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Article

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Three-Dimensionally Ordered Hierarchically Porous Tin Dioxide Inverse Opals and Immobilization of Palladium Nanoparticles for Catalytic Applications

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Abstract

A high surface area 3D ordered SnO_2 inverted opal with walls composed of interconnected nanocrystals, is reported using a facile approach with a tin acetate. The hierarchically porous structure exhibits porosity on multiple lengths scales (cm down to nm). The thickness of the IO wall structure comprising nanocrystals of the oxide can be tuned by multiple infilling of the precursor. Using highly monodisperse Pd nanoparticles, we show how the SnO_2 IO can

be functionalized with immobilized Pd NP assemblies. We show that the Pd NP size dispersion is controlled by utilizing weak ligand-metal interactions and strong metal-oxide interactions for the immobilization step. The resulting SnO_2-Pd IOs were investigated X-ray photoelectron spectroscopy indicating electronic interactions between the Pd and SnO_2 and alterations to NP surface chemistry. Pd NPs assembled with excellent dispersion on the ordered SnO_2 IOs show superior catalytic performance for liquid phase chemical synthesis via Suzuki coupling reactions, and allow easy removal of the catalyst substrate post reaction. Higher mass electrocatalytic activity is also demonstrated for formic acid oxidation, superior to commercial Pd/C catalysts, which is shown to be due to better access to the catalytically active sites on SnO_2-Pd IOs. The high surface area interconnected phase-pure SnO_2 IO, with programmable porosity forms a functional material for catalytic applications.

Introduction

Semiconductor oxides with a 3-dimensional ordered macroporous (3-DOM), or inverse opal (IO) morphologies have gained much attention in recent years due to their unique structural features such as high surface area and programmable, ordered porosity[1]. Tin dioxide (SnO_2) is a wide band gap semiconductor ($E_g \sim 3.6$ eV) with applications such as high capacity lithium battery anodes[2-4], oxidation catalysts[5], sensors[6, 7] and optoelectronic devices[8, 9]. Porous SnO_2 structures demonstrate superior sensing ability due to improved diffusion and higher surface area[10, 11] and the periodic macroporous structure of SnO_2 IOs have shown to be a near ideal morphology for CO sensing applications[12]. The use of hierarchical macroporous materials such as IOs are attractive materials for heterogeneous catalysts.[13] Hierarchical porosity of 3-dimensionally ordered systems are beneficial for heterogeneous catalytic applications as they provide small pores for nanoparticle immobilization and the presence of

larger pore networks reduces mass transport limitations.[14] Incorporation of mesoporosity into catalytic supports can lead to improved metal dispersion, greater active site accessibility and reaction rate enhancements.[15, 16]

In addition to optimal structural properties of the support, size reduction of the metal nanoparticles is critical to their (electro)catalytic activity but controlling nanoparticle size and dispersion on solid supports remains challenging[17]. Supported catalysts prepared by impregnation or precipitation methods typically exhibit broad diameter distributions resulting from the annealing/reduction treatments required during synthesis. Solid-state synthesis through pyrolysis of organometallic precursors facilitates a single-step synthesis but controlling monodispersity is limited[18]. It is well recognized that excellent diameter control can be obtained through solution synthesis of colloidal nanoparticles.[19] Achieving high catalyst loadings of colloidal nanoparticles often require pre-immobilization treatment of the support material such as oxidative or thermal activation or surface functionalization to introduce a chemical linker group[20, 21]. A more significant drawback with colloidal nanoparticles is the presence of a capping layer, which is usually undesirable for catalytic applications and often needs to be removed by annealing treatments. Annealing, however generally results in loss of diameter control due to particle agglomeration.[22, 23] Similarly, oxidative removal such as ozone treatment can lead changes in the surface chemistry of the metal.[24] Lopez-Sanchez *et al*.[25] demonstrated that up to 25% of polyvinyl alcohol (PVA) capping ligands on Au an AuPd NPs can be removed by solvent extraction thereby conserving the NP size and morphology.

