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Density Functional Theory Predictions of the Composition of ALD-grown Ternary Oxides

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The surface reactivity of various metal precursors with different alkoxide, amide and alkyl ligands during the atomic layer deposition (ALD) of ternary oxides was determined using simplified theoretical models. Quantum chemical estimations of the Brønsted reactivity of a metal complex precursor at a hydroxylated surface are made using a gas-phase hydrolysis model. The geometry optimized structures and energies for a large suite of 17 metal precursors (including cations of Mg, Ca, Sr, Sc, Y, La, Ti, Zr, Cr, Mn, Fe, Co, Ni, Cu, Zn, Al and Ga) with 5 different anionic ligands (conjugate bases of tert-butanol, tetramethyl heptanedione, dimethyl amine, isopropyl amidine and methane) and the corresponding hydrolyzed complexes are calculated using density functional theory (DFT) methods. The theoretically computed energies are used to determine the energetics of the model reactions. These DFT models of hydrolysis are used to successfully explain the reactivity and resulting stoichiometry in terms of metal cation ratios seen experimentally for a variety of ALD-grown ternary oxide systems.

KEYWORDS: Atomic layer deposition, density functional theory, ternary oxides, stoichiometry, metal precursors, ligand.

Ternary oxides $(M_x^1 M_{1-x}^2 O_y)$ are a large class of functional materials (including the well known perovskites with ABO₃ structures¹) which have produced much interest due to their potential in a wide-range of technological applications. The properties of these materials are dependent on the ratios of metal cations in their structures. Atomic layer deposition (ALD)² has been applied successfully in the growth of high quality thin films of binary oxides $(M^1O_x)^{3-6}$ and is seen as a potential method for depositing nanoscale ternary oxide $(M^1M^2O_x)$ structures with controlled stoichiometries. ALD ternary oxide thin films are produced in a similar manner to binary oxides: a substrate is exposed to a metal precursor containing metal M^1 within an ALD reactor for a set pulse time allowing a coating of the precursor to form; the ALD reactor is then purged/evacuated to remove excess metal precursor; after purging, a second precursor pulse occurs where an oxygen source (e.g. ozone) is allowed to flow into the reactor, removing the remaining ligands. This cycle is repeated n_1 times until a sufficient amount of M^1O_x is deposited. This is followed by n_2 cycles using a different metal precursor containing M^2 and an appropriate oxygen source (which could be different *e.g.* water), depositing M^2O_x and creating an $M^1M^2O_x$ thin film. The M^1+M^2 supercycle is repeated until a desired $M^{1}M^{2}O_{x}$ film thickness is achieved.

Film composition of $M^1M^2O_x$ solids have been found to depend in a consistent but nonlinear way on the ALD pulse ratios, $P(M^1) = n_1 / (n_1 + n_2)$, revealing differences between the chemistry of the two metal precursors^{7,8}. From the application of a 'Surface Utilization Model' to the resulting stoichiometries of a series of ternary oxides produced using β diketonate metal precursors, the fraction of ligands eliminated by Brønsted reactions in the metal pulse before the O₃ pulse is found to be strongly dependent on the identity of the pair of metal cations in the particular ternary oxide⁸. However, the Surface Utilization Model could not account for some experimental stoichiometries (e.g. LaMnO_x, CaMnO_x and CoFeO_x),

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which suggests that more complex surface chemistry is at play. In this study we extend the model and attempt to account for the resulting stoichiometries of a series of metal precursors that have been combined pairwise in thermal ALD by determining the relative reactivities of these precursors towards ligand elimination. The reactivity of the precursors affects the deposition rates of individual metal cations and hence the final composition of the ternary oxide.

Two mechanisms have been proposed for the non-linear growth of ternary oxides using the ALD of organo-metallic precursors combined pair-wise followed by oxidation with ozone or oxygen plasma⁸. Mechanism A assumes no elimination of the surface-bound ligands before the oxidizing pulse (either because the ligands are inert or because there are no surface hydroxyl groups), followed by removal of all ligands via combustion. This leads to an expression for the final stoichiometry of the oxide layer that is simply dependent on the number of precursor pulses, n, and the charge of the metal cation, q.

$$D_1^A = (n_1/q_1)/[(n_1/q_1) + (n_2/q_2)]$$
(1)

There is no fitting to experimental data for Mechanism A. We note that, as shown in Figure 1, D^A is a linear function of pulse ratio only if $q_1=q_2$.

Mechanism B makes the assumption that hydroxyl groups remain on the surface from previous oxidation steps⁹. These may then act as a source of protons and allow a fraction of the ligands to be eliminated before combustion, thus permitting more metal precursor to be deposited. Therefore in Mechanism B (Figure 1), the growth rate of a metal oxide and stoichiometry of the thin film are additionally dependent on the fraction of ligands eliminated by combination with protons available on the surface, *f*, which is obtained by attempting to fit D^B to the experimental data-

$$D_1^B = (n_1(f_1+1)/q_1)/[(n_1(f_1+1)/q_1) + (n_2(f_2+1)/q_2)]$$
(2)

 A third mechanism that may affect the ALD growth rate of a metal oxide and influence the resulting ternary oxide composition is a ligand exchange process. In this proposed Mechanism C, a rapid exchange of ligands may occur from an adsorbed M^1 metal precursor to an M^2 metal cation at the growing oxide surface. Ligand elimination via Mechanism B may then be considered not from M^1 but from M^2 to which the ligand is now bonded. We suggest that Mechanism C explains cases where neither Mechanism B nor Mechanism A fully account for experimental stoichiometries.



Figure 1. Predicted ratios of M^1 ions in $M^1M^2O_x$ ternary oxide, $D(M^1)$, as a function of the M^1 pulse ratio, $P(M^1)$, with different ionic charges, q_1 and q_2 . The ratios predicted using Mechanism A (Equation 1) are shown in black. The compositions predicted using Mechanism B (Equation 2) with higher ratios of Brønsted eliminated M^1 ligands compared to M^2 are shown by the solid blue (3:2), green (4:1) and red (1:0) curves while those with lower

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ratios of M^1 ligands eliminated are shown by the dashed blue (2:3), green (1:4) and red (0:1) curves.

Quantum chemical and in particular density functional theory (DFT) calculations offer a convenient and reliable way of predicting many experimentally measureable chemical properties of the majority of main-group element molecules e.g. molecular geometries and vibrational frequencies¹⁰. Calculations involving transition metal complexes are susceptible to errors in electronic exchange and correlation and careful selection of the DFT functional is important to ensure accurate geometries¹¹⁻¹³ and reaction enthalpies¹⁴. ALD surface chemistries may be accurately described by explicitly modeling the atomic structure of a given oxide using, for example, slab calculations with periodic boundary conditions. This is so time-consuming that it has been carried out for only a small selection of binary oxides so far¹⁵. Instead, the approach taken here is to study a simple gas-phase model that describes the bond making and bond breaking essential to the surface reactions of interest¹⁶, thus allowing a larger number of chemical systems to be considered^{17,18}. The model takes no account of surface geometry or reaction pathways, other than the possibility of proton transfer (Mechanism B) and ligand exchange (Mechanism C).

