt^{\frac{3}{2}}} f\left(\frac{w}{t}\right) (18)$$

365

366 with
$$f\left(\frac{w}{t}\right) = \left[1.9 - \frac{w}{t}\left(1 - \frac{w}{t}\right)\left(2.15 - 3.93\frac{w}{t} + 2.7\left(\frac{w}{t}\right)^2\right)\right] \frac{3\sqrt{\frac{w}{t}}}{2\left(1 + 2\frac{w}{t}\right)\left(1 - \frac{w}{t}\right)^{1.5}}$$
 (19)

367

368 Owing to limitations of possible sample dimensions, the adopted test procedure was not in 369 strict accord with ASTM specification for the measurement of fracture toughness (as the 370 beam depth of 20 mm was too low). Hence the estimated levels of K_{IC} can only be regarded 371 as indicative.

372

373 3.3 Axi-Bending Tests

Regarding the axi-symmetric bending tests, three sets of loading tests were performed on the texture analyser for biscuits with no visible checks, for biscuits with naturally occurring

376 checks and lastly for biscuits with pre-defined checks. In each case, the biscuit was 377 supported by resting on a thin steel ring with a circumference of 34 mm radius. The loading 378 indenter with a tip radius, r_0 of 3 mm was applied at the centre of the biscuit. From equation 3 379 (using a biscuit thickness of 7.2 mm) the equivalent loading radius r'_0 was 3.28 mm which 380 limits the maximum stress at the biscuit inside this zone. Loading speed was 1 mm/s. Force 381 versus deflection was measured up to the point of breakage. The broken biscuit was 382 photographed after fracture and the crack shape and fragment distribution analysed. For some 383 tests, high speed photography was employed to investigate the dynamic progression of the 384 crack at the point of breakage.

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4. RESULTS & DISCUSSION

388 4.1 Physical & Mechanical Properties of Biscuits

389 The variation in diameter and thickness between biscuits was found to be described by the 390 Normal distribution. Mean and standard deviation for diameter were 71.7 ± 0.9 mm 391 respectively and for thickness 7.2 ± 0.3 mm, respectively. Bulk density was 463.18 kg/m³, 392 solid density 1401.4 kg/m³ and hence porosity was estimated to be 67 % (0.67). From the 393 three-point bending tests, the tensile strength of the biscuits had a mean value of 420 kPa and 394 standard deviation of 31 kPa, (420 ± 31) . These were in good agreement with values reported 395 in the literature for semi-sweet biscuits (Kim et al., 2012), Ahmad (2001) and Saleem (2005). The average K_{IC} value obtained from notched bending test was 18 kPam^{0.5} with a standard 396 deviation of 3.0 kPam^{0.5} (18.0 \pm 3.0). While this value can only be regarded as an estimate, it 397 398 does lie at the lower end in the range of values reported by Kim et al., (2012). The coefficient 399 of variation is considerably larger for the fracture toughness than for tensile strength 400 indicating a much higher level of natural dispersion for the former quantity. While this may 401 reflect experimental sample size effects, it could also indicate that fracture toughness is more 402 sensitive to the random and heterogeneous structure of the biscuit than tensile strength.

403

404 4.2 Failure of Un-Checked Biscuits

In total over 160 biscuits with no visible defects were loaded under axisymmetric bending until failure occurred. The distribution in maximum breakage force is shown in frequency histogram form in figure 6. The average magnitude of the breakage force was 12.5 N, the standard deviation was 1.2 N (12.5 ± 1.2) and it ranged from a minimum of 9.7 N to a

409 maximum value of 15.3 N. The distribution in failure force can be represented by the Normal 410 distribution. Video analysis indicated that the cracks tended to start where the load was 411 applied and propagate out along the radial direction to the edge. All the biscuits fractured in 412 tension along a lower line at the lower surface where maximum stresses are predicted. Biscuit 413 breakage patterns conformed to three basic types; two radial cracks (either collinear or non-414 collinear); three radial cracks and four radial cracks. Each type is illustrated in figure 7. The 415 majority of the biscuits, 67 % failed with the formation of three cracks while 26 % produced 416 two cracks with only 7 % giving four radial cracks. Higher breakage forces are associated 417 with a larger number of fracture planes but not at a statistically significant level.

