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Modulation bandwidth studies of recombination processes in blue and green InGaN quantum well micro-light-emitting diodes

Richard P. Green,1,a) Jonathan J. D. McKendry,2 David Massoubre,2 Erdan Gu,2 Martin D. Dawson,2 and A. E. Kelly3
1Department of Physics, University College Cork, Cork, Ireland
2Institute of Photonics, SUPA, University of Strathclyde, Glasgow G4 0NW, United Kingdom
3School of Engineering, University of Glasgow, Glasgow G12 8QQ, United Kingdom

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We report modulation bandwidth measurements on a number of InGaN-based quantum well LEDs emitting at 450 and 520 nm wavelengths. It is shown that for these devices the data can be interpreted in terms of Auger recombination, taking account of the carrier density dependence of the radiative coefficient. We find values for the Auger coefficient of (1 ± 0.3) × 10⁻²⁰ cm⁶ s⁻¹ at 450 nm and (3 ± 1) × 10⁻³⁰ cm⁶ s⁻¹ at 520 nm. © 2013 American Institute of Physics.

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There is much current interest in the recombination processes governing carrier lifetimes in GaN-based LED structures. One reason for this is that the maximum modulation speeds attainable with these devices are governed by the processes governing carrier lifetimes in GaN-based LED devices.1–3 These have included demonstration of error-free data transmission at rates up to 1 Gbit/s. Fast modulation speeds would also be advantageous for the use of these devices in time-resolved fluorescence measurements.4

The objective of the present work is to investigate whether differential lifetimes in GaN-based micro-LEDs (μLEDs) can be described by the simple ABC model that is successful in describing carrier lifetimes in narrow-gap semiconductors.12 This rate equation model considers that the current through the device is made up of three contributions; i.e., a non-radiative current due to Shockley-Read-Hall (SRH) recombination at defect sites, a current due to radiative recombination of electrons and holes, and an Auger current. The total current is thus

\[ I = I_A + I_B + I_C = eN A N + B N^2 + C N^3, \]

(1)

where \( N \) is the carrier density (per unit volume) within the QWs and \( A, B, \) and \( C \) the SRH, radiative, and Auger coefficients, respectively. The electronic charge is denoted by \( e \) while \( a \) and \( d \) are the device area and the total thickness of the QWs, respectively. The differential lifetime \( \tau \) is then given by the derivative of the recombination rate with respect to carrier density12

\[ \tau^{-1} = A + 2BN + 3CN^2. \]

(2)

This simple model is made under the assumption that \( A, B, \) and \( C \) are independent of \( N \), and the carrier density is the same in all QWs. In fact, when working with narrow-gap materials such as InGaAs, good agreement with experiment can only be found over a large range of carrier densities if \( B \) is a decreasing function of \( N \). This is attributed to a reduction in the optical matrix element for the interband transitions as carriers begin to fill states away from the \( \Gamma \) point.13 This can be described by the empirical formula14

\[ B(N) = \frac{B_0}{1 + N/N_0} \]

(3)
or by a linear approximation to this.

We carried out measurements of the differential lifetime in InGaN based μLEDs as a function of bias current, in order to test the applicability of the above model. The μLEDs were fabricated from standard commercial MQW wafers, emitting at 450 nm or 520 nm into 24–84 μm diameter mesas. The combined thickness of the quantum wells within each wafer was 14 nm. A plan view and schematic cross-section of the device structure are shown in Figs. 1(a) and 1(b), respectively. Full details of device design and fabrication techniques have been given elsewhere.1,2

Each LED was contacted using a high speed (40 GHz bandwidth) ground-signal-ground probe, and the frequency response was measured under a range of biases using a network analyser and a high-speed silicon photodetector (Newport 818-BB21A). The differential carrier lifetime is related to the –3 dB point of the frequency response15 as

\[ \tau^{-1} = 2\pi f_{-3dB}. \]

(4)

By using μLEDs, we were able to ensure that the device bandwidth was determined by the carrier lifetime, rather

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1Electronic mail: r.green@ucc.ie.
than the device capacitance. We confirmed this by comparing results obtained from devices of different diameter; the value of $\tau$ at a given current density was very similar for all devices measured (see Fig. 5 in Ref. 2). The material properties obtained in this way would also be applicable to larger devices made from the same wafer.