In this paper we report the synthesis of 3D ordered, multi length scale porous SnO_2 IOs using tin acetate precursors, and the ability to functionalize the hierarchically porous network with Pd nanoparticles (NPs) with functional catalytic and electro catalytic behaviour. The use of the acetate as precursor avoids the need of corrosive $SnCl_2$ and moisture sensitive

tin alkoxide precursors, typically used for IO synthesis.[11, 26] Furthermore, preparation of SnO_2 IO by the acetate produces structures with a hierarchical porosity making them ideal as catalyst support materials for liquids and gases.[27] We also report a facile method for the deposition of Pd nanoparticles with controlled diameter and size distribution. Immobilization of Pd nanoparticles onto these surfaces is then demonstrated without the need for post-annealing treatments to remove the capping ligands, which can be largely removed by solvent extraction in acetic acid. We demonstrate the viability of these SnO_2-Pd nanocomposites for catalytic applications. Firstly, the SnO_2-Pd IOs are demonstrated as functional catalytic coatings for chemical synthesis with Suzuki coupling reactions, which are important carbon-carbon bond forming reactions used extensively in pharmaceutical synthesis. The activity of SnO_2 Pd IOs as anodes for electrocatalytic oxidation of formic acid is also investigated. The use of oxide support materials for fuel cells are promising replacements for carbon supports which are susceptible to corrosion[26]. SnO_2 is reported to display remarkably enhanced CO tolerance leading to lowering poisoning effects and also exhibit promotional effects due electronic interactions between SnO_2 and the active metal.[28, 29] The use of SnO_2 IOs as supports for Pd-catalyzed formic acid oxidation (FAO) has not be reported thus far, and they have potential for improved catalytic performance through increased surface area and hierarchical porosity.

2. Experimental

Polystyrene (PS) spheres (0.5 μm) were purchased from Polysciences Europe, Eppelheim, Germany. All other chemicals were purchased from Sigma-Aldrich.

2.1 Preparation of SnO_2 IO using Sn(II) and Sn(IV) acetates. Opal templates were purified by centrifugation of 5 ml of PS-spheres. The supernatant water was removed and the sample

was re-dispersed. Deposition of the spheres onto stainless steel substrates was carried out by electrophetic deposition (EPD).[30] A 2.5 wt% solution of spheres in EtOH was prepared and NH_4OH solution was added dropwise to obtain a pH of 10. Two stainless steel substrates, placed 5 mm apart were used as the anode and cathode. To immobilize the templates, 2.5 V was applied for 1 h, followed by 150 V for 10-20 s. This procedure gave rise to templates of ~200 μm thick. Infiltration of the polymer templates with SnO_2 was carried out using saturated solutions of Sn(II) or Sn(IV) acetate in acetic acid prepared by dissolving the maximum amount of the precursor salt in ~2 ml of acetic acid at 100 °C. The solution was cooled to room temperature under stirring before drop casing onto the polymer sphere template. Infiltrated templates were dried at 40°C for at least 2 hours in a drying oven. In case of multiple infiltrations, the infiltration procedure was repeated as often as necessary once the samples were dried and cooled to RT. Subsequently, the templates were dried at 40 °C for at least two hours until complete evaporation of the acetic acid after every infiltration step. To remove the PS spheres, the substrates were calcined in air at a temperature of 600 °C for 5 h. To study the effect of annealing rate on the IO structure, two annealing rates of 0.5 °C min^{-1} and 5 °C min^{-1} were used.

2.2 Synthesis and immobilization of Pd NPs. Pd NPs were synthesized using modified literature procedures[31, 32]. The synthesis was carried out under Ar using standard Schlenk techniques. In a typical synthesis, 75 mg of $Pd(acac)_2$ was dissolved in 15 ml of oleylamine (OA) and heated to 60 °C under magnetic stirring. 1 ml of triphenylphosphine (TOP) was injected into the solution. In a separate flask, 300 mg of borane *t*-butylamine was dissolved in ~3 ml OA under Ar. This amine complex solution was then injected into the Pd precursor solution and heated to 90 °C. The reaction was aged for 3 h. After the solution cooled to room temperature, ethanol was added to precipitate the NPs, which were collected by

centrifuge. The precipitate was redispered in 10 ml of hexane and a further 30 ml of EtOH was added to precipitate the NPs. The NPs were sonicated and collected by centrifugation at 8000 rpm. This purification procedure was repeated three times and the purified NPs were redispersed in hexane. After purification the typical mass of NPs obtained was 55-60 mg. Preparation of other NPs used in this study were prepared by literature methods and are described in the Supporting Information[33].

