

	,
Title	Reaction mechanism of the metal precursor pulse in plasma- enhanced atomic layer deposition of cobalt and the role of surface facets
Authors	Liu, Ji;Lu, Hongliang;Zhang, David Wei;Nolan, Michael
Publication date	2020-05-15
Original Citation	Liu, J., Lu, H., Zhang, D. W. and Nolan, M. (2020) 'Reaction Mechanism of the Metal Precursor Pulse in Plasma-Enhanced Atomic Layer Deposition of Cobalt and the Role of Surface Facets', The Journal of Physical Chemistry C, 124(22), pp. 11990-12000. doi: 10.1021/acs.jpcc.0c02976
Type of publication	Article (peer-reviewed)
Link to publisher's version	https://pubs.acs.org/doi/10.1021/acs.jpcc.0c02976 - 10.1021/ acs.jpcc.0c02976
Rights	© 2020 American Chemical Society. This document is the Accepted Manuscript version of a Published Work that appeared in final form in Journal of Physical Chemistry C copyright © American Chemical Society after peer review and technical editing by the publisher. To access the final edited and published work see https://pubs.acs.org/doi/10.1021/acs.jpcc.0c02976
Download date	2025-07-31 01:32:53
Item downloaded from	https://hdl.handle.net/10468/10602



Reaction Mechanism of the Metal Precursor Pulse in Plasma-

Enhanced Atomic Layer Deposition of Cobalt and the Role of

Surface Facet

Ji Liu<sup>a</sup>, Hongliang Lu<sup>b</sup>, David Wei Zhang<sup>b</sup>, and Michael Nolan<sup>a,\*</sup>

<sup>a</sup> Tyndall National Institute, University College Cork, Lee Maltings, Dyke Parade, Cork, T12 R5CP, Ireland

<sup>b</sup> State Key Laboratory of ASIC and System, Shanghai Institute of Intelligent Electronics & Systems, School of Microelectronics, Fudan University, Shanghai 200433, China

Corresponding author:

\*E-mail: Michael.nolan@tyndall.ie. Tel: +353 021 2346983

#### Abstract

Cobalt is a potential candidate in replacing copper for interconnects and has been applied in the trenches and vias in semiconductor industry. A non-oxidizing reactant is required in plasmaenhanced atomic layer deposition (PE-ALD) of thin films of metals to avoid O-contamination. PE-ALD of Co has been demonstrated experimentally, but the growth mechanism and key reactions are not clear. In this paper, the reaction mechanism of metal cyclopentadienyl (Cp, C<sub>5</sub>H<sub>5</sub>) precursors (CoCp<sub>2</sub>) and NH<sub>x</sub>-terminated Co surface is studied by density functional theory (DFT) calculations. The Cp ligands are eliminated by CpH formation via a hydrogen transfer step and desorb from metal surface. The surface facet plays an important role in the reaction energies and activation barriers. The results show that on the NH<sub>x</sub>-terminated surfaces corresponding to ALD operating condition (temperature range 550K to 650K), the two Cp ligands are eliminated completely on Co(100) surface during the metal precursor pulse, resulting in Co atom deposited on the Co(100) surface. But the second Cp ligand reaction of hydrogen transfer is thermodynamically unfavourable on the Co(001) surface, resulting in CoCp fragment termination on Co(001) surface. The final terminations after metal precursor pulse are 3.03 CoCp/nm<sup>2</sup> on NH<sub>x</sub>terminated Co(001) surface and 3.33 Co/nm<sup>2</sup> on NH<sub>x</sub>-terminated Co(100) surface. These final structures after metal precursor pulse are essential to model the reaction during the following Nplasma step.

#### 1. Introduction

Copper (Cu) has been widely used in the semiconductor industry as interconnect for 20 years.<sup>1</sup> However, continuous deposition of Cu films needed for interconnect in nanoelectronics is difficult and a barrier layer is required to prevent the diffusion of Cu into the dielectric layer and Si substrates. Cu also aggregates into 3D structures.<sup>2-3</sup> Finding suitable barrier/liner layer is still a challenge because issues with copper reduce the electrical resistivity of the interconnect, especially for devices at nanosize dimension. One solution is to replace Cu with metals that do not suffer these issues. Transition metal Cobalt (Co) is important as candidates in replacing Cu for interconnects and has been applied in the trenches and vias with the downsizing of semiconductor devices. 4-6 There is also the question of deposition of nanoscale films, in particular on high aspect ratio structures, where different surface facets maybe present. Atomic layer deposition (ALD) is applied for conformal deposition and growth control at the atomic level, which is needed for deposition onto high aspect ratio structures.<sup>7-8</sup> Generally, ALD consists of two self-limiting half cycles, where the reactions will stop after all available surface sites are consumed. In addition to the successful application of ALD in microelectronics and the semiconductor industry, it is further applied in the areas of catalysis and energy conversion and storage.<sup>5, 9-10</sup>

For the ALD of Co, Cp based precursors such as  $CoCp_2$ ,  $CoCp(CO)_2$  and Co(CpAMD) have been developed and applied. <sup>11-12</sup> For thermal ALD of Co with metal precursor and hydrogen reactant, the required growth temperature can be as high as 350°C, but the growth rate is as low as 0.12  $\Box$ /cycle. <sup>13</sup> With the application of plasma-enhanced ALD (PE-ALD), the temperature can be reduced to 75°C. The reported growth cycle with  $CoCp_2$  is increased to 1.5  $\Box$ /cycle. <sup>14-15</sup> For the N-plasma source, a mixture of  $N_2$  and  $H_2$  is used and the properties of the deposited Co thin film greatly depend on the  $N_2/H_2$  gas flow ratio. <sup>16-17</sup> It is noted that  $H_2$  plasma alone or individual  $N_2$ 

and H<sub>2</sub> plasma results in high resistivity and low purity Co thin films. Previous studies argue that the presence of NH<sub>x</sub> species is needed to deposit low resistivity and high purity Co thin film.<sup>18</sup> NH<sub>x</sub> species are needed for chemisorption of metal precursor and removal of the Cp ligand. But they are not incorporated in the film, because most of the N may desorb in the form of either NH<sub>3</sub> or N<sub>2</sub>. The detailed mechanism requires deeper study, but one possible reason is that highly reactive radicals from N-plasma source should be present for successful PE-ALD of Co. The final structure and termination after the metal precursor pulse are vital and essential to model the plasma step, which is the key advance of current paper.

Density functional theory (DFT) calculations have been successfully applied to study the ALD of metals and metal oxides. <sup>19-22</sup> Theoretically, the design and large-scale screening of precursors based on specific criteria such as the thermodynamic stability and kinetic stability have been performed in ALD modelling. <sup>23-24</sup> DFT calculations can also be applied to reveal the reaction mechanisms and the derived growth rate serves as guideline to experiments. In the early stage, the deposition of metal oxide, such as Al<sub>2</sub>O<sub>3</sub> from trimethylaluminum and O<sub>3</sub>/H<sub>2</sub>O as the co-reactant, has been studied theoretically. <sup>25-29</sup> The surface hydroxyl groups are formed and their surface coverages affect the growth rate. For the deposition of metals, Elliott has proposed a mechanism for deposition of noble metals including Pd, Ir and Pt using homoleptic precursors and oxygen from DFT calculations. <sup>30</sup> It is found that each ligand is replaced by a hydroxyl group, which can be further eliminated by Brønsted-type reaction.

The reaction mechanism using oxidizing reactant such as O<sub>3</sub> and H<sub>2</sub>O is well-established. However, when depositing metals, O-source can promote oxidation of the metal surface and therefore cause contamination. Non-oxidizing reactants such as NH<sub>3</sub> in PE-ALD of transition metals have been experimentally developed. A complete PE-ALD process using N-plasma (NH<sub>3</sub>

or mixture of  $N_2$  and  $H_2$ )<sup>references</sup> is as follows. Firstly, it is vital to note that at the post-plasma stage, the metal surface is actually the  $NH_x$ -terminated metal surface. In the first half-cycle, the metal precursor  $CoCp_2$  reacts with  $NH_x$ -terminated metal surface. The Cp ligand is eliminated by hydrogen transfer from the surface to form CpH, which desorbs from surface. In the second half-cycle, the plasma generated radicals such as  $N_xH_y$  will react with the precursor fragment terminated metal surface and the Co atoms are deposited on the surface, which is covered by  $NH_x$  groups at the end of second half cycle. In our recent published work, the nature and stability of  $NH_x$ -terminated metal surfaces were studied. The results show that at ALD operating condition (temperature range 550K to 650K), on the low energy (001) surface, NH-termination is the most stable surface termination, while on the high energy (100) surface, a mixture of NH and  $NH_2$  is the most stable surface termination.