From this data, it is possible to obtain the QW carrier densities corresponding to a given current $I$ through the integral

$$N(I) = \frac{\eta_{\text{inj}}}{e\alpha I} \int_0^I \tau dI.$$  

(5)

$\eta_{\text{inj}}$ represents the injection efficiency of the carriers into the QWs. For the present we assume this to be independent of the injection current; this will be discussed later. A value of 0.7 is used as reported by Scheibenzuber et al. based on slope efficiency measurements of GaN-based laser diodes. The measured carrier lifetimes are plotted against the carrier densities obtained in this way in Fig. 2. We find that the values of $B$ and $C$ obtained exhibit relatively small changes when this parameter is varied. A change from $\eta_{\text{inj}} = 0.7$ to 0.6 results in a 30% increase in the value obtained for $B$, much smaller than that due to the uncertainty in the value of injection efficiency, as discussed below. The same change in $\eta_{\text{inj}}$ results in a $\sim$60% increase in the value of $C$ obtained.

The significance of this will be discussed below.

To give a more robust fitting of the lifetime data to the $ABC$ model, we used an independent means of establishing values for $B$, based on the relationship between spontaneous emission power and current (the LI curve). The output power was measured using a calibrated 12 mm diameter Si photodetector placed in close proximity above the device. This LI measurement gives a reliable determination of the shape of the $B(N)$ versus $N$ curve. This will also give accurate values for $B$, if it is combined with a knowledge of the proportion of photons emitted by the MQWs which are incident on the photodiode (the collection efficiency, $\eta_{\text{coll}}$). It is difficult to estimate the value of $\eta_{\text{coll}}$, due to the substrate emitting geometry of our devices, and the fact that scattering of light from the mesa sidewalls will play a much more significant role in $\mu$LEDs than in larger devices. Nevertheless, there are certain clear statements that we can make about $\eta_{\text{coll}}$ for the devices used in this work. First, the 450 nm devices were grown on a patterned sapphire substrate designed to enhance the proportion of light escaping. Huh et al. reported that this resulted in an increase in emitted light of $\sim33\%$ (for top surface collection) and $\sim19\%$ (for substrate side collection). The $\mu$LEDs reported here are in flip-chip format with a reflective p-contact, so we consider that $\eta_{\text{coll}}$ for the 450 nm devices is likely to be about 50% higher than would be the case for an equivalent structure grown on a plain substrate and will be higher than that for the 520 nm devices.

We can estimate a lower bound for $\eta_{\text{coll}}$ using a simple model of the Fresnel reflectivity at the GaN-sapphire and sapphire-air interfaces. Because this does not include the effects of either multiple reflections or scattering from mesa sidewalls, this will provide an underestimate for $\eta_{\text{coll}}$. This model suggests that a suitable lower bound will be $\eta_{\text{coll}}(\text{min}) = 4\%$ (520 nm) and 6.6% (450 nm, including the substrate effect). The work by Lee et al. provides a corresponding upper bound. They analysed the light emission from a variety of LED geometries using a Monte-Carlo ray tracing technique. Based on this, we expect that $\eta_{\text{coll}}(\text{max}) = 30\%$ (450 nm) and 20% (520 nm). The data presented in this paper uses midpoint values of $\eta_{\text{coll}} = 18\%$ (450 nm) and 12% (520 nm). While this uncertainty will have a strong effect on the values we obtain for $B$, its effect on $C$ is small, as will be discussed later.

The carrier density dependence of $B$ obtained from these measurements is shown in Fig. 3. A number of significant features are visible in this. The values of $B$ found for the 520 nm emitting devices are generally an order of magnitude below those of the 450 nm devices. The lower $B$ coefficient found for our 520 nm emitting device is consistent with literature reports showing that the internal quantum efficiency of
green-emitting InGaN LEDs is lower than their blue-emitting counterparts. This phenomenon is commonly referred to as the “green gap” or “green-yellow gap” and may be attributed to stronger piezoelectric and spontaneous polarisation effects in higher indium content InGaN devices. Laubsch et al. reported values of $B$ in the region of $1.2 \times 10^{-12}\text{cm}^3\text{s}^{-1}$ for a 520 nm emitting device, compatible with those shown in Fig. 3. In contrast, the values of $B$ reported for blue-emitting devices are larger; Meneghini et al. report $B \sim 1 - 2 \times 10^{-11}\text{cm}^3\text{s}^{-1}$ for 450 nm emitting LEDs, which are close to the values we obtain at high currents. The decrease in the observed value of $B$ with carrier density is generally attributed to the effects of phase space filling. We observe a reduction in $B$ of about 80% over the current range measured. This large change is attributed to the high current densities attainable in the µLED geometry used in this work.