418

419 The validity of the proposed failure criterion for un-checked or sound biscuits given by 420 equation 7 was then checked. A biscuit will fail when the predicted maximum stress (the limited circumferential stress predicted using equation (2) exceeds the tensile strength value 421 422 reported in section 4.1 above. The issue is complicated because the measured value of tensile strength is statistically distributed and the distribution in biscuit thickness (and any other 423 424 parameters in equation 2) affects the predicted stress value. Hence the validity of equation 7 must be tested statistically. The experimentally measured tensile strength (from the 3 point 425 426 bending test) is 420 ± 31 kPa. The predicted failure stress (obtained from equation 2 using the 427 measured failure force) is 438 ± 42.1 kPa. While the predicted failure stress value is larger 428 than the experimentally measured failure stress, the difference between them is not 429 statistically significant (at the 5 % confidence level using the t statistic) demonstrating that the criterion of equation 7 is valid. 430

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- 432

433 4.3 Failure of Checked Biscuits

434 Breakage force for biscuits with the presence of checking was also recorded. In total 50 435 biscuits exhibiting checking were loaded and broken. Figure 8 illustrates such checked 436 biscuits. Mean breakage force for the biscuits was 9.6 N and the standard deviation 2.62 N 437 (9.6 ± 2.62) compared to (12.5 ± 1.2) N respectively for unchecked biscuits. The average 438 breakage force for checked biscuits is 23 % lower than for the unchecked product demonstrating the significant influence of checking on biscuit integrity. Moreover the 439 440 standard deviation in breakage force for checked biscuits is over twice as large as for 441 unchecked indicating much greater dispersion in strength which is also an adverse quality

442 feature. Caution is required in interpreting these results because of the very heterogeneous 443 nature of the checks that were present and lack of accurate characterisation of their precise 444 geometry. Nonetheless it is clear that when checking occurs, it has a major impact on biscuit 445 strength and resistance to breakage. No theoretical analysis was carried out into the failure of 446 these biscuits. Each biscuit tended to have more than one check on its surface and each check 447 had a complex, tortuous path precluding any analytical application of fracture mechanics 448 theory. However these results clearly demonstrate that defects such as checks significantly 449 affect biscuit strength and hence quality and underlie the importance of investigations in this 450 area.

451

452 4.4 Failure of Cracked Biscuits (Experimental)

453 In total, 18 different crack types were investigated. For each type, both the mean and standard 454 deviation in breakage force was quantified. In addition, the reason for failure (overloading or 455 crack propagation) was noted. The presence of a crack in the biscuit does not automatically 456 mean that failure is as a result of crack propagation when the load is applied; if the maximum 457 stress in the biscuit exceeds the failure stress before the local stress intensity factor exceeds the critical stress intensity factor, then failure is due to overloading. There are two aspects to 458 459 breakage that indicate the failure mode; the location of the fracture plane and the magnitude of the failure force. If the fracture plane initiates at the biscuit centre (where stress is 460 461 maximum), this is indicative of failure due to overloading as is the case for an uncracked biscuit. If the fracture plane initiates at the defined crack, then this can be taken to be failure 462 463 resulting from crack propagation. Also an uncracked biscuit requires a breakage force of 12.5 N. Breakage forces in this region are indicative of failure by overloading while as the 464 465 measured breakage force falls away from these levels, failure by crack propagation is more likely. Owing to the intrinsic variability in the breakage force (which ranges from 10 N up to 466 467 15 N), this parameter alone is not a definitive indicator. Table 3 summarises the experimental 468 results for the cracks giving the crack type, breakage force statistics and failure mode. At the 469 top of the table the corresponding results for a biscuit without cracks are shown for 470 comparison.

471

472 As shown in table 3, crack types A, B, C, D are all radial cracks, with a midpoint at the 473 biscuit centre and 1 mm deep. The breakage force fell consistently from 11.76 N (crack A) to 474 8.27 N (crack D) as crack length increased from 5 mm to 70 mm respectively. Crack types E,

475 F, G, H differed from the above by just being 2 mm deep. Breakage force followed the same 476 pattern as above, falling from 9.76 N (crack E) to 6.5 N (crack H) though in all cases was 477 lower reflecting the fact that the cracks are deeper and so the biscuits failed more easily. For 478 these eight crack types, failure was by crack propagation. Figure 9 plots breakage force 479 versus crack length for the two crack depths that were analysed. There is a definite, almost 480 linear, relationship between breakage force and the length of the crack and a clear 481 relationship between crack depth and breakage force. Long, deep cracks can reduce the 482 strength of a biscuit by up to 50 % compared to an uncracked biscuit.

483

Cracks I, J and K are also radial though all having a midpoint at 15 mm from the biscuit 484 485 centre so the local stress at the crack will be lower than for the previous eight cracks. Crack 486 type I was short (5 mm in length) and the biscuit did not fail by crack propagation but by overloading with a high breakage force of 11.82 N. The biscuits with crack types J and K 487 failed by crack propagation with the breakage force of close to 9 N. Breakage force is 488 489 generally higher for these three crack types than the previous radial cracks because they are 490 less heavily stressed being away from the biscuit centre. Finally crack type L is also radial 491 though short in length (5 mm), relatively shallow (1 mm) and quite removed from the biscuit 492 centre with its midpoint at a radial distance of 27.5 mm. Thus the stress at it is relatively low 493 and hence the biscuit failed by overloading with a high breakage force of 9.92 N.