SnO_2 IOs were dried under vacuum overnight and then immersed in the NP solution. The mixture was briefly sonicated and agitated overnight. The SnO_2 IOs were removed from the NP solution without rinsing and dried in a vacuum oven for ~4 h at 60 °C followed by drying in air at 120 °C overnight. To clean the nanocomposite of residual NPs the IOs were immersed in fresh hexane solution and agitated for 30 min, followed by immersion into EtOH. The substrates were further rinsed in fresh EtOH and finally hexane, before being dried under vacuum at 60 °C.

2.3 Materials Characterization. Scanning Electron Microscopy (SEM) characterization was performed using a Hitachi S-4800 SEM cold field emission apparatus or a SU-70 SEM hot field emission apparatus. Transmission Electron Microscopy (TEM) and energy dispersive x-ray (EDX) spectroscopy was carried out using a JEM2010-TEM equip with an Oxford X-Max 80 detector and Inca analysis software. Fourier-transform Infrared (FTIR) was performed on a Nicolet 6700 FTIR. X-ray Photoelectron Spectroscopy (XPS) was acquired using a KRATOS AXIS 165 monochromatized X-ray photoelectron spectrometer equipped with an Al Kα (hv = 1486.6 eV) X-ray source. Spectra were collected at a take-off angle of 90° and all spectra were reference to the C 1s peak at 284.6 eV. The Sn 3d core level spectra were fit to Gaussian (70%)-Lorentzian (30%) profiles. The Pd 3d core level spectrum of the

unsupported NPs was fit to Gaussian (90%)-Lorentzian(10%) profile for elemental Pd centred at 334.1 eV with FWHM = 0.9. After immobilization of the Pd NPs onto SnO_2 IOs, the Pd 3d core level required different fitting parameters in order to obtain a satisfactory fit. The elemental Pd peak was centred at 335.5 eV and fit to a broader peak (FWHM = 1.2). The oxide associated Pd^{2+} and Pd^{4+} peaks were fit with FWHM = 1.5. The oxide thickness was calculated by the substrate overlayer model using Eqn 1:[34]

$$d = \lambda_{ox} Cos\theta Ln\left(\frac{N_{Pd}\lambda_{Pd} I_{ox}}{N_{ox}\lambda_{ox} I_{Pd}} + 1\right) \quad (1)$$

where λ is the effective photoelectron attenuation length, I_{Pd} and I_{Ox} are the photoelectron intensities of the Pd and PdO_x species, respectively, N is the number of atoms per unit volume and θ is the detection angle. The density of Pd and PdO was taken to be 12.08 g cm^{-3} and 8.3 g cm^{-3} respectively. The density of PdO_2 was estimated by structural data taking the unit cell volume to be 62.32 $Å^3$.[35] The attenuation length (λ_{AL}) of Pd 3d electrons moving through the metallic core and oxide overlayer was estimated using the CS2 semi-empirical method developed by Cumpson and Seah,[36] and described by Eqn 2:

$$\lambda_{AL} = 0.31a^{3/2}\left\{\frac{E}{z^{0.45}\left[Ln\left(\frac{E}{27}\right)+3\right]} + 4\right\} \quad (2)$$

where E is the kinetic energy (eV), a is the lattice constant or monolayer thickness and z is the average atomic number.