We present here the potential ALD metal complexes $M(L)_q$ where M is a metal cation with a charge q of +2, +3 or +4 and L is an anionic ligand with a –1 charge. 17 metals in total were examined: 7 M(II) cations (M = Mg, Ca, Sr, Co, Ni, Cu or Zn), 8 M(III) cations (M = Sc, Y, La, Cr, Mn, Fe, Al or Ga) and 2 M(IV) cations (Ti or Zr). The ligands considered include two that bond via oxygen namely, tert-butanolate, O^tBu [OC(CH₃)₃], and the βdiketonate thd [2,2,6,6-tetramethylheptane-3,5-dionate], two that bond via nitrogen namely, dimethyl amide, dma [N(CH₃)₂] and the isopropyl amidinate amd [R¹NC(R²)NR¹ with R¹ = ⁱPr = C(CH₃)₂ and R² = CH₃] and one alkyl, methyl (CH₃). These ligands may be grouped not only by their coordinating atoms (O, N and C) but also by their denticity where O^tBu, dma and CH₃ are monodentate (κ_1) and thd and amd are bidentate (κ_2). It is appreciated that many of the complexes presented in this work are experimentally unviable due to unstable M-L bonds or steric hindrance at coordination sites but are included for completeness. A simple model for the elimination and exchange of precursor ligands by reaction with surface groups will be presented in sections 2.1 and 2.2 and thermodynamic energies for these reactions determined using DFT in sections 3.3, 3.4 and 3.5. The DFT results will be compared in section 4 to experimental results in the literature and in particular to those in ref 8, where the reaction mechanisms proposed using the surface utilization model for the deposition of ternary oxides using M(thd)_q metal precursors will be assessed.

2. Computational method

2.1.Gas-phase hydrolysis model



Figure 2. Representation of gas-phase hydrolysis (top) as a model for Brønsted elimination of ligand from ALD precursor at a surface (bottom).

As a model for the elimination of H-L from the surface, the gas-phase energetics for the hydrolysis of $M(L)_q$ may be calculated using quantum chemical methods^{17,18}.

$$\Delta E_{\text{hyd}} = 1/q \cdot \Delta E \text{ for } M(\kappa_1 - L)_q + q(H_2O) \to M(OH)_q + q(H-L)$$
(3a)

$$\Delta E_{\text{hyd}} = 1/q \cdot \Delta E \text{ for } M(\kappa_2 - L)_q + 2q(H_2O) \rightarrow M(OH)_q(OH_2)_q + q(H-L)$$
(3b)

In equations 3a and 3b, ΔE is quoted per ligand to allow comparison between the different ML₂, ML₃ and ML₄ complexes. A more negative ΔE_{hyd} value corresponds to a more exothermic hydrolysis reaction and a greater possibility of H-L elimination. As this model is concerned solely with the changes in bonding, temperature effects are neglected. In this model H₂O represents the source of hydroxyl groups on the surface while the substitution of OH groups for the ligands models the formation of M-O bonds in the solid (Figure 2). We stress that although H₂O appears in the calculations, the mechanisms assume ALD with an

oxidizing co-reagent such as ozone or oxygen plasma. In order to negate the effects from the change in coordination number for ligands of higher denticity (*e.g.* κ_2), the energies for the hydrated complexes M(OH)_q(OH₂)_q have been calculated rather than M(OH)_q for the monodentate ligands (κ_1). The extent to which H-L is eliminated will therefore depend on the free energy change for the hydrolysis reaction, ΔE_{hyd} .

 ΔE_{hyd} may be separated into two factors. The first factor is the Brønsted basicity (BB) of L⁻ relative to OH⁻ in forming the neutral H-L molecule and is given by:

$$\Delta E_{\rm BB} = \Delta E \text{ for } L^- + H_2 O \rightarrow H - L + O H^-$$
(4)

This factor is used to compare the Brønsted basicity of the ligands, *i.e.* the more negative ΔE_{BB} is, the stronger the BB of the ligand anion. The second factor is the Lewis basicity (LB) of L⁻ relative to OH⁻ in forming adducts with a particular metal:

$$\Delta E_{\rm LB} = 1/q \cdot \Delta E \text{ for } \mathsf{M}(\mathsf{OH})_q + q(\mathsf{L}^-) \to \mathsf{M}(\kappa_1 - \mathsf{L})_q + q(\mathsf{OH}^-)$$
(5a)

$$\Delta E_{\text{LB}} = 1/q \cdot \Delta E \text{ for } \mathsf{M}(\mathsf{OH})_q(\mathsf{OH}_2)_q + q(\mathsf{L}^-) \to \mathsf{M}(\kappa_2 - \mathsf{L})_q + q(\mathsf{OH}^-) + q(\mathsf{H}_2\mathsf{O})$$
(5b)

The more exothermic ΔE_{LB} is, the more stable ML_q is compared to $M(OH)_q$ and the less likely M^{+q} is to form M-O bonds. Comparing the expressions for hydrolysis and Brønsted and Lewis basicity it is clearly seen that $\Delta E_{\text{hyd}} = \Delta E_{\text{LB}} - \Delta E_{\text{BB}}$ *i.e.* the tendency to eliminate ligands is due to the difference in ligand basicity for M^{q+} and H^+ . Consequently, the more negative ΔE_{BB} is relative to ΔE_{LB} , the more likely it is for H-L to be eliminated from ML_q .

2.2. Ligand exchange model



Figure 3. Representations of the gas-phase model (top) for ligand exchange at a surface (bottom).

Another possible reaction that may influence the growth rate is the exchange of a ligand from an incoming or adsorbed M^1L_q complex with pre-existing M^2 metal ions at the growing oxide surface. Lewis acidic and basic sites at typical metal oxide surfaces allow dissociative adsorption of metal complexes¹⁹ and migration of ligands²⁰. Brønsted elimination of L may then follow from ligands now bound to the original surface metal (M^2 -L) rather than the incoming precursor. Following the simple model developed for the hydrolysis of ML_q , the free energy change may be calculated for the reaction of M^1L_q with $M^2(OH)_q$ [or $M^2(OH)_q(OH_2)_q$ for κ_2] resulting in $M^1(OH)_q$ [or $M^1(OH)_q(OH_2)_q$ for κ_2] and M^2L_q . In this model the hydroxylated metal complexes represent the metal ions in the oxide surface. All ΔE values are divided by the total number of ligands L *i.e.* the product of *p* and *q* in equations 6a and 6b.

$$\Delta E_{\text{exc}} = \Delta E / (p \cdot q) : p M^{1}(\kappa_{1} - L)_{q} + q M^{2}(OH)_{p} \rightarrow p M^{1}(OH)_{q} + q M^{2}(\kappa_{1} - L)_{p}$$
(6a)

$$\Delta E_{\text{exc}} = \Delta E / (p \cdot q) : p M^{1}(\kappa_{1} - L)_{q} + q M^{2}(OH)_{p}(OH_{2})_{p}$$
(6b)

 $\rightarrow p M^1(OH)_q(OH_2)_q + q M^2(\kappa_1-L)_p$

When assessing the elimination of L before the ozone pulse, direct elimination by reaction of the M^1L_q precursor with surface hydroxyl groups is considered first using ΔE_{hyd} . If hydrolysis is predicted to be thermodynamically unfavourable or if the ΔE_{hyd} values for two ML_q reactions are similar, then the possibility of ligand exchange between the precursor metal M^1 and surface metal M^2 is considered using ΔE_{exc} . Elimination of L is then reassessed using the ΔE_{hyd} value for M^2L .

2.3.DFT calculations

The optimized geometries and ground state energies of all reactants and products were determined as isolated molecules in vacuum. DFT calculations were performed using the TURBOMOLE suite of programs^{21,22}. The ground state properties for all species were calculated using the generalized gradient approximation (GGA) density functionals BP86^{23,24} and PBE^{25,26} (with the resolution of identity, RI approximation^{27,29}) and the hybrid functional PBE0^{25,26,30} (sometimes referred to as PBE1PBE). Ahlrich's def2-TZVPP basis sets³¹ were used for all atoms while effective core potentials were applied for the larger Sr, Y, Zr and La atoms^{32,34}. Calculations were carried out using different DFT functionals to assess the reliability of applying DFT calculated energies to determine the thermodynamic values presented in Sections 2.1 and 2.2 above. BP86 was chosen for this work as it had been used previously in a similar study of lanthanum complexes¹⁷ while the other GGA functional PBE was used to investigate the robustness of GGA-type functionals. The hybrid DFT functional PBE was used to investigate the robustness of GGA-type functionals in calculating reaction

 energies^{14,35}. Where more than one stable structural isomer was found to exist, the lowest energy isomeric form was determined and selected for the reaction energy calculations.