494

Crack types M, N, O and P are all tangential in orientation. For tangential cracks, radial stress 495 496 will be the critical perpendicular stress which is lower than tangential stress. These cracks are 497 all 1 mm deep but the length and mid-point radial location vary. Crack types M, N and O are 498 at a considerable distance from the biscuit centre where the maximum stress acting on the 499 crack is low and so the biscuits all failed by overloading with the breakage force always 500 exceeding 9 N. Only crack type P which was 5 mm from the centre caused the biscuit to fail 501 by crack propagation and had the lowest breakage force of the four types of tangential crack. 502 Finally the table gives the data for two cracks types (types Q and R) whose midpoint orientations are defined by the angles of 30° and 60° respectively. For these cracks it was not 503 504 possible to definitively state the failure mechanism although the crack pattern was more 505 indicative of failure by overload. For all the crack types explored, the standard deviation in 506 breakage force was of the same order of magnitude as that for an uncracked biscuit (1.2 N). 507 Hence the presence of a single crack in the biscuit did not appear to promote any greater dispersion in biscuit breakage characteristics but rather just acted to lower the mean breakageforce.

510

511 4.5 Failure of Cracked Biscuits (Theoretical)

512 The validity of the failure criterion expressed by equation 8 was examined. Failure by 513 fracture occurs when the calculated stress intensity factor, K_I equals the critical stress 514 intensity factor (fracture toughness), K_{IC} of the biscuit. Because the magnitude of breakage 515 forces for each crack is distributed (as quantified by the standard deviation in table 3) and the 516 fracture toughness of the material itself varies (estimated mean value of 18 kPam^{0.5} with standard deviation of 3 kPam^{0.5}), the level of agreement must be quantified statistically. From 517 518 the experimental analysis presented in table 3, the cracked biscuits where failure was by fracture rather than crack propagation were identified. Table 4 summarises the results for 519 these biscuits giving the crack identifier, the magnitudes of the relative depth, w/t and relative 520 length, 2a/t, the maximum perpendicular stress at the crack (obtained from the mean value of 521 522 breakage force in each case and using equation 6) and the corresponding stress intensity 523 factor. Generally the theoretical predictions agree very well with the experimental findings with the average stress intensity factor for each crack type being close to the mean level of 524 fracture toughness (18 kPam^{0.5}). The only exceptions are the three crack types E, J and P. 525 The reason for the discrepancy for crack E is the inability of the Rice & Levy method to 526 527 calculate the correct magnitude of the stress intensity factor for very short cracks. For crack types J and P, where the calculated stress intensity factor is considerably less than the fracture 528 529 toughness the reason for the poor agreement is unknown but could reflect a statistical outlier effect. To assess the failure criterion more rigorously, the data is displayed graphically in 530 531 figure 10. The stress intensity factor for each crack type is shown with error bars 532 corresponding to ± 1 standard deviation. Additionally the fracture toughness (critical stress 533 intensity factor) is indicated by a solid line with the broken line representing its ± 1 standard 534 deviation limits. Differences between the means of the stress intensity factor for each crack 535 type and the mean critical stress intensity factor are quite small compared to the variability in K_I within each crack type (apart from types E, J and P). Applying the F statistic from 536 537 ANOVA, demonstrated that the validity of the failure model (equation 8) could be accepted at the 5 % confidence level. 538

539

540 For all biscuits containing cracks, the mode of failure (fracture versus overloading) can be 541 predicted by two loading ratios; the SIF ratio (K_I/K_{IC}) for the former and the failure stress 542 ratio (σ_{max}/σ_f) for the latter. Whichever ratio is closer to 1, should determine the failure 543 response; crack propagation for the former and overloading for the latter. This is the case for 544 most of the crack types with failure by overloading occurring when the stress ratio is high and 545 the stress intensity ratio relatively low while failure by crack propagation occurs for the 546 reverse condition. Figure 11 gives a scatter plot of the two failure criteria for each crack type. 547 Biscuits that failed by crack propagation are indicated with square markers and those by 548 overloading with triangular markers. There is good demarcation between the failure 549 mechanisms with biscuits that failed by overloading lying at the lower, right quadrant and 550 biscuits that failed by crack propagation at the upper, left quadrant. Because the stress ratio 551 lies with quite tight limits of 0.7 and 1.2 while the SIF ratio varies more widely (between 0.2 552 and 1.2,), the influence of the stress ratio, while present, is more difficult to discern. Finally regarding the critical crack size analysis of Section 2.4, inputting the data for our work gives 553 a magnitude for the critical crack depth of 0.61 mm. In other words any long crack passing 554 555 through the central zone with radius r_0 ' (3.28 mm) that is deeper than 0.61 mm should result 556 in failure by fracture. This is confirmed by the experimental data of this work.