The task of finding values for $A$ and $C$ is now simplified by having an independent means of studying the carrier density dependence of $B$. From inspection of Eq. (2), it is apparent that $A$ can be found by extrapolating the data in Fig. 2 back to the $N=0$ axis. The values of $A$ obtained were in the range $1.2 \times 10^8\text{s}^{-1} - 3.2 \times 10^8\text{s}^{-1}$ for the 450 nm devices, and $0.8 \times 10^8\text{s}^{-1} - 1.8 \times 10^8\text{s}^{-1}$ for the 520 nm ones. These are within the range of those which have previously been published. One possible reason for the consistently differing values of $A$ found between devices emitting at the two different wavelengths could be the use of a patterned substrate; this has previously been reported to affect the dislocation densities. For both wavelengths, a clear dependence of $A$ on the device diameter was observed, as seen in the inset to Fig. 2. This is attributed to the greater influence of surface effects on the smaller devices, with a high density of recombination sites at the edge of the mesa arising from ICP etching during the fabrication process. A similar dependence of $A$ on the device diameter has been reported for GaAs/AlGaAs VCSEL devices of similar size.

The above discussion gives a method of estimating the portion of the total current through the device due to recombination at defect sites and that due to radiative recombination. We assume here that all the remainder of the current is due to Auger recombination, and we denote this $I_C$, as defined in Eq. (1). This implies that the value of $C$ will be given by

$$C = \frac{1}{\varepsilon_0 d \partial N^3}.$$  

Fig. 4(a) shows the values of $I_C$ obtained for two representative devices, plotted against $N^3$. Except at very low carrier densities, the data can be fitted to a straight line, indicating that the Auger coefficient is not significantly changing over this range. This fitting process was carried out for a number of devices of different diameters and the two wavelengths being considered. The values of $C$ obtained from this process were $C_{450} = (1 \pm 0.3) \times 10^{-29}\text{cm}^6\text{s}^{-1}$ and $C_{520} = (3 \pm 1) \times 10^{-30}\text{cm}^6\text{s}^{-1}$. There was no trend observed in the value found for $C$ for devices of different diameters. These values are shown in Fig. 4(b), together with a number of experimental and theoretical values for $C$ which have previously been reported in the literature.

We now consider the influence that the uncertainty in our estimate of the collection efficiency has on the results we have obtained. Since the values of $A$ are obtained directly from the differential lifetime measurements, this will be unaffected by any such error. It is similarly obvious that the values of $B$ (at a given value of $N$) will be inversely proportional to $\eta_{\text{coll}}$. Accordingly, the values at high carrier density could be within the range of $B_{450} = 0.33 \times 10^{-11}\text{cm}^3\text{s}^{-1}$ and $B_{520} = 0.4 \times 10^{-12}\text{cm}^3\text{s}^{-1}$, as shown in Fig. 4(b); the effect of this will depend on the relative importance of the three recombination processes considered here. To understand this, we have analysed the data presented here, but with the upper and lower bounds for the collection efficiency, as discussed above. We find that the change in the value of $C$ is of the order of 15% over the full range of reasonable estimates for collection efficiency; lower than the standard deviation of the values obtained from the individual
pixels (about 30%). This low influence in the value of $B$ is to be expected if the current through the device is dominated by non-radiative recombination processes. There is also a $\sim$60% uncertainty in the value of $C$ arising from the value chosen above for $\eta_{\text{irr}}$. This can be seen to be relatively small, when compared to the large spread of reported values for $C$ shown in Fig. 4.

In conclusion, we have measured the modulation bandwidth of a number of GaN-based micro-LEDs, emitting at 450 and 520 nm wavelength. These results were analysed in a rate equation model, making proper allowance for the carrier density dependence of the radiative coefficient, $B$, which was determined from LI measurements. By doing this, the carrier lifetimes can be understood by considering Shockley-Read-Hall and Auger recombination as the only significant non-radiative recombination mechanisms and do not require any carrier overflow mechanisms or variation of the injection efficiency to explain them. Values were found for the Auger coefficient of $C_{450}=(1\pm0.3)\times10^{-29}\text{cm}^6\text{s}^{-1}$ and $C_{520}=(3\pm1)\times10^{-30}\text{cm}^6\text{s}^{-1}$.

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