In this paper, we explore the reaction mechanism for the metal precursor pulse by DFT calculations. We do not consider the initial nucleation and growth on typical substrates such as XXXX, but instead focus on the metal precursor pulse at the metal substrate, which is a common approach in modelling metal ALD, e.g. refs. The Journal of chemical physics 2017, 146 (5), 052822, Chemistry of Materials 2016, 28 (17), 6282-6295. The elimination of Cp ligand is the key step during the metal precursor pulse. Single metal precursor CoCp<sub>2</sub> is adsorbed on the NH<sub>x</sub>-terminated surfaces<sup>31</sup> and the hydrogen transfer step is studied in detail with calculation of proton migration barrier. After the first CpH formation and desorption, the possibility of the loss of second Cp ligand is also investigated. With the NH<sub>x</sub> terminations at ALD operating condition, on the Co(100) surface, the metal precursor can undergo two hydrogen transfer steps and the two Cp ligands are eliminated completely, resulting in Co atom deposition on the surface, binding to N atom. However, only one Cp ligand is eliminated on the Co(001) surface, resulting in CoCp

fragments on the surface after the metal precursor pulse. The precursor coverage effect is studied with two metal precursors CoCp<sub>2</sub> adsorbed on the surface. A neighbouring CoCp has hindered the reactivity by increasing the activation barrier of the first hydrogen transfer on Co(001) surface, but promoted the reactivity by lowering the activation barrier of the first hydrogen transfer on Co(100) surface. The determined surface coverage of final terminations after the metal precursor pulse are 3.03 CoCp/nm<sup>2</sup> on NH<sub>x</sub>-terminated Co(001) surface and 3.33 Co/nm<sup>2</sup> on NH<sub>x</sub>-terminated Co(100) surface.

## 2. Methods and Computational Details

All the calculations are performed on the basis of periodic spin-polarized density functional theory (DFT) within a plane wave basis set and projector augmented wave (PAW) formalism<sup>32</sup>, as implemented in the Vienna *ab initio* simulation package (VASP 5.3) code. The generalized gradient approximation (GGA) with the parameterization of Perdew-Burke-Ernzerhof (PBE) is used for the exchange-correlation functional.<sup>33-34</sup> We use 9 valence electrons for Co, 5 for N, 4 for C, and 1 for H. The plane wave energy cutoff is set to be 400eV. The convergence of energy and forces are set to be  $1\times10^{-4}$ eV and 0.01eV/ $\Box$ , respectively. The bulk Co crystal structure is optimized by simultaneously relaxing the ionic positions, cell volume and cell shape at a higher plane wave energy cutoff of 550eV and using a Monkhorst-Pack grid k-point mesh<sup>35</sup> of  $12\times12\times6$ . The resulting lattice constants are a=b=2.49Å, and c=4.03Å for Co bulk.

The deposited Co films by PE-ALD are polycrystalline and have random surface orientations after low temperature deposition. Based on our previous study<sup>31</sup> on the stability of NH/NH<sub>2</sub>

terminations, we have chosen the most stable (001) surface and a less stable but high reactivity surface, namely (100), to investigate the precursor reaction mechanism. While defects and imperfections in the metal surface can be present, producing low-coordinated sites or sites such as edge Co atoms, we limit ourselves to these two crystalline surfaces and the role of surface defects and low-coordinated sites is a topic for future study. A (4×4) supercell is used to model the (001) surface with a surface lattice of  $a = b = 9.96 \square$  (surface area = 0.99 nm²), while a (3×3) supercell, with a surface lattice of  $a = 7.47 \square$ ,  $b = 12.10 \square$  (surface area = 0.90 nm²), is used to model the (100) surface. For the Co(001) surface, a five-layer slab is used, with the bottom three layers fixed during the calculation; while for the Co(100) surface, due to zigzag structure, a four-bilayer (eight atomic layer) slab is built with the bottom two bilayers (bottom four layers) fixed during the calculations. From our previous studies, fixing these-layers is sufficient to model these Co surfaces. A k-point mesh of  $2 \times 2 \times 1$  is used in (4×4) supercell and for the (3×3) supercell a  $3 \times 2 \times 1$  mesh is used.

Our previous DFT study<sup>31</sup> of NH<sub>x</sub> saturation coverage shows that at zero-K condition, the Co(001) surface is terminated with 0.67ML NH and 0.23ML NH<sub>2</sub>, which contains 10 NH and 4 NH<sub>2</sub> in a  $(4\times4)$  supercell. On the Co(100) surface, the termination is 1ML NH and 1ML NH<sub>2</sub> due to the trench structure, which contains 9 NH and 9 NH<sub>2</sub> in  $(3\times3)$  supercell. The saturation coverages are summarized in Table 1. The configurations of the NH<sub>x</sub>-terminated Co surfaces at zero-K condition are shown in Figure 1(a)-(b).

Table 1. The calculated saturation coverages on Co (001) and (100) surfaces at zero-K (maximum coverages) and ALD conditions (low coverages).

	Co(001)	Co(100)
	(4×4)	(3×3)
Zero-K condition	$0.67ML\ NH + 0.23ML\ NH_2$	$1ML NH + 1ML NH_2$
ALD condition	0.56ML NH	$0.67ML\ NH + 0.67ML\ NH_2$

At ALD operating condition (temperature range 550K to 650K), some of the surface NH<sub>x</sub> species desorb from the surface. The preferred NH<sub>x</sub> coverages are<sup>31</sup>: Co(001) surface is terminated with 0.56ML NH, which is 9 NH in (4×4) supercell. On the Co(100) surface, the preferred surface termination is 0.67ML NH and 0.67ML NH<sub>2</sub>, which contains 6 NH and 6 NH<sub>2</sub> in (3×3) supercell. The configurations of NH<sub>x</sub> terminations Co surfaces at the ALD operating condition are shown in Figure 1(c)-(d). On the (100) surface, due to the trench structure, NH prefers the subsurface channel bridge site and NH<sub>2</sub> prefers the surface bridge site. The channel bridge is a bridge site with two channel metal atoms and the surface bridge is the bridge site with two surface metal atoms. The configuration of single NH and NH<sub>2</sub> adsorbed on channel bridge and surface bridge sites is shown in Figure S1 in the supporting information.

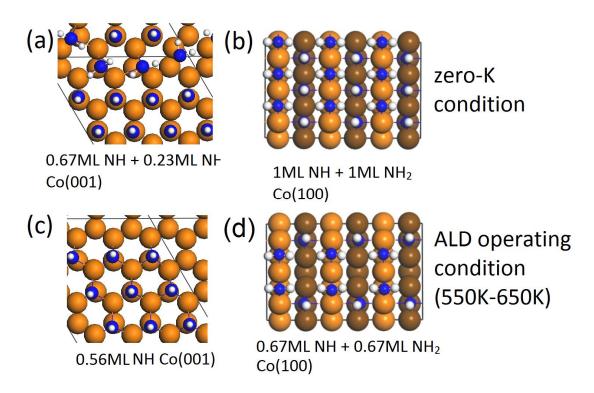


Figure 1. The top view of  $NH_x$ -terminated metal surfaces at zero-K condition including (a) Co(001), and (b) Co(100) and at ALD operating condition including (c) Co(001), and (d) Co(100). Co atoms are represented by orange colour for surface terminating atoms and soil colour for the channel bridge subsurface atoms in the (100) surface; N atom and H atom are represented by dark blue and white atom, respectively.

The molecular geometries of the metal precursor CoCp<sub>2</sub> are relaxed in the same supercell as Co(001), with an energy cutoff of 400eV and Gamma point sampling. The van der Waals correction is applied with the PBE-D3 method to ensure an accurate description of the metal precursor adsorption energy.<sup>36</sup> The activation barriers reported in this paper are computed using climbing image nudged elastic band (CI-NEB) method<sup>37</sup> with 6 images including the starting and ending geometries and with the forces converged to 0.05eV/Å.

## 3. Results and Discussions

### 3.1 Metal precursor adsorption on $NH_x$ -terminated Co (001) and (100) surfaces

The structure of the gas phase metal precursor is first described. For CoCp<sub>2</sub>, the Cp-Cp distance is between 3.39Å to 3.40Å. The Co-C distance is between 2.08Å to 2.10Å, indicating little tilting of the two Cp rings. When adsorbed on the NH<sub>x</sub>-terminated metal surfaces, the metal precursor can be placed perpendicular to substrate with one Cp ring interacting with the surface (which we term the **upright** adsorption mode) or parallel to surface with both Cp rings interacting with the surface (which we term the **horizontal** adsorption mode). The adsorption energy is calculated from:

$$E_{ad} = E_{tot} - E_{\frac{NHx}{Metal}} - E_A \tag{1}$$

where E<sub>tot</sub>, E<sub>NHx/Metal</sub>, and E<sub>A</sub> are the energy of the NH<sub>x</sub>-terminated metal slab with adsorbed precursor CoCp<sub>2</sub>, the slab model for the NH<sub>x</sub>-terminated metal surface, and the isolated CoCp<sub>2</sub> precursor, respectively. All the energies are computed with the van der Waals correction included. A negative adsorption energy corresponds to exothermic adsorption. We will use the NH<sub>x</sub> coverages obtained at zero K and ALD conditions to explore the effect of NH<sub>x</sub> coverage on the CoCp<sub>2</sub> pulse.