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5. CONCLUSIONS

560 This work primarily has explored the force needed to break sound and cracked biscuits. It 561 also measured the breakage pattern (number and size of fragment pieces). In particular, the 562 structural behaviour of circular biscuits supported around the circumference and loaded by a 563 central concentrated force has been examined. For a biscuit without defects, breakage 564 occurred when biscuit stress reached or exceeded the failure stress. For a biscuit with defects 565 such as checks or cracks, breakage occurred as above or alternatively when the critical stress 566 intensity factor was reached. For the latter case, the breakage force was considerably reduced 567 showing that cracks or checks considerably weaken the strength and integrity of the biscuit. 568 Furthermore, a stress intensity factor model to quantify the effect of a crack on biscuit 569 response has been proposed and verified. The effect of a crack on biscuit strength is 570 dependent on crack depth, length, orientation and location on the biscuit surface. Shallow, 571 short, radial cracks near the biscuit centre are more injurious to its integrity than deep, long 572 cracks out near the biscuit circumference. Variability in biscuit properties, principally the

573 tensile strength and critical stress intensity factor, complicate the issue and explain the scatter 574 in the data. Biscuit breakage behaviour is closely connected to the quality parameter of 575 texture. The results are relevant to understanding the maintenance of biscuit integrity through 576 the post-manufacture, supply, distribution and transport chain that the biscuit endures prior to 577 sale.

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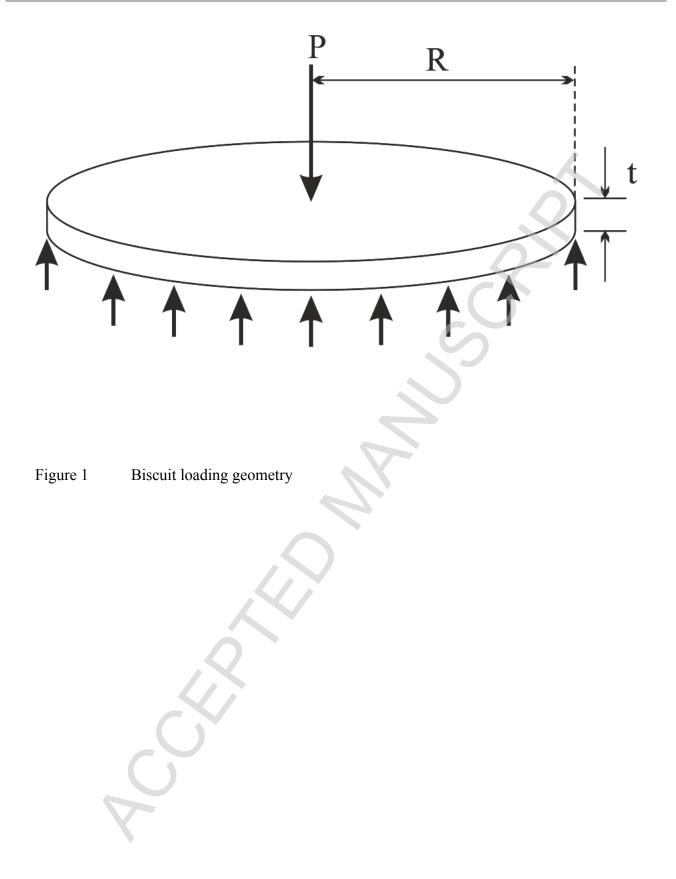
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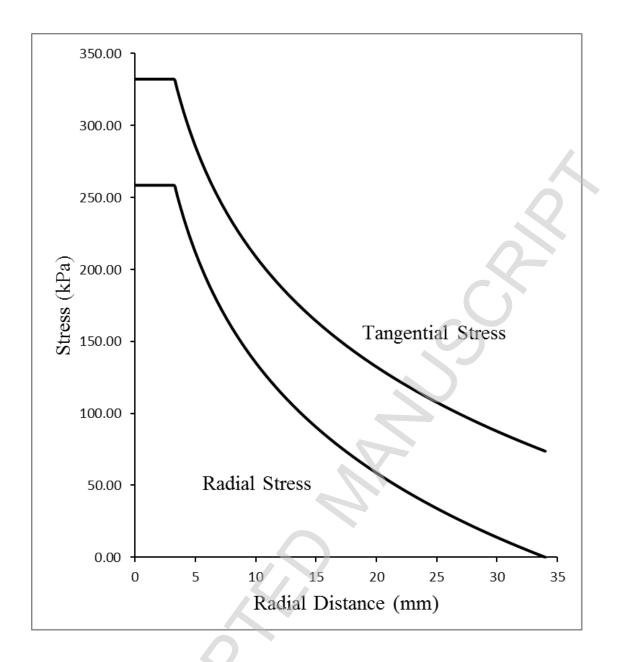