3. Results and discussion

3.1.Electronic state

Table 1. Calculated total electronic energies in atomic units (E_h) of the various spin states of Mn(thd)₃ and Mn(OH)₃(OH₂)₃ complexes using different DFT functionals. Note: Energy quoted for BP86 Mn(thd)₃ singlet is for spin contaminated form.

	BP86		PBE		PBE0	
	Mn(thd) ₃	Mn(OH) ₃ (OH ₂) ₃	Mn(thd) ₃	$Mn(OH)_3(OH_2)_3$	Mn(thd) ₃	$Mn(OH)_3(OH_2)_3$
Singlet	-2895.307	-1608.290	-2892.546	-1607.352	-2892.707	-1607.324
Triplet	-2895.325	-1608.330	-2892.594	-1607.392	-2892.769	-1607.376
Quintet	-2895.314	<u>-1608.368</u>	-2892.583	-1607.433	-2892.784	-1607.431
	1		1		1	

The spin states of the complexes were varied so as to find the electronic ground state. This showed that most neutral molecular species were closed shell electronic systems, except the Fe and Cu complexes which were found to have doublet ground states, the Cr complexes which had quartet ground states and the Co and Mn complexes. For the $Co(OH)_2$ and $Co(OH)_2(H_2O)_2$ molecular systems the quartet ground state was determined to be lowest in energy for calculations using the three DFT functionals used in this work. However, for the calculations using the BP86 and PBE functionals, the energies of doublet ground states of the other CoL_q complexes were found to be lower than those of the quartet. Calculations using PBE0 were consistent with the hydroxylated Co complexes. For all manganese complexes considered in this work, with the exception of Mn(thd)₃, the quintet ground state with 4

unpaired electrons was determined using BP86, PBE and PBE0 functionals to have the lowest energy spin state.

The wavefunction for Mn(thd)₃ proved difficult to determine using the BP86 functional as no singlet state could be converged to a spin-pure state with the correct occupancy. The application of Fermi orbital shifting³⁶ to an initially singlet state led to significant spin contamination with $\langle S^2 \rangle = 1.010$. The lowest energy complex with a stable wavefunction determined for Mn(thd)₃ using this functional was a non-spin-contaminated triplet state with two unpaired electrons while the quintet state with 4 unpaired electrons had a slightly higher energy (see Table 1).

Calculations using the PBE and PBE0 functionals for all the Mn complexes converged to singlet, triplet or quintet wavefunctions without spin contamination. As was found using BP86, for calculations with PBE, the triplet ground state for Mn(thd)₃ was determined to have the lowest energy. For both BP86 and PBE functionals, the quintet state (*i.e.* 4 unpaired electrons) of Mn(OH)₃(OH₂)₃ was the most stable, which was typical for all other MnL₃ complexes except Mn(thd)₃. For the calculations using PBE0, the quintet state had a lower energy than the singlet and triplet ground states for both Mn(thd)₃ and Mn(OH)₃(OH₂)₃. When calculating the changes in free energy, only energies from Mn and Co complexes with the same spin state were used (*e.g.* ΔE_{hyd} was calculated for BP86 using the energies of Mn(thd)₃ and Mn(OH)₃(OH₂)₃ both with $\langle S^2 \rangle = 2$).

3.2.Optimized geometries



Figure 4. Optimized geometries of neutral H-L species: (a) α -ene β -hydroxy ketone H-thd; (b) di-isopropyl acetamidine (H-amd); (c) tert-butanol (H-OtBu); (d) dimethyl amine (H-dma); (e) methane (CH₄). Depictions are not shown to scale.

The geometries for the anionic ligand species L^- and corresponding neutral H-L molecules were optimized using DFT methods. The lowest energy forms for the 5 H-L molecules are depicted in Figure 4. The enol tautomer of H-thd shown in Figure 4.a was found to be lower in energy than both the keto tautomer and alternative enol form with the H atom rotated away from the non H bonded carbonyl group in agreement with " β -diketones" in previous calculations³⁷ and experiments in weakly interacting media³⁸. Both the H-thd molecule and thd⁻ ligand were determined to have a planar "backbone" with the carboxyl and alcohol groups as well as the heptane carbon chain, including the C atoms of the methyl groups *trans* to the O groups, lying in one plane. The bulkier methyl group attached to the central C atom of the two amd species (H-amd shown in Figure 4b) were found to interact with the terminal



isopropyl groups, distorting the backbone of both the anionic ligand and the neutral molecule slightly out of plane.



Figure 5. Examples of DFT optimized geometries of $M(L)_q$ complexes determined using PBE0/TZVPP. A.I-IV thd complexes of Zr, Mn, Mg and Cu respectively, B.I-IV Ti, Fe, Ca and Ni complexes of amd, C.I-IV Ti, Ga, Sr and Co complexes of O^tBu, D.I-IV Zr, La, Zn and Cu complexes of dma and E.I-IV Ti, Al, Zn and Ni complexes of CH₃. Grey=C,

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white=H, red=O, blue=N, metals are various colours. Depictions of the molecules are not shown to scale.

The coordination geometry of the complexes containing the bidentate ligands thd and amd were generally very similar. Two different geometrical alignments of the thd and amd ligands were determined for the ML₂ complexes. For M = Mg, Ca, Sr and Zn, the geometries in the coordination sphere were found to be of D_{2d} symmetry, with the metal bound to the 4 O/N atoms of the ligands in a distorted tetrahedral arrangement (cf. Figure 5 A.III and B.III). The $M(thd)_2$ and $M(amd)_2$ complexes of transition metals (M = Co, Ni and Cu) have an approximate symmetry of D_{2h} with the 4 ligand-metal bonds in one plane (cf. Figure 5 A.IV and B.IV). The optimized geometries for all $M(thd)_3$ and $M(amd)_3$ molecules, where M = Sc, Y, La, Mn, Fe, Cr, Al and Ga, were found to be of $\sim C_3$ symmetry with the bulky ligands arranged radially around the metal centre and their O atoms approximating an octahedron (cf. Figure 5 A.II and B.II). For the $M(thd)_4$ complexes calculated in this work, the central metal ions were coordinated to the ligands in an approximate anti-prism geometry (cf. Figure 5 A.I). $Zr(amd)_4$ was found to have a similar geometric arrangement of ligands to the M(thd)₄ molecule with Zr-N bond lengths of between 2.3 and 2.4 Å. In the structure determined for $Ti(amd)_4$, only one of the amd ligands was bidentate (Ti-N bond lengths of 2.19 and 2.1 Å) with the other 3 ligands monodentate with Ti-N distances of ~2.0 and ~3.0 Å (cf. Figure 5 B.I). This difference in structure between Ti and Zr and complexes is probably due to the smaller ionic radius of Ti and the resulting steric hindrance in the ligand sphere. The geometric arrangement of the atoms of the thd ligands were similar to those for the free thd⁻ anion while for most the $M(amd)_q$ complexes the atoms exhibited a similar atomic arrangement as the free anion (c.f. H-thd and H-amd in Figure 4). A different atomic arrangement of the amd ligand, where one of the terminal isopropyl groups was rotated by 180° relative to that shown in Figure 4, was found to be the lower energy configuration for the bulkier $Ti(amd)_4$ and $Zr(amd)_4$ complexes and for the Ni(amd)₂ complex.