3.1.1 Metal precursor adsorption on NHx-terminated Co (001) and (100) surfaces at high coverage

The calculated adsorption energies of the metal precursors on  $NH_x$ -terminated Co (001) and (100) surfaces at maximum  $NH_x$  coverage (Table 1) are shown in Table 2. Co(001) is terminated with 0.67NL NH and 0.23ML  $NH_2$  and Co(100) has mixed termination with 1ML NH and 1ML  $NH_2$ . On the (001) surface, the metal precursor prefers to bind to the substrate in the upright adsorption

mode through one Cp ring. On the (100) surface, the metal precursor prefers to bind to the substrate through both Cp rings with the precursor in the horizontal configuration. These structures are shown in Figure 2(a)-(b). The configurations of the corresponding less stable adsorption structures are shown in Figure 2(c)-(d).

Table 2. The calculated adsorption energy of metal precursor  $CoCp_2$  adsorbed on  $NH_x$ -terminated Co (001) and (100) surfaces. The  $NH/NH_2$  terminations correspond to the maximum  $NH_x$  coverage.

	Co(001)	Co(100)
upright	-0.10	-1.41
horizontal	3.16	-1.73

This difference in preferred binding mode at the two surface facets is due to the different surface structures. The (001) surface has a flat surface structure, while (100) surface has a unique zigzag structure. On the (001) surface, an upright position with one Cp ring close to metal surface results in stronger adsorption strength as this will maximise the number of interactions between the surface and carbon atom of the Cp ring. By contrast, the horizontal CoCp<sub>2</sub> structure on (001) does not allow carbon-surface interactions and therefore the interaction is less favourable. In addition, in the upright binding mode, each carbon atom in the Cp ring is available for the hydrogen transfer step to form CpH. On Co(001), the distances between the two Cp rings in the upright structure elongate to 3.43Å, while the metal-C distances elongate to 2.11Å for the Cp ring interacting with the surface, which is a small elongation compared to the free precursor.

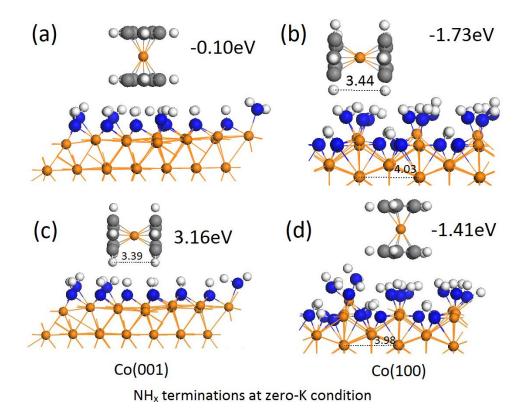


Figure 2. The configurations of the most stable adsorption of precursor  $CoCp_2$  on (a) Co(001) surface, and (b) Co(100) surface and the less stable adsorption of precursor  $CoCp_2$  on (c) Co(001) surface, and (d) Co(100) surface. The  $NH_x$  termination is with respect to zero-K condition. Co atoms are represented by orange spheres; Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

On the Co (100) surface, the distance across the trench (between two neighbouring surface metal atoms) is 4.03Å. The distances between the two Cp rings in horizontal configuration are in the range of 3.34Å to 3.44Å and the metal-C distances are 2.07Å to 2.10Å. The metal precursor can therefore be well-accommodated within the trench structure of the Co(100) surface, which promotes the adsorption of the precursor, as indicated by the computed adsorption energies. Compared to free metal Cp precursors, the two Cp rings are tilted with shorter ring-ring distance

for the atoms away from the surface and longer ring-ring distance for the atoms closer to the surface.

3.1.2 Metal precursor adsorption on NHx-terminated Co (001) and (100) surfaces at low NH<sub>x</sub> coverage, corresponding to the ALD operating condition

The calculated adsorption energies of the metal precursors on the  $NH_x$ -terminated Co~(001) and (100) surfaces corresponding to ALD operating condition are shown in Table 3. The initial  $NH_x$  terminations are now 0.56ML NH on Co(001) and the mixed termination with 0.67ML NH and 0.67ML  $NH_2$  on Co(100).

Table 3. The calculated adsorption energy of the metal precursor  $CoCp_2$  adsorbed on  $NH_x$ -terminated Co (001) and (100) surfaces in the two adsorption configurations. The  $NH/NH_2$  terminations correspond to the ALD operating condition (temperature range 550K - 650K).

	Co(001)	Co(100)
upright	-0.68	-0.34
horizontal	-0.56	-1.67

The binding preference of  $CoCp_2$  is the same as the higher  $NH_x$  coverage surfaces and the relaxed adsorption structures are shown in Figure 3(a)-(b). The configurations of less stable adsorption structures are shown in Figure 3(c)-(d). On Co(001) surface, an upright position with one Cp ring close to metal surface can result in stronger adsorption strength, although the horizontal adsorption mode is now much more favourable compared to the same adsorption mode at higher  $NH_x$  coverage. The distances between the two Cp rings are in the range of 3.37Å to 3.40Å on Co(001) surface. Compared with free  $CoCp_2$ , the two Cp rings are slightly tilted. The distances for metal-C are between 2.08Å and 2.10Å.

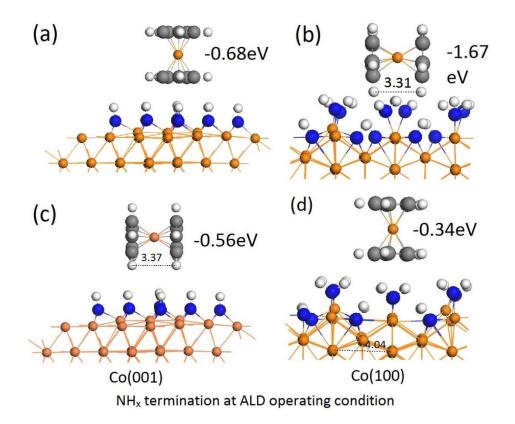


Figure 3. The configurations of the most stable adsorption of precursor  $CoCp_2$  on (a) Co(001) surface, and (b) Co(100) surface and the less stable adsorption pf precursor  $CoCp_2$  on (c) Co(001) surface, and (d) Co(100) surface. The NH<sub>x</sub> termination is with respect to ALD operating condition. Co atoms are represented by orange spheres; Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

The distance across the trench between two neighbouring metal atoms is 4.03Å on Co(100) surface. The distances between the two Cp rings are in the range of 3.31Å to 3.40Å. The distances for metal-C are 2.06Å to 2.10Å. The horizontal adsorption mode of metal precursor can be well-accommodated within the trench of the (100) surfaces, which can result in stronger adsorption strength. Compared to free metal Cp precursors, the two Cp rings are tilted with shorter ring-ring distances for the atoms closer to the surface and longer ring-ring distances for the atoms away from the surface.

3.2 Single precursor reaction pathway on Co (001) and (100) surfaces with  $NH_x$  terminations at ALD operating condition

The reaction pathway during the metal precursor pulse is studied with respect to the NH<sub>x</sub> terminations at ALD operating condition. In this section, we address the reaction mechanism when a single CoCp<sub>2</sub> precursor is adsorbed on NH<sub>x</sub>-terminated Co (001) and (100) surfaces. Once the metal precursor is adsorbed on NH<sub>x</sub>-terminated metal surfaces, the Cp ligand can undergo hydrogen transfer, CpH formation, CpH desorption, second hydrogen transfer, and second CpH formation and desorption.

Upon adsorption, no spontaneous hydrogen transfer was observed on any NH<sub>x</sub>-terminated Co (001) and (100) surfaces. This means that the hydrogen transfer step must overcome an activation barrier. The possible reactions of a single adsorbed molecule of CoCp<sub>2</sub> on NH<sub>x</sub> terminated metal surfaces can be illustrated as follows:

$$A: H^* + CoCp_2 \rightarrow CoCp^* + HCp$$

$$B: H^* + CoCp \rightarrow Co^* + HCp$$

where reaction A involves the first Cp ligand and reaction B involves the second Cp ligand. We have calculated the energy along the reaction pathway and the activation barriers for hydrogen transfer at each step. Note that the reaction energies of precursor adsorption ( $E_{adsorption}$ ) and the first hydrogen transfer ( $E_{hydrogen}^{I}$ ) are with reference to the NH<sub>x</sub>-terminated metal surface and free CoCp<sub>2</sub>. If one CpH molecule desorbs from the surface, the reaction energies of first CpH desorption ( $E^{Des}_{CpH}^{I}$ ), second hydrogen transfer ( $E_{hydrogen}^{II}$ ), and second CpH desorption ( $E^{Des}_{CpH}^{II}$ ) are with reference to NH<sub>x</sub>-terminated metal surface, free CoCp<sub>2</sub>, and free CpH.