Figure 2: Typical variation of tangential and radial stress with radial distance (R = 34 mm, t = 7.2 mm, $r_0 = 3$ mm, v = 0.2, P = 10 N).

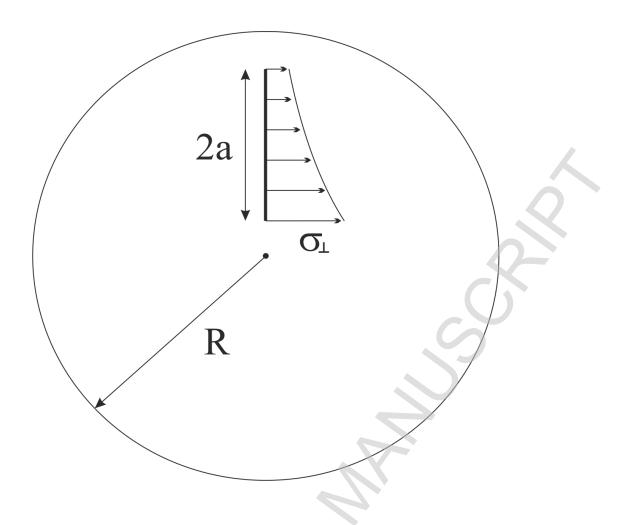
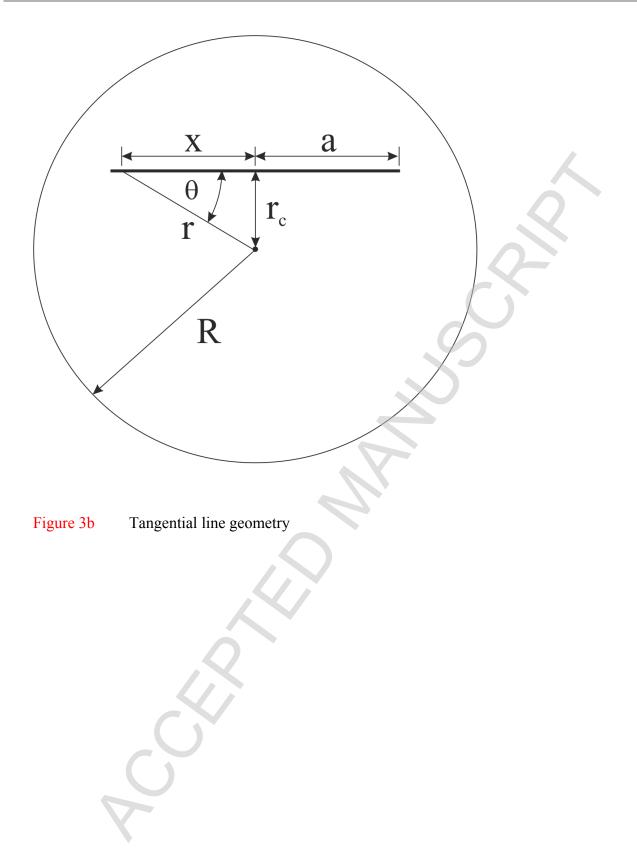
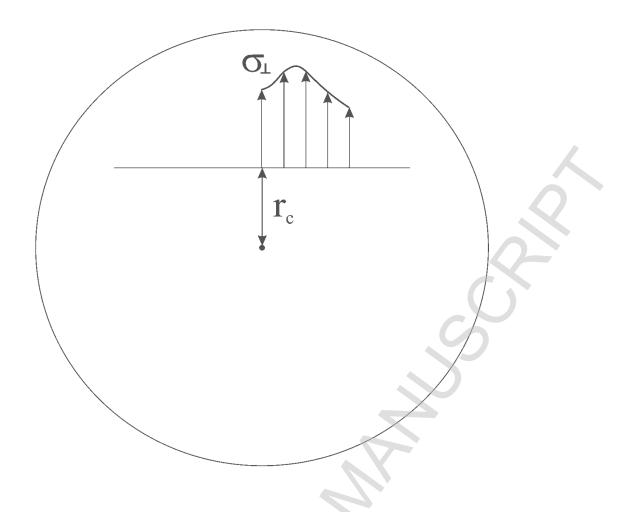
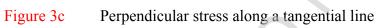
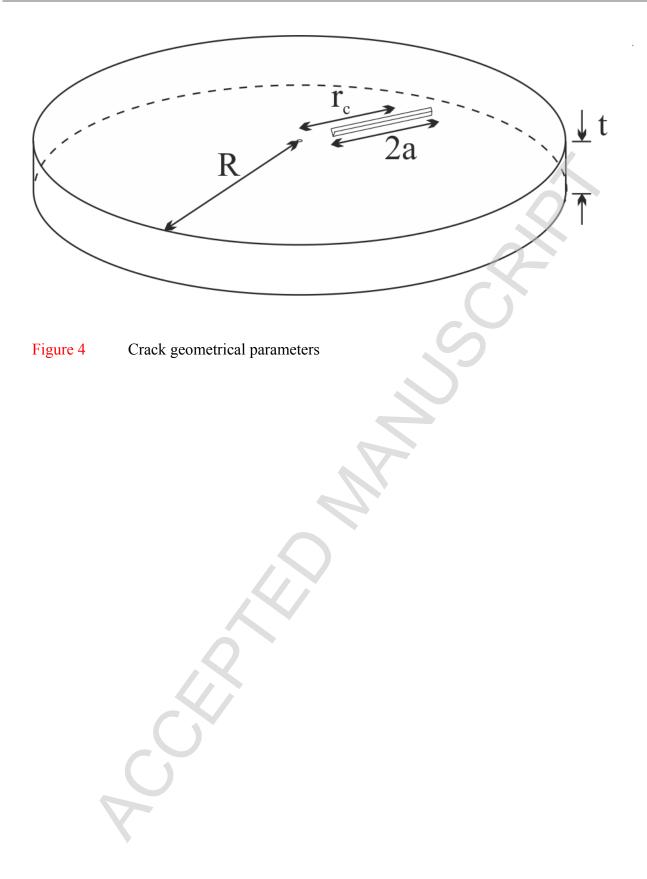