Less steric hindrance and more flexible M-L bonding leads to a greater variety of geometric structures for the monodentate ligand complexes. Of the $M(O^{1}Bu)_{2}$ complexes, Mg, Co and Cu have linear O-M-O bonds. Mg, Co and Cu have C-O-M bond angles of 180°, 149° and 129° respectively (Co(O¹Bu)₂ in Figure 5 C.III). Ca, Sr and Ni(O¹Bu)₂ are bent, with O-M-O angles of 162°, 139° and 142° respectively (Figure 5 C.IV), while $Zn(O^{1}Bu)_{2}$ has a slightly bent O-M-O angle of 172°. Like Mg, Ca and Sr have linear C-O-M bond angles whereas Zn and Ni have C-O-M angles of 124° and 132° respectively. In all the $M(O^{1}Bu)_{3}$ complexes the 3 O atoms were found to coordinate in a trigonal planar manner, except for La(O¹Bu)₃ where a O-La-O angle of 114° resulted in a trigonal pyramidal structure. Al and Ga were determined to have bent C-O-M linkages of 136° and 126° in the MO₃ plane (Figure 5 C.II), while Fe has an out of plane C-O-M angle of 135°. Mn(O¹Bu)₃ has a mixture of two in plane C-O-M angles with 139° each and one out of plane angle of 158°. Sc, Y and La complexes have linear arrangements of C-O-M. In the Ti and Zr(O¹Bu)₄ complexes, the O atoms have a tetrahedral coordination structure with C-O-M bond angles of 155° and 180° degrees respectively (Figure 5 C.I).

The seven $M(dma)_2$ complexes were determined to have linear N-M-N bonding, differing between each other in the relative orientation of the two dma ligands. Mg, Ca, Sr and Zn complexes had an approximate symmetry of D_{2d} where the C-N-C planes of the ligands were at 90° to each other (Figure 5 D.III) whereas Co, Cu and Ni had $\sim D_{2h}$ symmetry with both sets of C-N-C bonds in the same plane (Figure 5 D.IV). For all the M(dma)₃ complexes, the N atoms were coordinated in a trigonal planar manner to the metal centre. For Sc, Y, Al and Ga the dma ligands were rotated slightly out of plane with a C₃ point group symmetry. The ligands of Fe(dma)₃ were at right angles to the MN₃ plane while in La(dma)₃ the ligands were

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in-plane (Figure 5 D.II). $Mn(dma)_3$ was determined to have 2 ligands at 90° to the coordination plane and 1 ligand in-plane. Ti and $Zr(dma)_4$ were calculated to have geometries with T_d symmetry about the metal.

Five of the M(CH₃)₂ complexes considered in this work were found to have geometries of C_{2v} symmetry with Ni, Co, Sr, Cu and Ca having bent C-M-C bond angles of 103.4°, 110.2°, 114.2°, 130.8° and 131.1° respectively. The remaining two M(CH₃)₂ complexes with M = Zn and Mg had C-M-C bond angles of 180°. Sc, Y, La, Cr and Fe had pyramidal geometries with C-M-C bond angles of 118.5°, 110.7°, 105.6°, 112.8° and 108.4°. Trigonal planar coordination geometries were determined for Al and Ga complexes (C-M-C bond angles of 120°) while the C coordinating atoms of Mn(CH₃)₃ were also planar two of the C-Mn-C bond angles were 135° with the third angle 90°. For Ti(CH₃)₄ and Zr(CH₃)₄, the C atoms were coordinated in an ideal tetrahedral arrangement (*i.e.* C-M-C = 109.5°).

The optimized geometries with the lowest energy isomeric configuration for the hydroxylated metals, $M(OH)_q$, and the corresponding hydrated species, $M(OH)_q(OH_2)_q$, were determined for all 17 metallic cations M with charge q. For the $M(OH)_2$ complexes, the O-M-O bond angle was linear for M = Mg, Co, Cu and Zn whilst Ca, Sr and Ni had bent structures with O-M-O angles of 163°, 140° and 139° respectively. For the $M(OH)_3$ molecules, Al, Sc, Mn and Ga complexes were found to be planar with in-plane bent M-O-H bonds. Y(OH)₃ was also planar but with linear M-O-H bonds. The coordinating O atoms and central M cation for the Fe and Cr hydroxylated complexes were also calculated to be planar but with the terminal H atoms bent out of plane. Unlike the other $M(OH)_3$ complexes, La(OH)₃ was found to have a slightly pyramidal structure with an O-La-O bond angle of 117°. Ti(OH)₄ and Zr(OH)₄ were found to have similar structures with the O atoms coordinated the central metal with a T_d symmetry.

The optimum arrangement of the OH and H_2O groups around the metal centres for the $M(OH)_2(OH_2)_2$ complexes reflected the pseudo-planar/tetrahedral coordination determined for the respective ML₂ bidentate complexes (*i.e.* Ca, Sr, Mg and Zn tetrahedral and Ni, Co and Cu planar) with hydrogen bonding between the H atom of a water group and the O atom of a hydroxyl. For the geometry optimized $M(OH)_3(OH_2)_3$ complexes, the O atoms were found to form an approximate octahedron within the coordination sphere with hydrogen bonding between one of the H atoms of the H₂O groups and O atom of the OH ligands, distorting the O_h coordination symmetry. The *fac* isomer, where the OH/H₂O ligands are mutually *cis*, was determined to be the lowest energy isomeric form. The H₂O groups in Ti(OH)₄(H₂O)₄ and Zr(OH)₄(H₂O)₄ were determined to be positioned between pairs of OH ligands where hydrogen bonding was apparent between a H atom of H₂O and on O atom of one OH group and the O atom of H₂O with the H atom of another OH group.

3.3. Brønsted and Lewis Basicity

Table 2. ΔE_{BB} values (Equation 4) in kJ mol⁻¹ for the 5 ligands considered in this work determined using energies calculated using BP86, PBE and PBE0 functionals and the def2-TZVPP basis set.

	amd	thd	O ^t Bu	dma	CH ₃
BP86	131.27	222.90	110.17	27.81	-87.45
PBE	136.23	227.60	114.12	31.37	-85.86
PBE0	123.67	222.36	101.22	16.96	-89.27

The Brønsted basicities (Equation 4) for the different ligand anions determined using BP86, PBE and PBE0 functionals are shown in Table 2. For all functionals the same trend in Brønsted basicity was found, which is, in order of increasing basicity: thd > amd > O^tBu > dma > CH₃. Only the very reactive CH_3^- ion is predicted to be more basic than OH⁻. Using these values alone, M(CH₃)_q followed by the M(dma)_q complexes are predicted to be the

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most reactive towards hydrolysis and hence towards ALD. $M(O^tBu)_q$ and $M(amd)_q$ complexes are predicted to have similar reactivity, with O^tBu reactions slightly more exothermic, and the $M(thd)_q$ complexes predicted to be the least reactive. Another general trend seen in these data is the up to 14 kJ mol⁻¹ difference in ΔE_{BB} calculated using the 3 functionals with PBE > BP86 > PBE0.



Figure 6. Plot comparing the ΔE_{LB} values determined for M(thd)_q systems (Equation 5b). Energies were calculated using the BP86, PBE and PBE0 functionals and TZVPP basis sets. The ΔE_{LB} values presented here for Mn and Co are for the quintet state and the quartet state respectively and not for the lower energy triplet state for Mn and doublet state for Co found using BP86 and PBE functionals.

The ΔE_{LB} values determined for the M(thd)_q complexes using BP86, PBE and PBE0 functionals are shown in Figure 6. These results are shown as an example of the Lewis basicity calculations of L⁻ relative to OH⁻ in forming ML_q and M(OH)_q complexes (Equations 5a and 5b). Similar to the ΔE_{BB} calculations seen above, differences between the

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 ΔE_{LB} values calculated using the three functionals are apparent, where generally PBE > BP86 > PBE0. There are however some M(thd)_q systems that do not follow this trend where ΔE_{LB} for Ti(thd)₄ and Zr(thd)₄ the BP86 and PBE results are almost identical. These results predict the relative thermodynamics for the hydrolysis of M(thd)_q systems, where more positive ΔE_{LB} values mean that hydrolysis is more favourable and thus mean that thd shows lower Lewis basicity with M. For instance, of the metal cations considered here, thd is least basic in combination with M = Ti(IV) and most basic with Ni(II). ΔE_{LB} does not take into account the formation of the H-L molecule determined using ΔE_{BB} and therefore cannot alone predict the relative reactivity of ML_q complexes with different ligands L. To do this one must determine the energy change during the complete hydrolysis, ΔE_{hyd} (Equations 3a and 3b). The ΔE_{LB} results for the other MLq systems (L = amd, O^tBu, dma and CH₃) are plotted in Figures S.1-S.4 of the Supplementary Material.