The results for NH<sub>x</sub> terminations at ALD operating condition are summarized in Figure 4 and the calculated barriers for the hydrogen transfer steps are presented in Table 4. In order to assess any role of NH/NH<sub>2</sub> coverage, the results for NH<sub>x</sub> terminations at highest coverage, i.e. zero K, are summarized in Figure S2 and the calculated barrier for the hydrogen transfer steps are presented in Table S1 in supporting information.

Table 4. The calculated reaction energy for hydrogen transfer step and reaction barriers,  $E_{barrier}$ , on Co (001) and (100) surfaces with  $NH_x$  terminations corresponding to ALD operating condition.  $E_{adsorption}$  is the energy change upon precursor adsorption,  $E_{hydrogen}$  I

and  $E_{hydrogen}{}^{II}$  are the energy change for the first and second hydrogen transfer, and  $E^{Des}{}_{CpH}{}^{I}$  is the desorption energy of CpH.

	$H^* + CoCp_2 \to CoCp^* + HCp$			$H^* + Coo$	$Cp \rightarrow Co^* + D$	НСр
	$E_{adsorption} \\$	$E_{\text{hydrogen}}{}^{I}$	Ebarrier	$E^{\mathrm{Des}}_{\mathrm{CpH}}{}^{\mathrm{I}}$	$E_{\text{hydrogen}}^{\text{II}}$	Ebarrier
Co(001)	-0.68	-0.63	0.56	-1.69	-0.72	1.57
Co(100)	-1.67	-2.19	0.52	-1.32	-1.15	0.85

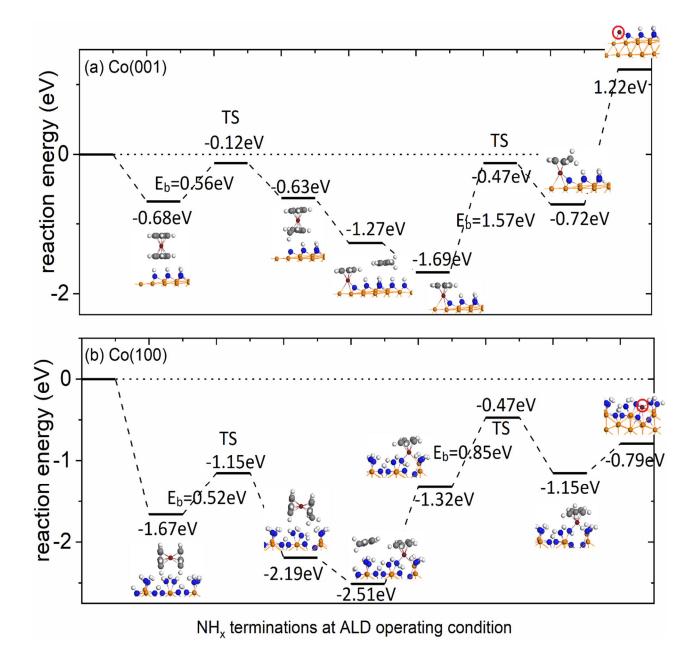


Figure 4. The plotted metal precursor reaction pathway on (a) Co(001) surface and (b) Co(100) surface with  $NH_x$  terminations at ALD operating condition. The Cp ligand is eliminated via hydrogen transfer. The substrate Co atoms are represented by orange spheres and the Co atom from  $CoCp_2$  is represented by a red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

On the Co(001) surface, the metal precursor  $CoCp_2$  has a moderate adsorption strength ( $E_{adsorption} = -0.68eV$ ). The first Cp ligand elimination via hydrogen transfer is thermoneutral with a moderate barrier of 0.56eV for transfer of hydrogen and a small energy change of 0.05eV. The formation and desorption of this CpH are exothermic and the elimination of this CpH makes the reaction exothermic with a total energy gain of -1.69eV.

Considering now the elimination of the second Cp ligand via hydrogen transfer is highly endothermic with a high activation barrier for hydrogen transfer, which is 1.57eV. The desorption of the second CpH to give a bare Co species that is bound to surface N atom is thermodynamically unfavourable. Thus, the final termination is a CoCp termination on Co(001) as a result of the high barrier and endothermic reaction for the elimination of the second Cp ligand.

At the highest  $NH_x$  coverage, the terminations on Co(001) are 0.67ML NH and 0.23ML  $NH_2$ . As shown on Figure S2(a) in supporting information, the overall reactions for Cp ligand eliminations via hydrogen transfer step are exothermic. The calculated barriers are 1.00eV and 1.24eV for first and second hydrogen transfer steps. However, the initial adsorption of metal precursor  $CoCp_2$  is quite weak, which has the value of -0.10eV, so that this reaction is not likely to proceed.

On the Co(100) surface, we find that for the hydrogen transfer step, the channel H atom is more reactive than the surface H atom. Table 5 compares the energies for transfer of these two hydrogen species. On the Co(100) surface, after relaxation, a channel hydrogen atom migrates to the surface N atom that has lost hydrogen and this recovers to form NH<sub>2</sub>. Thus, in the subsequent discussion of hydrogen transfer on the Co(100) surfaces, the channel H is the active species for hydrogen transfer to the Cp ligand.

Table 5. The calculated reaction energy for hydrogen transfer step from surface H and channel H on Co(100) surface. The results for transfer of different H species show that channel H is more reactive than surface H on the Co(100) surface.

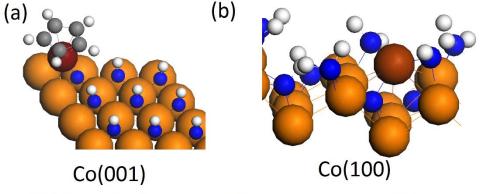
	High NH <sub>x</sub> Coverage	ALD Coverage
	Co(100)/eV	Co(100)/eV
Adsorption	-1.73	-1.67
hydrogen transfer channel H	-3.12	-2.19
hydrogen transfer surface H	Recover to NH <sub>2</sub>	Recover to NH <sub>2</sub>

The reaction pathway on Co(100) surface is shown in Figure 4(b). The reaction energies for all steps are exothermic, so that the loss of two Cp ligands as CpH via hydrogen is overall exothermic by -0.79eV. The CoCp<sub>2</sub> precursor initially adsorbs on the NH<sub>x</sub>-terminated Co(100) surface with a gain of -1.67eV. The first hydrogen transfer step is exothermic, with a moderate activation barrier of 0.52eV and a gain of -0.52 eV in energy. After the first Cp ligand desorption, the second hydrogen transfer step is practically thermoneutral with a moderate barrier of 0.85 eV and a small energy cost of 0.17eV. Finally, the desorption of the second CpH shows an energy gain of -0.79eV and the resulting surface structure shows a Co atom which binds to a nitrogen atom from which a H atom transferred to Cp.

At the highest  $NH_x$  coverage, the terminations on Co(100) are 1ML NH and 1ML  $NH_2$ . As shown on Figure S2(b) in supporting information, the overall reaction for Cp ligand eliminations via hydrogen transfer step is exothermic. The calculated barriers for the first and second hydrogen

transfer steps are 1.56 eV and 0.84 eV. However, the first CpH formation is endothermic with a high energy cost of 2.19eV. This is due to very strong adsorption of CoCp<sub>2</sub>H on the Co(100) surface after the hydrogen transfer step, which had an energy gain of -3.12eV. Thus, with the highest NH<sub>x</sub> coverage, the precursor is too strongly bound to the substrate.

To summarize, for NH<sub>x</sub> terminations of Co (001) and (100) at typical ALD operating conditions, at least one Cp ligand can be eliminated via a hydrogen transfer step. On the Co(001) surface, the activation barriers are 0.56eV and 1.57eV for the first and second hydrogen transfer steps. Due to high barrier and endothermic reaction, the elimination of second Cp ligand is unfavourable and the termination for a single Co(Cp)<sub>2</sub> precursor is a CoCp fragment on NH<sub>x</sub>-terminated Co(001) surface. On the Co(100) surface, the activation barrier for the first hydrogen transfer step is 0.52eV, while that for the second hydrogen transfer is 0.85eV. The first hydrogen transfer step should be facile, while the magnitude of the latter barrier, which is similar to earlier modelling work on metal oxides, e.g. https://pubs.rsc.org/en/content/articlehtml/2015/cp/c5cp01912e, Chemistry of Materials 2010, 22 (1), 117, means that this step would most likely be promoted at the upper end of the range of typical deposition temperatures. . The two Cp ligands are eliminated and the final desorption of CpH shows an energy gain of -0.79eV. The resulting surface termination after reaction of a single precursor is a Co atom deposited on NH<sub>x</sub>-terminated Co(100) surface, which binds to surface N atom. The distance between the deposited Co atom and nearest N atom is 1.84 □. These structures are shown in Figure 5. Not only the  $NH_x$  coverage, but also the surface facet plays a crucial role in the adsorption strength of the precursor and the energetics of the hydrogen transfer and CpH formation and removal steps.