Figure 3aPerpendicular stress along a radial line









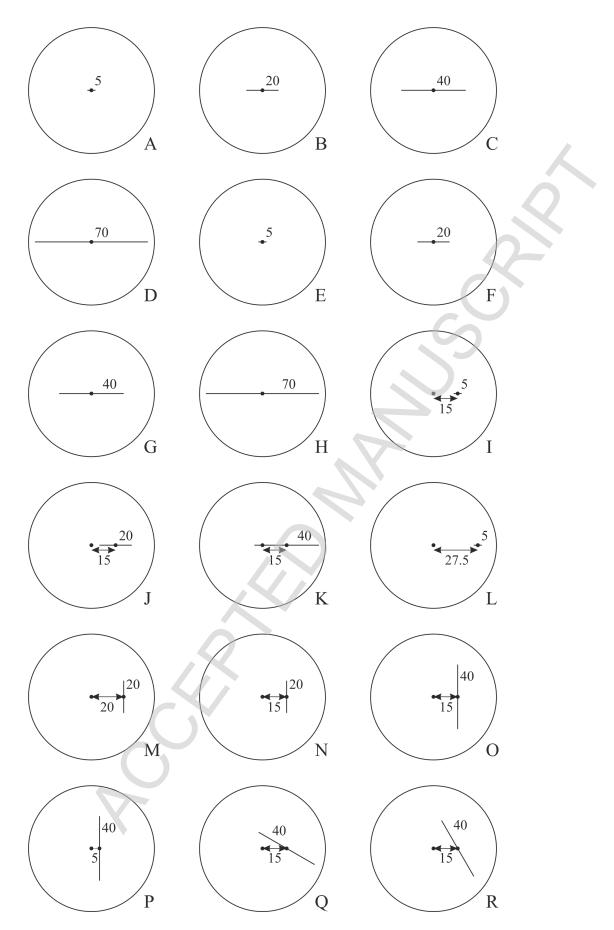


Figure 5: Crack geometries

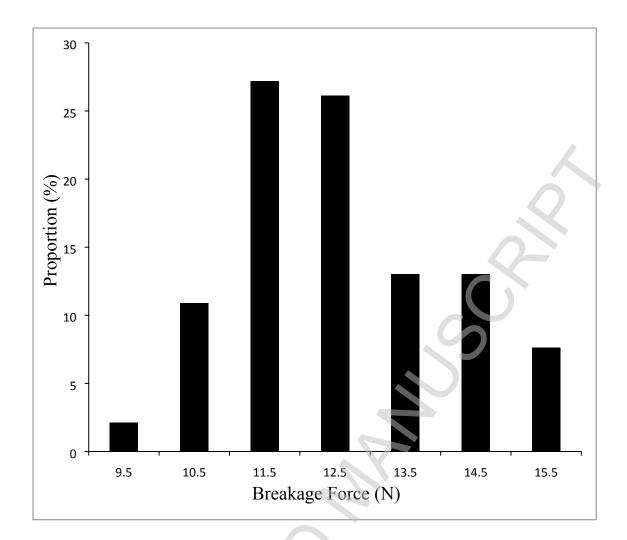


Figure 6 Distribution of the breakage force for biscuits without checks



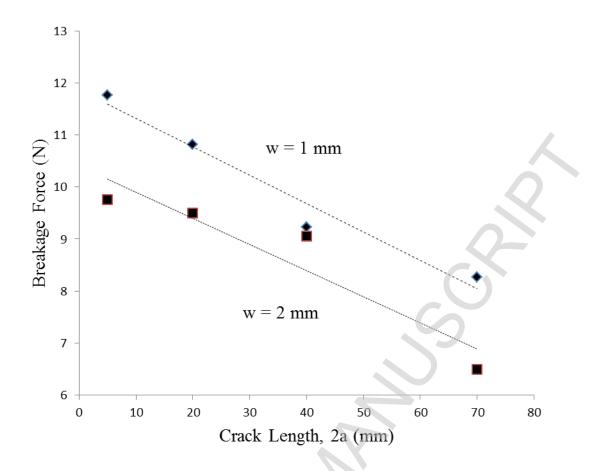