3.4. Free-energy change for hydrolysis

One of the main goals of the current work is to assess the reactivity of a metal precursor or fragment with surface hydroxyl groups or water pulses using a simple gas-phase model. We use ΔE_{hyd} as a metric for the elimination of H-L during ALD by reaction between ML_q and OH on the surface (Equations 3a and 3b). Therefore the absolute values of ΔE_{hyd} have limited significance, but the relative values allow one to compare the reactivity of precursors as a function of metal ion, ligand and coordination number. The more negative the ΔE_{hyd} value for gas-phase hydrolysis is, the more likely the ligand is to be eliminated. Conversely, the higher the relative ΔE_{hyd} value, the more stable the ML_q complex is to hydrolysis and the less likely the ligand is to be eliminated from the surface. From the comparison of the ΔE_{BB} and ΔE_{LB} values above (see Table 2 and Figure 6) it has been shown that although the absolute energies calculated using the various DFT functionals are different, the energy trends between the

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different molecular systems are, with a few exceptions, very similar. These comparable energy trends lead to similar conclusions to be made when considering relative reactivity whether comparing BP86, PBE or PBE0 results. We do however note that calculations using the PBE0 hybrid functional have compared very well to experimental results as well as other DFT functionals^{14,35} and will therefore be used in evaluating the ΔE_{hyd} results.



Figure 7. Plot of the DFT calculated energies of hydrolysis (Equation 3b) for a selection of ML_q ALD precursors. Reactivity towards hydrolysis increases with negative ΔE_{hyd} .

As shown in Figure 7, the ΔE_{hyd} values determined for the O-bound ligands (thd and O^tBu) are generally positive or only slightly negative indicating that hydrolysis would be thermodynamically unfavourable. The only negative energy results predicted for the M(thd)_q complexes using the PBE0 functional were those for the hydrolysis of Ti(thd)₄, Zr(thd)₄ and Mn(thd)₃ reflecting ΔE_{LB} (Section 3.3, Figure 6). Of the other thd complexes considered here, there is a small difference in the ΔE_{hyd} values, ranging from 7.6 kJ mol⁻¹ for Ga(III) to 47.1 kJ mol⁻¹ for Ni(II) using molecular energies calculated with PBE0. This relatively small average deviation of 15.5 kJ mol⁻¹ from the mean illustrates that the identity of the metal is a relatively minor factor in the case of thd. For the hydrolysis of the M(O^tBu)_q complexes, the nature of the metal is of even less importance. The difference between the most endothermic complex Zr(IV) ($\Delta E_{hyd} = 5.9 \text{ kJ mol}^{-1}$) and exothermic complex Al(III) ($\Delta E_{hyd} = -7.9 \text{ kJ}$ mol⁻¹) is predicted to be only 13.8 kJ mol⁻¹, with a small average deviation of 3.4 kJ mol⁻¹ about the mean.

The ΔE_{hyd} values of the N-based complexes M(dma)_q and M(amd)_q were found to be fairly similar. All M(amd)_q hydrolysis reactions are predicted to be negative. Ni(dma)₂ and Cu(dma)₂, with ΔE_{hyd} values of 28.0 and 17.7 kJ mol⁻¹ respectively, are calculated to be the only positive M(dma)_q reactivities. This similarity in reactivity between these two amide ligands was not predicted by the Brønsted base model (ΔE_{BB} , Section 3.3, Table 2) and shows the importance of the formation of H-L in the hydrolysis reactions. The M(CH₃)_q hydrolysis reactions are predicted to be the most exothermic, with all ΔE_{hyd} values negative. There is also a greater range of reactivity predicted for M(CH₃)_q complexes with an average deviation of 45.1 kJ mol⁻¹ from the mean for ΔE_{hyd} , probably reflecting M-specific variations in M-C covalency.

Some trends in reactivity can be extracted from the ΔE_{hyd} values for M(dma)_q, M(amd)_q and M(CH₃)_q complexes (and to a lesser extend in the results for M(thd)_q). Hydrolysis of the three group 2 metal complexes is relatively exothermic with a decrease in ΔE_{hyd} going from Mg to Sr. The group 3 metals show comparable reactivity to Ca with very little change going down the group. An increase is seen in ΔE_{hyd} for the transition metals going from group 4 to 9 and 10 where Co or Ni complexes are the least reactive. There is then a decrease in ΔE_{hyd} from group 11 to group 12. Notable outliers in these trends are probably due in the case of Ti

 and Zr to steric hindrance in the coordination spheres and for Cr and Mn, due to the high spin states of these complexes.

3.5.Ligand Exchange

Due to the large number of ligand exchange reactions possible (Mechanism C), a matrix of ΔE_{exc} (Equations 6a and 6b) results was calculated for each of the 5 groups of ligand complexes. An example of these results is shown in Table 3 for M¹(thd)_q ligand exchange with various M²O_x oxide surfaces. From this table it can be seen that the Ti(thd)₄ and Zr(thd)₄ precursors are predicted to exchange ligands with all M₂(OH)_q(OH₂)_q complexes to give M₂(thd)_q and Ti(OH)₄(H₂O)₄ or Zr(OH)₄(H₂O)₄ due to the inherent instability of these high coordination number molecules (Figure 6 and Figure 7). Conversely, some complexes like Co(thd)₂ and Ni(thd)₂ are very stable and are predicted to exhibit limited ligand exchange with other metal oxide surfaces.

These gas-phase results are interpreted for a surface as follows: we predict that adsorption of Ti(thd)₄ on an M^2 oxide surface will be followed rapidly by migration of thd to exposed M^2 cations and incorporation of the Ti cations into the oxide surface or even sub-surface layers. The result is a ternary oxide of Ti and M^2 , covered with a surface of M^2 -thd fragments. On the other hand, Ni(thd)2 is predicted to retain its ligands after adsorption onto a mixed oxide surface.

Tables for the other $ML_q \Delta E_{exc}$ values, where L = amd, O^tBu, dma and CH₃, are tabulated in Tables S.1-S.4 of the Supplementary Material. **Table 3.** ΔE_{exc} for L=thd determined using $pM^1L_q + qM^2(OH)_p(OH_2)_p \rightarrow pM^1(OH)_q(OH_2)_q + qM^2L_p$ (see Equation 6b), where metal M₂ is shown in italics. Values were calculated using molecular energies from PBE0/TZVPP calculations and are in units of kJ mol⁻¹.