NH<sub>x</sub> terminations at ALD operating condition

Figure 5. The configurations of the final structures after single metal precursor  $CoCp_2$  adsorption and ligand elimination on the (a) Co(001) and (b) Co(100) surface. The  $NH_x$  terminations are at ALD operating condition. The substrate Co atom is represented by orange colour and the Co atom from metal precursor  $CoCp_2$  is represented by wine red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

3.3 Precursor coverage effect on the reaction mechanism on Co (001) and (100) surfaces with  $NH_x$  terminations at ALD operating condition

We now address the adsorption and further reaction of two  $Co(Cp)_2$  precursors. The adsorption energy is calculated from:

$$E_{ad} = E_{tot} - E_{\frac{NHx}{Metal}} - 2 * E_A$$
 (2)

where  $E_{tot}$ ,  $E_{NHx/Metal}$ , and  $E_A$  are the energy of the  $NH_x$ -terminated metal slab with two precursor  $Co(Cp)_2$ , the slab model for the  $NH_x$ -terminated metal surface, and isolated precursor  $CoCp_2$ , respectively. Dividing the computed energy by two gives the adsorption energy per precursor. All energies are computed with the inclusion of the van der Waals corrections. The adsorption structures of two precursors on Co(001) and (100) surfaces at the ALD coverage of  $NH_x$  are shown in Figure 6.

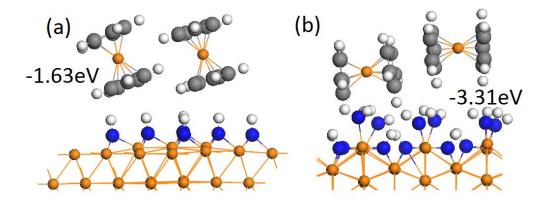


Figure 6. The configurations of two precursor CoCp<sub>2</sub> on NH<sub>x</sub>-terminated (a) Co(001) surface, and (b) Co(100). The Co atom is represented by orange. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

The preferred binding mode for two precursors is the same as for adsorption of a single precursor. On the Co(001) surface, the upright adsorption mode is the most stable, while on the (100) surface the horizontal adsorption mode is the most stable. These adsorption modes are exothermic with computed adsorption energies of -1.63eV and -3.31eV, giving adsorption energies per precursor are -0.81eV and -1.65eV. Thus on the (001) surface, precursor adsorption is enhanced compared with adsorption of a single precursor (-0.68eV). In this adsorption configuration we see tilting of the two CoCp<sub>2</sub> precursors. Compared with adsorption of a single CoCp<sub>2</sub> on the Co(100) surface, there is no difference in adsorption energy per precursor.

The further reaction of two precursor molecules of CoCp<sub>2</sub> on the NH<sub>x</sub>-terminated metal surfaces at ALD operating condition can proceed as follows:

$$C: H^* + CoCp_2 + CoCp_2 \rightarrow CoCp_2 + CoCp^* + HCp$$

$$D1: H^* + CoCp_2 + CoCp^* \rightarrow CoCp_2 + Co^* + HCp$$
 
$$D2: H^* + CoCp_2 + CoCp^* \rightarrow CoCp^* + CoCp^* + HCp$$

Here, after the first hydrogen transfer, Reaction C, the second hydrogen transfer can result in two different by-products. Reaction D1 results in a Co atom and an intact adsorbed CoCp<sub>2</sub>, while in reaction D2, two adsorbed CoCp fragments are present on the surface. The reaction energies of two precursor adsorption are with reference to the NH<sub>x</sub>-terminated metal surface and two free CoCp<sub>2</sub>. If one CpH molecule desorbs from the surface, the reaction energies are with reference to NH<sub>x</sub>-terminated metal surface, two free CoCp<sub>2</sub>, and free CpH. The reaction pathways are shown in Figure 7, with Reactions D1 and D2 shown in different colours and the calculated barriers for each hydrogen transfer step are presented in Table 6.

Table 6. The computed energy barriers for hydrogen transfer steps with respect to two metal precursors CoCp<sub>2</sub> adsorption on NH<sub>x</sub> terminated Co (001) and (100) surfaces at ALD conditions.

Computed Barriers/eV				
_	$C: H^* + CoCp_2 + CoCp_2 \rightarrow$	$D1: H^* + CoCp_2 + CoCp^* \rightarrow$	$D2: H^* + CoCp_2 + CoCp^* \rightarrow$	
	$CoCp_2 + CoCp^* + HCp$	$CoCp_2 + Co^* + HCp$	$CoCp^* + CoCp^* + HCp$	
Co(001)	0.74	1.18	0.75	
Co(100)	No barrier	0.79	1.78	

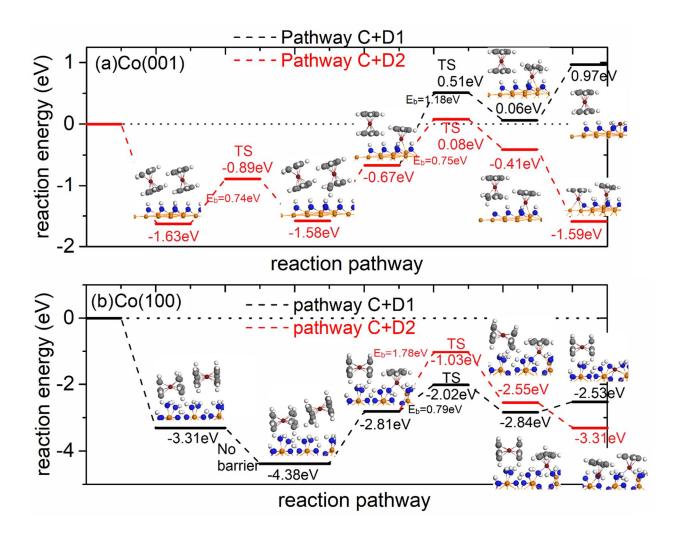


Figure 7. The CoCp<sub>2</sub> precursor reaction pathway on (a) Co(001) surface and (b) Co(100) surface for two CoCp<sub>2</sub> precursors. The black pathway is for reaction D1, resulting in a Co atom and adsorbed CoCp<sub>2</sub>. The red pathway is for reaction D2, resulting in two CoCp fragments at the surface. The substrate Co atom is represented by orange colour and the Co atom from metal precursor CoCp<sub>2</sub> is represented by wine red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

On the Co(001) surface, the first hydrogen transfer step is thermoneutral with a small energy change of 0.05eV and a moderate activation barrier of 0.74eV. This barrier is larger by 0.18eV compared to the barrier for the first hydrogen transfer for a single CoCp<sub>2</sub> precursor. This indicates that a neighbouring precursor might slightly hinder the reactivity on the Co(001) surface.

After the desorption of CpH, the elimination of the second Cp ligand for the same precursor, reaction D1, is unfavourable with an endothermic reaction energy and a high activation barrier of 1.18eV. The trend that formation of a bare Co atom is unfavourable on this surface persists with two CoCp<sub>2</sub> precursors.

Alternatively, following reaction D2, the intact CoCp<sub>2</sub> precursor undergoes hydrogen transfer from the surface, with a moderate activation barrier of 0.75 eV, which is similar to that of reaction C. The elimination of the second Cp ligand is not favoured. Thus, with adsorption of multiple precursors, the final termination remains CoCp fragments on Co(001) surface.

On the Co(100) surface, the adsorption of two metal precursors is exothermic with a large energy gain of -3.31eV. As previously described, the channel H atom is more reactive than surface H atom. The first hydrogen transfer of the channel H atom is exothermic and is barrierless. Thus, a neighbouring CoCp<sub>2</sub> precursor promotes the first hydrogen transfer step by reducing the activation barrier from 0.52eV to no barrier on Co(100) surface.

After the desorption of this CpH, the elimination of the second Cp ligand from the same precursor as reaction C, indicated as reaction D1, with a computed activation barrier of 0.79eV, has a much lower activation barrier than the hydrogen transfer to a Cp ligand of the neighbouring CoCp<sub>2</sub> precursor, denoted as reaction D2, which has an activation barrier of 1.78eV. Thus, the elimination of two Cp ligands from one precursor is favoured rather than elimination from each precursor and after two hydrogen transfer steps, the resulting structure is one Co atom deposited on the surface binding to N atom, and one intact CoCp<sub>2</sub> molecule.

After the elimination of two Cp ligands from a CoCp<sub>2</sub> precursor, we then further investigate hydrogen transfer and Cp elimination for the second CoCp<sub>2</sub> precursor, with the deposited Co on the Co(100) surface. The reactions are illustrated as follows:

$$E: H^* + CoCp_2 + Co^* \rightarrow CoCp^* + Co^* + HCp$$

$$F: H^* + CoCp^* + Co^* \rightarrow Co^* + Co^* + HCp$$

The results are summarized in Figure 8. Again, the channel H atom is more reactive than the surface H atom. The first hydrogen transfer step is slightly endothermic, but still highly exothermic overall, with a moderate activation barrier of 0.89eV. After the desorption of first CpH, the second hydrogen transfer step is exothermic and has a moderate barrier of 0.75eV. After the second CpH desorption, two Co atoms are deposited on the surface.