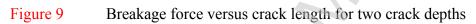


Figure 7 Breakage modes for biscuits without checks (Two radial cracks, Three radial cracks, Four radial cracks)



Figure 8 Checked biscuits





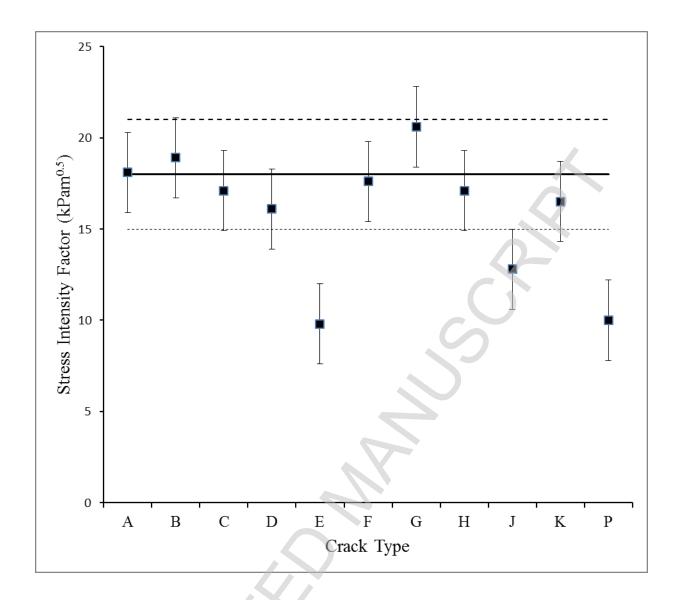
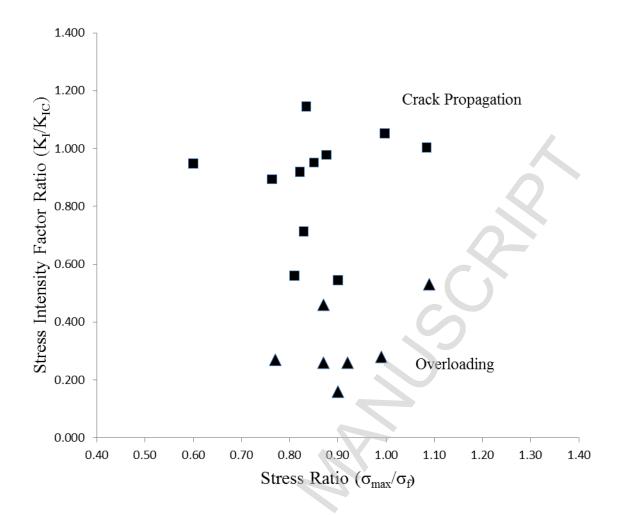