M_1/M_2	Mg	Ca	Sr	Sc	Υ	La	Ti	Zr	Cr	Mn	Fe	Co	Ŋ	Cu	Zn	Ν	Ga
Mg	0.0	-12.5	-18.6	-21.5	-20.5	-24.1	-90.2	-43.6	-3.6	-47.9	-16.7	4.0	11.7	-2.2	-9.2	-18.8	-27.8
Ca	12.5	0.0	-6.1	-9.0	-8.0	-11.6	-77.7	-31.0	9.0	-35.4	-4.2	16.5	24.2	10.4	3.3	-6.3	-15.2
Sr	18.6	6.1	0.0	-2.9	-1.9	-5.5	-71.6	-25.0	15.0	-29.3	1.9	22.6	30.3	16.5	9.4	-0.2	-9.2
Sc	21.5	9.0	2.9	0.0	1.0	-2.6	-68.7	-22.1	17.9	-26.4	4.8	25.5	33.2	19.4	12.3	2.7	-6.3
Y	20.5	8.0	1.9	-1.0	0.0	-3.6	-69.7	-23.1	16.9	-27.4	3.8	24.5	32.2	18.4	11.3	1.7	-7.3
La	24.1	11.6	5.5	2.6	3.6	0.0	-66.1	-19.4	20.6	-23.8	7.4	28.1	35.8	22.0	14.9	5.3	-3.6
Ti	90.2	77.7	71.6	68.7	69.7	66.1	0.0	46.7	86.7	42.3	73.5	94.2	101.9	88.1	81.0	71.4	62.5
Zr	43.6	31.0	25.0	22.1	23.1	19.4	-46.7	0.0	40.0	-4.4	26.9	47.6	55.2	41.4	34.4	24.8	15.8
Cr	3.6	-9.0	-15.0	-17.9	-16.9	-20.6	-86.7	-40.0	0.0	-44.4	-13.1	7.6	15.2	1.4	-5.6	-15.2	-24.2
Иn	47.9	35.4	29.3	26.4	27.4	23.8	-42.3	4.4	44.4	0.0	31.2	51.9	59.6	45.8	38.7	29.1	20.2
Fe	16.7	4.2	-1.9	-4.8	-3.8	-7.4	-73.5	-26.9	13.1	-31.2	0.0	20.7	28.4	14.6	7.5	-2.1	-11.1
Co	-4.0	-16.5	-22.6	-25.5	-24.5	-28.1	-94.2	-47.6	-7.6	-51.9	-20.7	0.0	7.7	-6.1	-13.2	-22.8	-31.8
Ni	-11.7	-24.2	-30.3	-33.2	-32.2	-35.8	-101.9	-55.2	-15.2	-59.6	-28.4	-7.7	0.0	-13.8	-20.9	-30.5	-39.5
Си	2.2	-10.4	-16.5	-19.4	-18.4	-22.0	-88.1	-41.4	-1.4	-45.8	-14.6	6.1	13.8	0.0	-7.0	-16.7	-25.6
Zn	9.2	-3.3	-9.4	-12.3	-11.3	-14.9	-81.0	-34.4	5.6	-38.7	-7.5	13.2	20.9	7.0	0.0	9.6-	-18.6
Η	18.8	6.3	0.2	-2.7	-1.7	-5.3	-71.4	-24.8	15.2	-29.1	2.1	22.8	30.5	16.7	9.6	0.0	-9.0
Ga	27.8	15.2	9.2	6.3	7.3	3.6	-62.5	-15.8	24.2	-20.2	11.1	31.8	39.5	25.6	18.6	9.0	0.0

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4. DFT model assessment of experimental compositions

In the following, reaction mechanisms for the growth of ternary oxides predicted the DFT hydrolysis model are assessed through comparison with experimental results found in the literature. Reaction mechanisms A and B (Section 1) had been previously proposed in ref. 8 from experimental stoichiometries of $M^1_x M^2_{1-x} O_y$ grown by alternating $M^1(thd)_q$ and $M^2(thd)_p$ metal precursors in thermal ALD with ozone oxidation. These mechanisms (parts A-E) are re-examined in light of the theoretical results presented above. Experimental results from other literature sources using different ML_q precursors are also examined in parts F-H.

A. LaMnO_x:





Figure 8. La stoichiometry of LaMnO_x ternary oxide as a function of La ALD pulse ratio⁸. D(La) is the molar fraction of La deposited relative to the total amount of metal cations and P(La) is the fraction of La(thd)₃ + O₃ sub-cycles relative to the total number of sub-cycles. The dashed line is a plot of the predicted composition using Mechanism A (Equation 1) and the solid line is a fit of Mechanism B (Equation 2) to the experimental data. A previous fit from the literature is represented using the dotted line³⁹.

For the deposition of LaMnO_x, from La(thd)₃, Mn(thd)₃ and O₃, a good linear fit had been previously⁸ found between both mechanisms and the experimental data (Figure 8). This would indicate that either no H-thd is eliminated before the oxidation pulse following the purely combustion mechanism A or, if elimination of H-thd occurs, thd ligands are transferred from one metal to the other via Mechanism C. One possible scenario for Mechanism C is the transfer of

 thd from Mn to La, forming more stable metal-ligand bonds. This mechanism is supported from experiments where the deposition of Mn(thd)₃ without any other metal precursor is not self limiting but when combined with La(thd)₃ or Ca(thd)₂ provides good ALD growth³⁹. Indeed, the PBE0 DFT calculations predict that Mn-thd bonds are more labile towards hydrolysis than the La-thd bonds (Figure 7). Hence the exchange model with M¹ = Mn and M² = La yields a value of $\Delta E_{\text{exc}} = -23.8 \text{ kJ mol}^{-1}$ (Table 3). This supports the proposed transfer of ligands from manganese to lanthanum during deposition, so that the surface always consists of La-thd at the end of either M(thd)_q pulse.





Figure 9. Ca stoichiometry of the CaMnO_x ternary oxide as a function of Ca ALD pulse ratio⁸. A previous fit of the experimental data³⁹ is obscured by the fit of Mechanism B.

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A good fit was found between both mechanisms A and B and the experimental data⁸ for CaMnO_x with a non-linear curve due to the difference in cationic charge: q_{Mn} =3, q_{Ca} =2 (Figure 9). Like the deposition of LaMnO_x, it is again difficult to distinguish between the two mechanisms. From the DFT calculations Mn(thd)₃ is predicted to be less stable in the presence of OH groups than Ca(thd)₃ and H-thd elimination is more likely to occur for the Mn complex. However, the computed ΔE_{exc} value of -35.4 kJ mol⁻¹ for M¹ = Mn and M² = Ca may again confirm the argument for Mechanism C, where ligand transfer occurs at the surface from Mn to Ca.





Figure 10. Co stoichiometry of the $CoFeO_x$ ternary oxide as a function of Co ALD pulse ratio⁸.

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Again for the deposition of CoFeO_x, it is difficult to distinguish between the two mechanisms, with very similar curves fitting the experimental data (Figure 10). A possible process allowing Mechanism B to occur would be a similar reactivity of Co(thd)₂ and Fe(thd)₃ to H-thd elimination. Comparing the energies calculated for the hydrolysis of these two complexes, ΔE_{hyd} for Co(thd)₂ is 20.7 kJ mol⁻¹ more positive than that of Fe(thd)₃. Transfer of thd is thus predicted to take place from the more reactive Fe(thd)₃ complex to the less reactive Co(thd)₂ complex with a ΔE_{exc} value of -20.7 kJ mol⁻¹. This means that, if Mechanism B were in operation, Fe(thd)₃ would be more susceptible to H-thd elimination. However both ΔE_{hyd} values are more positive than those for the group 2 and 3 metal complexes that are predicted to exhibit H-thd elimination. Thus, these gas-phase results may indicate that no Brønsted elimination can occur and that only the combustion type Mechanism A is in operation.

D. LaFeO_x:



Figure 11. La stoichiometry of the LaFeO_x ternary oxide as a function of La ALD pulse ratio⁸. The dotted line represents a previous fit to the experimental data⁴⁰ that is almost identical to the fit using Mechanism B.

Unlike the previous $M^1M^2O_x$ ALD reactions presented above, there is a clear difference between the fits of the two mechanisms for the deposition of LaFeO_x. Mechanism A predicts a linear relationship between the ratio of La:Fe deposited and the pulse ratio (dashed line Figure 11). This is in contrast to the near perfect fit of Mechanism B with the experimental data⁸ which indicates that the deposition of La is favoured over Fe (solid line Figure 11). If Mechanism B is correct, La is more reactive to Brønsted elimination of H-thd than Fe. This is supported by the DFT calculated energies where ΔE_{hyd} for La(thd)₃ is lower than that of Fe(thd)₃ but only by 7.4 kJ mol⁻¹ per ligand.



Figure 12. Sr stoichiometry of the SrFeO_x ternary oxide as a function of Sr ALD pulse ratio⁸. The dotted line represents a previous fit to the experimental data⁴⁰.