The computed activation barriers for H transfer for a single  $CoCp_2$  precursor, two surface bound  $CoCp_2$  species, and one precursor with a bare Co atom  $Co + CoCp_2$  on Co(100) surface are summarized in Table 7. We see that the activation barriers for Cp elimination via hydrogen transfer steps are all moderate and can be overcome at ALD operating condition and on this surface CpH elimination proceeds to leave Co atoms on the Co(100) surface. These Co atoms are bound to nitrogen atoms with Co-N distances of  $1.84\Box$  for single  $CoCp_2$  precursor and  $1.76\Box$  and  $1.88\Box$  for two Co atoms deposited on Co(100) surface (two surface bound  $CoCp_2$  species).

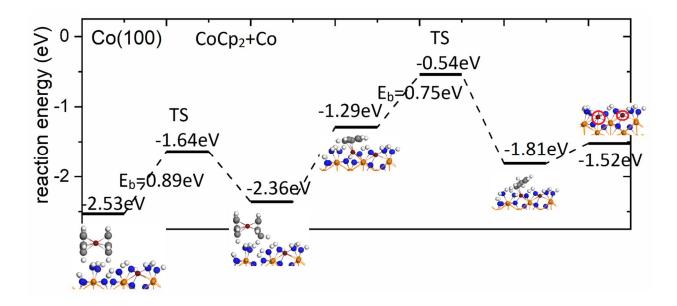


Figure 8. The plotted metal precursor reaction pathway on Co(100) surface with CoCp<sub>2</sub> reaction with one Co atom deposited on the surface. The substrate Co atom is represented by orange colour and the Co atom from metal precursor CoCp<sub>2</sub> is represented by wine red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

Table 7. The calculated activation barriers for first and second hydrogen steps on  $NH_x$ -terminated Co(100) surface with one precursor  $CoCp_2$ , two precursors  $CoCp_2 + CoCp_2$ , and one precursor and one Co atom deposited on the surface  $Co + CoCp_2$ .

Barriers/eV			
	$CoCp_2$	$CoCp_2 + CoCp_2$	$Co + CoCp_2$
1st hydrogen transfer	0.52	No barrier	0.89
2 <sup>nd</sup> hydrogen transfer	0.85	0.79	0.75

3.4 Final structures after metal precursor pulse on Co (001) and (100) surfaces at ALD operating condition

Based on the discussion in section 3.3, on the Co(001) surface, the preferred termination is the CoCp fragment. With this in mind, we consider the saturation coverage of CoCp fragments on Co(001) to determine the maximum coverage of CoCp after the metal precursor pulse. The results are summarized in Figure 9. We see that at most three CoCp fragments can be adsorbed on  $NH_x$ -terminated Co(001) surface, which results in a coverage of 3.03  $CoCp/nm^2$ .

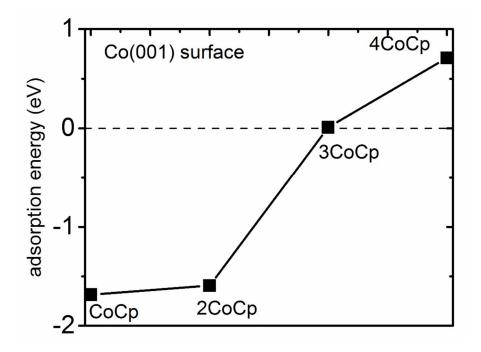


Figure 9. The adsorption energies of CoCp fragments on NH<sub>x</sub>-terminated Co(001) surface at various coverages.

On the Co(100) surface, the channel hydrogen atoms are involved in the hydrogen transfer steps and the two Cp ligands are eliminated with hydrogen transfer, CpH formation and desorption. At ALD operating condition, the NHx-termination is  $6NH + 6NH_2$ . In total, three Co atoms are deposited on the surface. To explore if the surface H atoms can be consumed in the Co(Cp)<sub>2</sub> pulse, we adsorb one CoCp<sub>2</sub> precursor on a structure in which all channel H atoms have been lost to study

the energetics of hydrogen transfer via surface H from NH<sub>2</sub>. The calculated reaction pathway is shown in Figure 10.

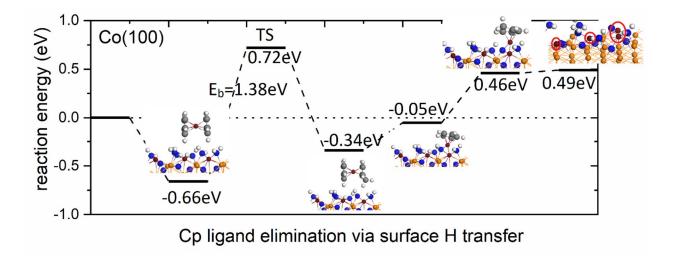


Figure 10. The plotted metal precursor reaction pathway on Co(100) surface with Cp ligand elimination via surface hydrogen transfer. The substrate Co atom is represented by orange colour and the Co atom from metal precursor CoCp<sub>2</sub> is represented by wine red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

With only surface H present, a CoCp<sub>2</sub> precursor has a moderate adsorption strength with a computed adsorption energy of -0.66eV. The first hydrogen transfer is endothermic by 0.32 eV and has a high barrier of 1.38eV. After desorption of CpH, the second hydrogen transfer is not favoured as the reaction energy is endothermic. Compared with channel hydrogen transfer, the Cp ligand elimination via surface hydrogen transfer is endothermic and has high activation barrier. Thus, we concluded that the reaction will stop after all channel H atoms are consumed. The final structure on Co(100) surface shows Co atoms deposited on the surface with the coverage of 3.33 Co/nm<sup>2</sup>. The final terminations after metal precursor pulse on Co (001) and (100) surfaces are shown in Figure 11. The reported growth rate of PE-ALD of Co using CoCp<sub>2</sub> and N-plasma varies

from 0.26 Å/cycle to 0.97 Å/cycle. $^{14, 17-18}$  Any remaining Cp ligands and NH $_{x}$  species can be eliminated during the next N-plasma step.

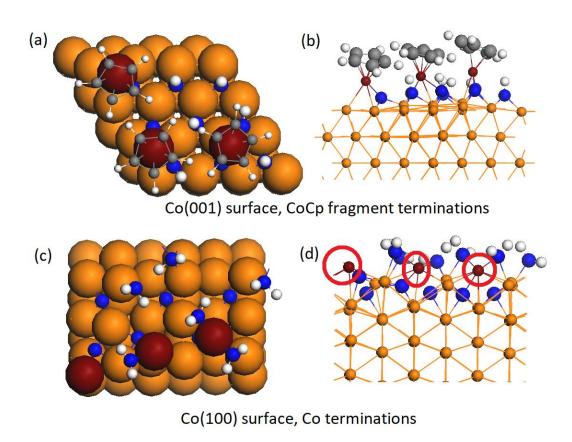


Figure 11. The configurations of final terminations after metal precursor pulse on Co(001) surface with (a) top view, and (b) side view, and Co(100) surface with (c) top view, and (d) side view. The substrate Co atom is represented by orange colour and the Co atom from metal precursor CoCp<sub>2</sub> is represented by wine red colour. Carbon, nitrogen and hydrogen atoms are represented by grey, blue and white colour, respectively.

#### 4. Discussion

The deposition of Co thin film using a  $CoCp_2$  precursor has been demonstrated experimentally.<sup>6</sup>, <sup>18, 38-39</sup> In plasma ALD studies, the co-reactants are NH<sub>3</sub> plasma or a mixture of N<sub>2</sub> and H<sub>2</sub> plasma.

The reported growth per cycle (GPC) and Co thin film resistivity varies with operating temperature and co-reactants. For example, when the NH<sub>3</sub> plasma is used, at a deposition temperature of 300°C, the GPC and resistivity were  $0.48 \square$  and  $10 \mu\Omega$ cm, respectively. In another study with a co-reactant of N<sub>2</sub>/H<sub>2</sub> plasma, the GPC was in the range of  $0.26 \square$  to  $0.65 \square$  for deposition temperatures in the range of 150°C to 450°C. The reported resistivity was  $18 \mu\Omega$ cm. It was found that a small amount of N (ca. 2-3 at. %) was present in the deposited Co film when using NH<sub>3</sub> or mixture of N<sub>2</sub> and H<sub>2</sub> plasma. A slightly higher amount of N (ca. 4.5 at. %) is present in the subsurface region. In the subsurface

In this paper, we start with NH<sub>x</sub>-terminated Co surfaces from ref. 31. This is in accordance with the experimental conclusions that the NH<sub>x</sub> species play an important role in the growth mechanism. <sup>18, 38-39</sup> On a bare Co surface, the dissociation of CoCp<sub>2</sub> precursor is difficult as a result of the strong Co-C bond. On NH<sub>x</sub>-terminated Co surfaces, the Cp ligand is eliminated via hydrogen transfer step and will desorb the surface as CpH. This is the key reaction mechanism during the metal precursor pulse.