Figure 10: Failure analysis of cracked biscuits



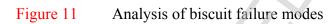


Figure List

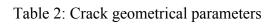
Figure 1	Biscuit loading geometry
Figure 2	Typical variation of tangential and radial stress with radial distance
Figure 3a	Perpendicular stress along a radial line
Figure 3b	Tangential line geometry
Figure 3c	Perpendicular stress along a tangential line
Figure 4	Crack geometrical parameters
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- Figure 9Breakage force versus crack length for two crack depths
- Figure 10 Failure analysis of cracked biscuits
- Figure 11 Analysis of biscuit failure modes

Relative crack depth (w/t)	c ₁ Parameter	c ₂ Parameter
0.14	0.0755	0.7795
0.28	0.2004	0.4006
0.46	0.2105	0.0869

Table 1: Values of the fracture model parameters c_1 and c_2 (extracted from Rice & Levy, 1972)

Crack Identifier A B C D C D E F G H H J J K	Crack Orientation Radial Radial Radial Radial Radial Radial Radial Radial	Mid-Point Location mm 0 0 0 0 0 0 0 0 0 0 0	Crack Length mm 5 20 40 70 5 20 40 40 70	Crack Depth 1 1 1 2 2 2 2 2
A B C D E F G H I J	Radial Radial Radial Radial Radial Radial Radial	mm 0 0 0 0 0 0 0 0 0 0	mm 5 20 40 70 5 20 40	mm 1 1 1 1 2 2 2
B C D E F G H I J	Radial Radial Radial Radial Radial Radial	0 0 0 0 0 0 0 0	5 20 40 70 5 20 40	1 1 1 2 2
B C D E F G H I J	Radial Radial Radial Radial Radial Radial	0 0 0 0 0 0 0	20 40 70 5 20 40	1 1 1 2 2
C D E F G H I J	Radial Radial Radial Radial Radial	0 0 0 0 0 0	40 70 5 20 40	1 1 2 2
D E F G H I J	Radial Radial Radial Radial	0 0 0 0	70 5 20 40	2
E F G H I J	Radial Radial Radial	0 0 0	5 20 40	2
F G H I J	Radial Radial	0	20 40	2
F G H I J	Radial Radial	0	20 40	2
G H I J	Radial	0	40	
H I J				2
I J	Radial	0	70	
J			70	2
J				
	Radial	15	5	1
K	Radial	15	20	1
	Radial	15	40	1
L	Radial	27.5	5	1
М	Tangential	20	20	1
N	Tangential	15	20	1
0	Tangential	15	40	1
()				
Р	Tangential	5	40	1
Q	30° Angle	15	40	1
	0 -			
R	60° Angle	15	40	1



Crack	Crack Type	Breakage Force	Failure
dentifier			Mode
		Ν	
Unci	cacked Biscuit	12.5 ± 1.2	Overload
А	Radial	11.76 ± 1.15	Crack Prop.
В	Radial	10.82 ± 1.39	Crack Prop.
С	Radial	9.23 ± 1.41	Crack Prop.
D	Radial	8.27 ± 1.35	Crack Prop.
			()
Е	Radial	9.76 ± 0.33	Crack Prop.
F	Radial	9.5 ± 2.68	Crack Prop.
G	Radial	9.06 ± 1.13	Crack Prop.
Н	Radial	6.5 ± 0.82	Crack Prop.
Ι	Radial	11.82 ± 1.05	Overload
J	Radial	9.0 ± 1.78	Crack Prop.
K	Radial	8.89 ± 1.33	Crack Prop.
L	Radial	9.92 ± 1.07	Overload
	X		
М	Tangential	9.72 ± 0.7	Overload
Ν	Tangential	10.7 ± 0.54	Overload
0	Tangential	9.36 ± 1.05	Overload
	\odot		
Р	Tangential	8.79 ± 0.93	Crack Prop.
Q	30°	9.36 ± 1.05	Overload
R	60°	8.34 ± 0.75	Overload

Table 3: Failure loads and mechanisms for cracks

Crack Identifier	w/t	2a/t	$\sigma_{\perp max}$	KI
			kPa	kPam ^{0.5}
А	0.139	0.694	412	18.1
В	0.139	2.778	379	18.9
С	0.139	5.556	323	17.1
D	0.139	9.722	290	16.1
Е	0.278	0.694	342	9.8
F	0.278	2.778	333	17.6
G	0.278	5.556	317	20.6
Н	0.278	9.722	228	17.1
			\sim	
J	0.139	2.778	257	12.8
K	0.139	5.556	312	16.5
Р	0.139	5.556	188.5	10.0

Table 4: Cracked biscuit failure analysis

List of Tables

Table 1: Values of the fracture model parameters c_1 and c_2 (extracted from Rice & Levy 1972)

Table 2: Crack geometrical parameters

Table 3: Failure loads and mechanisms for cracks

Table 4: Cracked biscuit failure analysis