Mechanism A for the deposition of SrFeO_x underestimates the ratio of Sr deposited to that of Fe (dashed line Figure 12). In contrast, mechanism B has an excellent fit to the experimental data (solid line Figure 12), similar to that seen for LaFeO_x. The fit of Mechanism B to the experimental data appears to indicate elimination of H-thd from the deposition of Sr-thd using up all available OH protons, in contrast to no elimination of H-thd from the deposition of Fe-thd. Looking at the results from the DFT calculations, Brønsted elimination is marginally favoured for Sr(thd)₂ by 1.9 kJ mol⁻¹ per ligand but not to the same extent as seen in the ALD experiments. Indeed, the difference between the ΔE_{hyd} values for Sr and Fe is not as great as that

predicted for La and Fe, which experimentally has a slightly lower ratio of LaO_x in the ALD $LaFeO_x$ thin films than the ratio for SrO_x in $SrFeO_x$.

F. YScO_x:

The stoichiometries of thin films of YScO_x produced using different pulse ratios of Y(thd)₃ and Sc(thd)₃ and O₃ as an oxygen source have been presented by Myllymäki *et al.*⁴¹. A linear relationship was found between the ratio of metal cations M in the resulting ternary oxide and the ratio of precursor pulses. This would suggest that either Mechanism A is in operation, where no H-thd elimination occurs, or that if Mechanism B is working then Y(thd)₃ and Sc(thd)₃ must have very similar reactivities towards surface OH groups. In the theoretical hydrolysis calculations, Y(thd)₃ and Sc(thd)₃ are seen to have similar ΔE_{hyd} values of 14.9 and 13.9 kJ mol⁻¹, (Figure 7) indicative of the similar reactivity of these group 3 metals.

G. SrTiO_x

SrTiO_x thin films were produced by Kosola *et al.*⁴² using Sr(thd)₂ and O₃ for the deposition of SrO and Ti(ⁱPrO)₄ and H₂O for the deposition of TiO₂. The process showed a near linear transfer of pulse ratio to stoichiometry, with for example a Sr:Ti pulse ratio of 1:1 resulting in Sr:Ti atomic ratio in the solid of 0.9. On the other hand, Mechanism A predicts that Sr with a smaller charge of +2 would deposit more metal than Ti with a charge of +4 (*cf.* Figure 1) which according to experiment is not the case. Mechanism B predicts a near linear relationship if the Ti precursor is far more reactive towards Brønsted elimination at the surface and almost no ligands are eliminated from the Sr precursor. In such a scenario, it does not affect the argument whether H₂O or O₃ is used as co-reagent prior to the unreactive Sr pulse. Assuming that the reactivity of the iso-propanolate ⁱPrO ligands is similar to that of the tert-butanolate ^tBuO ligands, then the

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computed ΔE_{hyd} value for Sr(thd)₂ is 14.8 kJ mol⁻¹ greater than that for Ti(^tBuO)₄ (Figure 7) and Ti(ⁱPrO)₄ is predicted to be more reactive than Sr(thd)₂. This supports Mechanism B and explains the experimental finding, although the difference in ΔE_{hyd} is not as great as might be expected from the experimental data.

H. ZnAlO_x

ZnAlO_x ternary oxide films were deposited by Elam and George⁴³ using $Zn(CH_2CH_3)_2$ (DEZ) and Al(CH₃)₃ (TMA) as metal precursors and H₂O as an oxygen source (not ozone). Our models for ternary oxide stoichiometry have been formulated for the case of well-defined hydroxyl coverage resulting from ozone as oxygen source, and so cannot be applied quantitatively in this case. However, the computed energetics can help to interpret the experimental results. At DEZ pulse ratios below 75%, much less ZnO is deposited compared to AlO_{3/2} than predicted using the "rule of mixtures" (i.e. Mechanism A, where the ratio of metal cations in the resulting thin film is dependent only on their charge). Above 75% DEZ pulses, a sharp increase in the relative amount of ZnO deposited is evident, approaching the ratios predicted for Mechanism A. ΔE_{hvd} for Al(CH₃)₃ is computed to be much more exothermic than $Zn(CH_3)_2$ (-139.5 kJ mol⁻¹ versus -53.6 kJ mol⁻¹ respectively. Figure 7), and it is likely that there is a similar difference between Al(CH₃)₃ and Zn(CH₂CH₃)₂. This large difference in ΔE_{hyd} may indicate that surface reactions as proposed by Mechanism B are occurring when pulsing <75% DEZ, with more CH₄ being eliminated from Al(CH₃)₃ than CH₃CH₃ from DEZ. However, this interpretation Mechanism B alone is complicated by an initial loss of mass after the Al precursor pulse of TMA⁴³. The authors proposed a mechanism for "zinc etching" where the Al(CH₃)₃ precursor exchanges methyl ligands with the ZnO surface producing the highly volatile Zn(CH₃)₂ molecule. This Mechanism C type reaction is predicted in the current work to be thermodynamically favourable

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with $\Delta E_{\text{exc}} = -82.6 \text{ kJ mol}^{-1}$ (where M¹ = Al and M² =Zn, *c.f.* Table S.4 in the Supporting Information). Mechanism C would allow the production, followed by desorption, of significant amounts of Zn(CH₃)₂ (which has a low boiling point of 46 °C) especially when a ZnO-covered surface is exposed to repeated doses of Al(CH₃)₃ e.g. for TEZ pulse ratios below 75%.

4.1. Summary of DFT model assessment of experimental compositions

To summarize the above, no inconsistencies are found between the computed energetic and the experimental data, although in some cases the magnitude of the observed effect exceeds what the computed value would suggest. In some cases (e.g. LaMnOx, CaMnOx), Mechanism B can account for the measured stoichiometries as long as ligands are transferred (*i.e.* Mechanism C) to the most acidic metal cations on the surface. Nevertheless, we are still unable to definitively distinguish between Mechanism A and Mechanism B in a few cases (e.g. LaFeOx, SrFeOx,). This inability of the DFT models to definitively determine the mechanism of growth for some of the M¹M²O_x solids, in particular for the MFeO_x ternary oxides, highlights some of the problems of using a gas-phase model to represent a reaction at a solid surface. One important factor not considered in the gas-phase models is the acidity of the surface hydroxyl protons. A more acidic M¹-OH covered surface would allow more elimination of HL from an incoming M^2L_a precursor, permitting more M² ions to be deposited. For example, a relatively more acidic Fe-OH surface would increase the reactivity of precursors towards Brønsted elimination and could explain the increased growth rates of La and Sr ions seen in the deposition of LaFeO_x and SrFeO_x. These differences in surface M-OH reactivities have been previously found to influence the relative deposition rates of Hf and Ti in the ALD of HfTiO ternary oxide thin films⁴⁴. A more accurate first principles model would entail calculating the energies for solid slabs of the different oxide surfaces with the precursors placed on the surface¹⁵. These larger models are of course more time

consuming and difficult to perform, precluding us from the scale of survey presented here (17 metal cations combined with 5 different ligands).

5. Conclusion

Our DFT calculations confirm that all metal complexes containing the β -diketonate thd ligand and alkoxide O^tBu are fairly unreactive with respect to elimination of H-thd during ALD. These ligands are removed from the surface through combustion by the co-reagent (*e.g.* ozone or oxygen plasma). The identity of the metal has in most cases a relatively minor effect. However, Ti(thd)₄, Zr(thd)₄ and Mn(thd)₃ have been predicted to be the most reactive of the thd systems studied here towards H-thd elimination.

Amide and amidinate complexes were predicted to show similar reactivity towards hydroxyl groups, with the hydrolysis of most $M(amd)_q$ and $M(dma)_q$ complexes being exothermic. Unsurprisingly, the most reactive complexes towards hydrolysis were determined to be those containing CH₃ ligands. The nature of the central metal cation was found to be more important for the reactivity of $M(amd)_q$, $M(dma)_q$ and $M(CH_3)_q$ systems than for $M(thd)_q$ and $M(O^tBu)_q$.