From the calculation of activation barriers, the final surface termination on the Co(001) surface is CoCp fragments. The NH<sub>x</sub> species are not fully eliminated during this step. On the Co(100) surface, the two Cp ligands in CoCp<sub>2</sub> are eliminated and the final termination is Co-terminated NH<sub>x</sub>-covered surface in which Co binds to surface N atom. On this surface, the preference is for the channel hydrogen to be removed, leaving NH<sub>x</sub> species with surface hydrogen. These surface hydrogens are not reactive enough to promote CpH elimination.

When we consider multiple  $Co(Cp)_2$  precursors on the  $NH_x$ -terminated (100) and (001) surfaces, the reaction pathway is similar to that for a single precursor. On Co(001), the preference is for CoCp species to be present, while on Co(100), Cp is lost, leaving Co atoms bound to the surface.

The simultaneous adsorption of two  $Co(Cp)_2$  precursors can promote the CpH elimination on the Co(100) surface.

In the following N-plasma step, we expect that the surface CoCp fragment and surface terminating NH<sub>x</sub> species are eliminated with plasma radicals such as N, H, NH and NH<sub>2</sub>. However, the channel N is difficult to remove completely. This is consistent with the experimental finding of subsurface N atoms in Co metal. <sup>18, 38-39</sup> In addition, surface defect sites may affect the growth rate and the quality of deposited Co thin film. For example, a small amount of N is observed in the deposited Co thin film by experimental studies, <sup>18</sup> while low-coordinated Co atoms may be present due to imperfections. The analysis of the role of these imperfections requires deeper studies that can build on this work.

The study of the plasma step is beyond the scope of this paper. However, we can explore some possible reactions of the surface bound  $NH_x$  species after CpH elimination on the Co(100) surface, leading to the formation of  $N_2H_4$ ,  $NH_3$  or  $N_2$ . We discuss the results of possible surface reactions for the removal of surface  $NH_x$  species such as  $N_xH_y$  formation at the end of metal precursor pulse. For Co(100), the two Cp ligands are removed completely via proton transfer and CpH formation and desorption. The channel H atoms are removed during the proton transfer step. The surface H atoms are not removed in this mechanism due to high proton transfer barrier. The coverage of deposited Co atoms on the surface is  $3.33 \text{ Co/nm}^2$ . We consider  $N_2H_4$  formation, NH formation, and  $H_2$  formation and the results are summarized in Figure 12. The energies are discussed using as reference the energy of final structure of Co deposited on  $NH_x$ -terminated Co(100) surface as shown in Figure 12(a). After formation of  $N_2H_4$  and relaxation, the resulting structure reverts to two  $NH_2$  species terminating the surface, shown in Figure 12(b). NH formation and  $H_2$  formation are both endothermic, with energy costs of 0.26eV for NH formation and 1.54eV for  $H_2$  formation

respectively, compared to original final structure. These are shown in Figure 12(c)-(d). Thus, surface reactions for the removal of surface  $NH_x$  species after the  $Co(Cp)_2$  precursor pulse are not favourable and the  $NH_x$  species will be removed in the following plasma step.

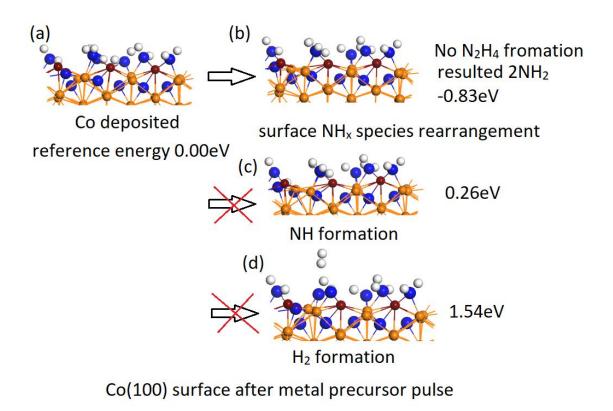


Figure 12. The configurations of surface reactions on Co(100) surface including (a) original Co deposited on  $NH_x$ -terminated surface, (b) surface  $NH_x$  species rearrangement, (c) NH formation, and (d)  $H_2$  formation.

#### 5. Conclusions

When depositing metals, a non-oxidizing reactant is preferred because the O-source will cause contamination and oxidize the metal. The PE-ALD of Co using metal precursor and N-plasma has been demonstrated experimentally, but the reaction mechanism is not well-understood. After the

N-plasma step, the resulted metal surfaces will be NH<sub>x</sub>-terminated. The nature and stability of NH<sub>x</sub>-terminated metal surfaces are studied in our previous published work.<sup>31</sup> The present work focuses on the reaction mechanism during the metal precursor pulse on the NH<sub>x</sub>-terminated surfaces. These final structures and terminations after metal precursor pulse are vital to model the following plasma pulse. The surface facets will result in different precursor adsorption orientation. CoCp<sub>2</sub> prefer up-right position with one Cp ring in close contact with NH<sub>x</sub>-terminated (001) surface, while they are in horizontal position with both of the Cp rings anchored to zigzag channel on NH<sub>x</sub>-terminated (100) surface.

The Cp ligands are eliminated via hydrogen transfer step and desorb from surface by forming CpH. The surface facet plays an important role in reaction energy and barriers for hydrogen transfer step. On the Co(001) surface, the NH<sub>x</sub> termination is 0.56ML NH at ALD operating condition. With single CoCp<sub>2</sub> adsorption, only one Cp ligand is eliminated with moderate activation barrier at the value of 0.56eV. With two CoCp<sub>2</sub> adsorption, the neighbouring CoCp<sub>2</sub> has hindered the reactivity by increasing the barrier to 0.74eV. The final termination on NH<sub>x</sub>-terminated Co(001) surface is CoCp fragments at the coverage of 3.03 CoCp/nm<sup>2</sup>.

On the Co(100) surface, the NH<sub>x</sub> termination is 0.67ML NH and 0.67ML NH<sub>2</sub> at ALD operating condition. Channel H atom is more reactive than surface H atom. With single CoCp<sub>2</sub> adsorption, the two Cp ligands are eliminated with moderate barriers at the value of 0.52eV and 0.85eV for the first and second hydrogen transfer. With two CoCp<sub>2</sub> adsorption, the neighbouring CoCp<sub>2</sub> has promoted the reactivity by lowering the first hydrogen transfer barrier to no barrier and the second hydrogen transfer barrier to 0.79eV. After all channel H atoms are consumed, the Cp ligand elimination with surface H transfer has high barrier at the value of 1.38eV. The final termination on NH<sub>x</sub>-terminated Co(100) is Co atoms deposited on the surface at the coverage of 3.33 Co/nm<sup>2</sup>.

During the following plasma step, the remaining Cp ligand (if any) and surface N atom are eliminated by  $N_xH_y$  radicals from the N-plasma (NH<sub>3</sub> or mixture of N<sub>2</sub> and H<sub>2</sub>). After the plasma pulse, the metal surface will be NH<sub>x</sub>-terminated and the while system is ready for the next cycle. The reaction mechanism of N-plasma step is currently the subject of further study.

## **Supporting information**

Configurations of the top view and side view of single NH and NH<sub>2</sub> adsorbed on the Co(100) surface. Plotted metal precursor reaction pathway on Co(001) surface and Co(100) surface with NH<sub>x</sub> terminations at zero K condition. The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/XXXXXXX

# Acknowledgements

We acknowledge generous support from Science Foundation Ireland (SFI) through the SFI-NSFC Partnership program, Grant Number 17/NSFC/5279, NITRALD and National Natural Science Foundation of China, Grant number 51861135105. Computing resources have been generously supported by Science Foundation Ireland at Tyndall and through the SFI/HEA-funded Irish Centre for High End Computing (www.ichec.ie).

## References

1. Tu, K., Recent Advances on Electromigration in Very-Large-Scale-Integration of Interconnects. *J. Appl. Phys.* **2003**, *94*, 5451-5473.