By modeling the exchange of ligands between different metal cations, the theoretical results have been used to corroborate a ligand transfer mechanism proposed for the deposition of LaMnO_x and CaMnO_x and thus explain the experimentally observed composition of the ternary oxides. Both Co(thd)₂ and Fe(thd)₃ are predicted to be fairly unreactive in ALD, which is consistent with the experimental stoichiometries of CoFeO_x films. La(thd)₃ and Sr(thd)₂ were calculated to be only slightly more reactive to hydrolysis than Fe(thd)₃, and so we suggest that a more acidic FeOH surface may explain the LaO_x and SrO_x rich films of LaFeO_x and SrFeO_x respectively that are observed experimentally. The DFT models developed in this work for metal precursor reactions with growing oxide surfaces successfully predicted the experimental metal cation ratios for many ALD ternary oxides and may be applied to other similar systems, for example, higher order quaternary oxides.

6. Acknowledgements

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7. Supporting Information

Some additional figures and tables are provided in the Supporting Information, including plots of ΔE_{LB} values and tables of ΔE_{exc} values for 4 ML_q systems where L = amd, O^tBu, dma and CH₃. This material is available free of charge via the Internet at <u>http://pubs.acs.org</u>.

8. Author Information

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Author Contributions

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

9. References

2 3 4		(1)	Mitchell, R. H. Perovskites: Modern and Ancient; Almaz Press: Ontario, Canada,
5 6 7	2002.		
8 9 10		(2)	George, S. M. Chem. Rev. 2009, 110, 111.
11 12 13		(3)	Kim, H.; Lee, HBR.; Maeng, W. J. Thin Solid Films 2009, 517, 2563.
14 15 16		(4)	Leskelä, M.; Ritala, M. Angew. Chem., Int. Ed. 2003, 42, 5548.
18 19 20		(5)	Niinistö, L.; Nieminen, M.; Päiväsaari, J.; Niinistö, J.; Putkonen, M. Phys. Status
21	Solidi	A 2004	, 201, 1443.
22 23 24 25		(6)	Knez, M.; Nielsch, K.; Niinistö, L. Adv. Mater. 2007, 19, 3425.
26 27 28		(7)	Seim, H.; Molsa, H.; Nieminen, M.; Fjellvag, H.; Niinisto, L. J. Mater. Chem.
29 30	1997, ′	7, 449.	
31 32 33 34		(8)	Elliott, S. D.; Nilsen, O. ECS Trans. 2011, 41, 175.
35 36 37		(9)	Elliott, S. D.; Scarel, G.; Wiemer, C.; Fanciulli, M.; Pavia, G. Chem. Mater. 2006,
38 39 40	18, 370	64.	
41 42 43		(10)	Murray, C.; Dozova, N.; McCaffrey, J. G.; FitzGerald, S.; Shafizadeh, N.; Crepin,
44 45 46	C. Phy	vs. Chen	n. Chem. Phys. 2010, 12, 10406.
40 47 48 49		(11)	Bühl, M.; Kabrede, H. J. Chem. Theory Comput. 2006, 2, 1282.
50 51		(12)	Waller, M. P.; Braun, H.; Hojdis, N.; Bühl, M. J. Chem. Theory Comput. 2007, 3,
52 53 54	2234.		
55 56			
57 58			37

(13) Bühl, M.; Reimann, C.; Pantazis, D. A.; Bredow, T.; Neese, F. J. Chem. Theory Comput. 2008, 4, 1449.

- (14) Johnson, E. R.; Becke, A. D. Can. J. Chem. 2009, 87, 1369.
- (15) Elliott, S. D. Semicond. Sci. Technol. 2012, 27, 074008.

- (16) Holme, T. P.; Prinz, F. B. J. Phys. Chem. A 2007, 111, 8147.
- (17) Elliott, S. D. Surf. Coat. Technol. 2007, 201, 9076.
- (18) Kazadojev, I.; Otway, D. J.; Elliott, S. D. Chem. Vap. Deposition 2012, In Press.
- (19) Elliott, S. D.; Greer, J. C. J. Mater. Chem. 2004, 14, 3246.
- (20) Klejna, S.; Elliott, S. D. J. Phys. Chem. C 2011, 116, 643.
- (21) TURBOMOLE V6.1 2009, a development of University of Karlsruhe and Forschungszentrum Karlsruhe GmbH, 1989-2007, TURBOMOLE GmbH, since 2007

(22) Ahlrichs, R.; Bär, M.; Häser, M.; Horn, H.; Kölmel, C. Chem. Phys. Lett. 1989, 162, 165.

- (23) Becke, A. D. Phys. Rev. A 1988, 38, 3098.
- (24) Perdew, J. P. Phys. Rev. B 1986, 33, 8822.
- (25) Perdew, J. P.; Burke, K.; Ernzerhof, M. Phys. Rev. Lett. 1996, 77, 3865.
- (26) Perdew, J. P.; Burke, K.; Ernzerhof, M. Phys. Rev. Lett. 1997, 78, 1396.

ACS Paragon Plus Environment

(27)	
(27)	Elenkorn, K.; Treutier, O.; Onm, H.; Haser, M.; Anifiens, R. Chem. Phys. Lett.
1995, 242, 65	52.
(28)	Eichkorn, K.; Weigend, F.; Treutler, O.; Ahlrichs, R. Theor. Chem. Acc. 1997,
97, 119.	
(29)	Sierka, M.; Hogekamp, A.; Ahlrichs, R. J. Chem. Phys. 2003, 118, 9136.
(30)	Adamo, C. J. Chem. Phys. 1999, 110, 6158.
(31)	Weigend, F.; Ahlrichs, R. Phys. Chem. Chem. Phys. 2005, 7, 3297.
(32)	Kaupp, M.; Schleyer, P. v. R.; Stoll, H.; Preuss, H. J. Chem. Phys. 1991, 94,
1360.	
(33)	Andrae, D.; Häußermann, U.; Dolg, M.; Stoll, H.; Preuß, H. Theor. Chem. Acc.
1990, 77, 123	3.
(34)	Dolg M · Stoll H · Savin A · Preuss H Theor Chem Acc 1989 75 173
(54)	Dolg, M., Stoll, H., Savill, A., Heuss, H. Hiebl. Chem. Acc. 1767, 75, 175.
(35)	Li, S.; Hennigan, J. M.; Dixon, D. A.; Peterson, K. A. J. Phys. Chem. A 2009,
113, 7861.	
(36)	Rabuck, A. D.; Scuseria, G. E. J. Chem. Phys. 1999, 110, 695.

(37) Delchev, V. B.; Mikosch, H.; Nikolov, G. S. Monatsh. Chem. 2001, 132, 339.

(38) Lozada-Garcia, R. R.; Ceponkus, J.; Chin, W.; Chevalier, M.; Crépin, C. Chem. Phys. Lett. 2011, 504, 142. (39) Nilsen, O.; Rauwel, E.; Fjellvag, H.; Kjekshus, A. J. Mater. Chem. 2007, 17, 1466.

(40) Lie, M.; Nilsen, O.; Fjellvag, H.; Kjekshus, A. Dalton Trans. 2009, 481.

(41) Myllymaki, P.; Nieminen, M.; Niinisto, J.; Putkonen, M.; Kukli, K.; Niinisto, L. J.Mater. Chem. 2006, 16, 563.

(42) Kosola, A.; Putkonen, M.; Johansson, L.-S.; Niinistö, L. Appl. Surf. Sci. 2003,211, 102.

(43) Elam, J. W.; George, S. M. Chem. Mater. 2003, 15, 1020.

(44) Popovici, M.; Delabie, A.; Van Elshocht, S.; Clima, S.; Pourtois, G.; Nyns, L.;Tomida, K.; Menou, N.; Opsomer, K.; Swerts, J.; Detavernier, C.; Wouters, D.; Kittl, J. A. J.Electrochem. Soc. 2009, 156, G145.