- 2. Greenslit, D. V.; Eisenbraun, E., Characterization of Ultrathin PEALD-Grown RuCo Films for Diffusion Barrier and Copper Direct-Plate Applications. *ECS Trans.* **2011**, *35*, 17-24.
- 3. Chakraborty, T.; Eisenbraun, E. T., Microstructure Analysis of Plasma Enhanced Atomic Layer Deposition-Grown Mixed-Phase RuTaN Barrier for Seedless Copper Electrodeposition. *J. Vac. Sci. Technol. A* **2012**, *30*, 020604 1-5.
- 4. Miikkulainen, V.; Leskelä, M.; Ritala, M.; Puurunen, R. L., Crystallinity of Inorganic Films Grown by Atomic Layer Deposition: Overview and General Trends. *J. Appl. Phys.* **2013**, *113*, 021301 1-101.
- 5. Johnson, R. W.; Hultqvist, A.; Bent, S. F., A Brief Review of Atomic Layer Deposition: From Fundamentals to Applications. *Mater. Today* **2014**, *17*, 236-246.
- 6. Kaloyeros, A. E.; Pan, Y.; Goff, J.; Arkles, B., Review—Cobalt Thin Films: Trends in Processing Technologies and Emerging Applications. *ECS J. Solid State Sci.* **2019**, *8*, P119-P152.
- 7. George, S. M., Atomic Layer Deposition: An Overview. *Chem. Rev.* **2009**, *110*, 111-131.
- 8. Profijt, H.; Potts, S.; Van de Sanden, M.; Kessels, W., Plasma-Assisted Atomic Layer Deposition: Basics, Opportunities, and Challenges. *J. Vac. Sci. Technol. A* **2011**, *29*, 050801 1-26.
- 9. Oviroh, P. O.; Akbarzadeh, R.; Pan, D.; Coetzee, R. A. M.; Jen, T. C., New Development of Atomic Layer Deposition: Processes, Methods and Applications. *Sci. Technol. Adv. Mater.* **2019**, *20*, 465-496.
- 10. Kim, H., Atomic Layer Deposition of Mmetal and Nitride Thin Films: Current Research Efforts and Applications for Semiconductor Device Processing. *J. Vac. Sci. Technol. B* **2003**, *21*, 2231-2261.
- 11. Knisley, T. J.; Kalutarage, L. C.; Winter, C. H., Precursors and Chemistry for The Atomic Layer Deposition of Metallic First Row Transition Metal Films. *Coordin. Chem. Rev.* **2013**, *257*, 3222-3231.
- 12. Kim, H., Area Selective Atomic Layer Deposition of Cobalt Thin Films. *ECS Trans.* **2008**, *16*, 219-225.
- 13. Lim, B. S.; Rahtu, A.; Gordon, R. G., Atomic Layer Deposition of Transition Metals. *Nat. Mater.* **2003**, *2*, 749-754.
- 14. Kim, H., High-Quality Cobalt Thin Films by Plasma-Enhanced Atomic Layer Deposition. *Electrochem. Solid-State Lett.* **2006**, *9*, G323-G325.
- 15. Kim, K.; Lee, K.; Han, S.; Jeong, W.; Jeon, H., Characteristics of Cobalt Thin Films Deposited by Remote Plasma ALD Method with Dicobalt Octacarbonyl. *J. Electrochem. Soc.* **2007**, *154*, H177-H181.
- 16. Zhu, B.; Ding, Z.-J.; Wu, X.; Liu, W.-J.; Zhang, D. W.; Ding, S.-J., Plasma-Enhanced Atomic Layer Deposition of Cobalt Films Using Co(EtCp)<sub>2</sub> as A Metal Precursor. *Nanoscale Res. Lett.* **2019**, *14*, 76 1-7.
- 17. Yoon, J.; Kim, D.; Cheon, T.; Kim, S.-H.; Kim, H., Atomic Layer Deposition of CoUusing N<sub>2</sub>/H<sub>2</sub> Plasma as A Reactant. *J. Electrochem. Soc.* **2011**, *158*, H1179-H1182.
- 18. Vos, M. F.; van Straaten, G.; Kessels, W. E.; Mackus, A. J., Atomic Layer Deposition of Cobalt Using H<sub>2</sub>-, N<sub>2</sub>-, and NH<sub>3</sub>-Based Plasmas: On the Role of the Co-reactant. *J. Phys. Chem. C* **2018**, *122*, 22519-22529.
- 19. Elliott, S. D., Atomic-Scale Simulation of ALD Chemistry. *Semicond. Sci. Tech.* **2012**, *27*, 074008 1-10.
- 20. Phung, Q. M.; Pourtois, G.; Swerts, J.; Pierloot, K.; Delabie, A., Atomic Layer Deposition of Ruthenium on Ruthenium Surfaces: A Theoretical Study. *J. Phys. Chem. C* **2015**, *119*, 6592-6603.
- 21. Elliott, S. D.; Dey, G.; Maimaiti, Y., Classification of Processes for The Atomic Layer Deposition of Metals Based on Mechanistic Information From Density Functional Theory Calculations. *J. Chem. Phys.* **2017**, *146*, 052822 1-11.
- 22. Phung, Q. M.; Vancoillie, S.; Pourtois, G.; Swerts, J.; Pierloot, K.; Delabie, A., Atomic Layer Deposition of Ruthenium on a Titanium Nitride Surface: A Density Functional Theory Study. *J. Phys. Chem. C* **2013**, *117*, 19442-19453.
- 23. Fang, G.; Xu, L.; Cao, Y.; Li, A., Theoretical Design and Computational Screening of Precursors for Atomic Layer Deposition. *Coordin. Chem. Rev.* **2016**, *322*, 94-103.
- 24. Holme, T. P.; Prinz, F. B., Atomic Layer Deposition and Chemical Vapor Deposition Precursor Selection Method Application to Strontium and Barium Precursors. *J. Phys. Chem. A* **2007**, *111*, 8147-8151.

- 25. Puurunen, R. L., Surface Chemistry of Atomic Layer Deposition: A Case Study for The Trimethylaluminum/Water Process. *J. Appl. Phys.* **2005**, *97*, 121301 1-52.
- 26. Elliott, S.; Scarel, G.; Wiemer, C.; Fanciulli, M.; Pavia, G., Ozone-Based Atomic Layer Deposition of Alumina from TMA: Growth, Morphology, and Reaction Mechanism. *Chem. Mater.* **2006**, *18*, 3764-3773.
- 27. Langereis, E.; Bouman, M.; Keijmel, J.; Heil, S.; Van de Sanden, M.; Kessels, W., Plasma-Assisted ALD of Al<sub>2</sub>O<sub>3</sub> at Low Temperatures: Reaction Mechanisms and Material Properties. *ECS Trans.* **2008**, *16*, 247-255.
- 28. Rai, V. R.; Vandalon, V.; Agarwal, S., Surface Reaction Mechanisms During Ozone and Oxygen Plasma Assisted Atomic Layer Deposition of Aluminum Oxide. *Langmuir* **2010**, *26*, 13732-13735.
- 29. Weckman, T.; Laasonen, K., First Principles Study of The Atomic Layer Deposition of Alumina by TMA–H<sub>2</sub>O-Process. *Phys. Chem. Chem. Phys.* **2015**, *17*, 17322-17334.
- 30. Elliott, S. D., Mechanism, Products, and Growth Rate of Atomic Layer Deposition of Noble Metals. *Langmuir* **2010**, *26*, 9179-9182.
- 31. Liu, J.; Nolan, M., Coverage and Stability of NHx-Terminated Cobalt and Ruthenium Surfaces: A First-Principles Investigation. *J. Phys. Chem. C* **2019**, *123*, 25166-25175.
- 32. Kresse, G.; Joubert, D., From Ultrasoft Pseudopotentials to The Projector Augmented-wave Method. *Phys. Rev. B* **1999**, *59*, 1758-1775.
- 33. Perdew, J. P.; Chevary, J. A.; Vosko, S. H.; Jackson, K. A.; Pederson, M. R.; Singh, D. J.; Fiolhais, C., Atoms, Molecules, Solids, and Surfaces: Applications of The Generalized Gradient Approximation for Exchange and Correlation. *Phys. Rev. B* **1992**, *46*, 6671-6687.
- 34. Perdew, J. P.; Burke, K.; Ernzerhof, M., Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **1996**, *77*, 3865-3868.
- 35. Monkhorst, H. J.; Pack, J. D., Special Points for Brillouin-zone Integrations. *Phys. Rev. B* **1976**, *13*, 5188-5192.
- 36. Maimaiti, Y.; Elliott, S. D., Precursor Adsorption on Copper Surfaces as the First Step during the Deposition of Copper: A Density Functional Study with van der Waals Correction. *J. Phys. Chem. C* **2015**, *119*, 9375-9385.
- 37. Henkelman, G.; Uberuaga, B. P.; Jónsson, H., A Climbing Image Nudged Elastic Band Method for Finding Saddle Points and Minimum Energy Paths. *J. Chem. Phys.* **2000**, *113*, 9901-9904.
- 38. Oh, I.-K.; Kim, H., Growth Mechanism of Co Thin Films Formed by Plasma-Enhanced Atomic Layer Deposition Using NH<sub>3</sub> as Plasma Reactant. *Curr. Appl. Phys.* **2017**, *17*, 333-338.
- 39. Reif, J.; Knaut, M.; Killge, S.; Winkler, F.; Albert, M.; Bartha, J. W., In Vacuo Studies on Plasma-Enhanced Atomic Layer Deposition of Cobalt Thin Films. *J. Vac. Sci. Technol. A* **2020**, *38*, 012405 1-9.

# **Table of Contents Graphic**



Final terminations after metal precursor  $CoCp_2$  pulse