2.4 Catalytic performance of SnO_2-Pd nanocomposites.

Suzuki Cross Coupling: In a typical reaction, 2 mmol of aryl halide, 2.2 mmol of phenylboronic acid and 2 equivalents of K_2CO_3 was added to 30 ml of EtOH:H_2O (2:1) solvent. A stainless steel substrate coated SnO_2 Pd IO (typical dimension 0.8 cm × 3 cm) was

immersed vertically into the solution and clamped at the top. The solution was stirred during the reaction which was carried out in air. Reactions were monitored by TLC. After the reaction, the EtOH was removed by rotary evaporation and the product was extracted with DCM (× 3). The organic layer was washed with water (× 3) and dried over $MgSO_4$. The turn over number (TON) and turn over frequencies (TOF) of the catalytic thin films was calculated based on the mass of SnO_2 IO. Where the TON = mol of aryl halide converted/ mol Pd and the TOF = mol converted arly halide/ mol Pd h^{-1}. The Pd concentration was determined by EDX analysis.

Electrocatalytic studies: The electrochemical properties of the supported catalysts was carried out in a three-electrode cell at room temperature using a BioLogic VSP potentiostat. A Pt wire mesh, Standard Calomel Electrode (SCE) and glassy carbon electrode were used as the counter, reference and working electrodes, respectively. To prepare the working electrode, the Pd SnO_2 IO catalysts were deposited on the electrode surface followed by a 10 µl droplet of 0.1 wt % Nafion solution. Linear sweep (LSV) and cyclic voltammograms (CVs) of the catalysts were obtained in Ar saturated 0.1 M perchloric ($HClO_4$) and 2 M formic acid (HCOOH) solution. The potential was scanned from -0.2 V to +1.0 V at a scan rate of 10 mV s^{-1}. Unless otherwise stated, all potentials are referenced with respect to SCE. The mass current was normalized to A/g Pd by dividing the measured electrode density over the mass Pd in the catalysts.[37]

Results and Discussion

Synthesis of hierarchically porous SnO_2 IOs using tin acetate precursors

Figure 1 shows SEM and TEM images of SnO_2 IOs synthesized with Sn(IV) acetate precursors under different treatment conditions. Figure 1 (a) and (d) display SEM images of

the IOs obtained by double (successive) infiltration and annealing rate of 0.5 °C min^{-1} and 5 °C min^{-1}, respectively. In this synthetic method, we find that faster annealing rates decrease the wall thickness from ~60 nm at 0.5 °C min to ~45 nm at 5 °C min^{-1}, as shown in the inset in figure 1 (a) and (d). The morphology of the walls annealed at 5 °C min^{-1} also display rougher surfaces compared to those annealed at 0.5 °C min^{-1}, which is shown in the TEM image in figure 1 (b) and (e). Specifically, the seemingly rough walls of the 3-dimensionally ordered porous network comprises an assembly of nanoscale crystallites of SnO_2 with an average size of ~5 nm. All such crystals are single crystals and they are fused together at grain boundaries; this assembly of SnO_2 nanocrystals is itself porous. The detail of this fused structures can be seen in TEM image in figure 1 (c). The high resolution TEM in figure 1 (b) inset shows a particle with an interplaner d-spacing of 0.26 nm, which can be indexed to SnO_2 (101).[38]

The number of precursor infiltrations influence the structures of the IOs with single infiltration often resulted in an incomplete IO network. The wall thickness of the IOs after a single infiltration were slightly thinner (~50 nm) compared to IOs after a double infiltration due to a lower number of crystallites making up the thickness of the wall, but displayed similar nanocrystallite size (5 nm), shown in figure 1 (f)-(g). Three infiltrations of Sn acetates dissolved in acetic acid does completely infill in the PS template, however results in an irregular IO structure due to excess SnO_2 (Supporting Information, Figure S1).

SnO_2 IOs derived from the Sn(II) precursors have similar structure to those of Sn(IV) precursors in terms of the definite size of the nanocrystals and their arrangement to form porous walls of similar thickness (Supporting Information, Figure S1 (a)-(e)). SnO_2 IOs prepared from Sn(II) precursors also crystallize similarly to Sn(IV) acetates, with faster annealing rates resulting in thinner walls and smaller nanocrystalline size. The average nanocrystal size is smaller (<5 nm) and of higher density.

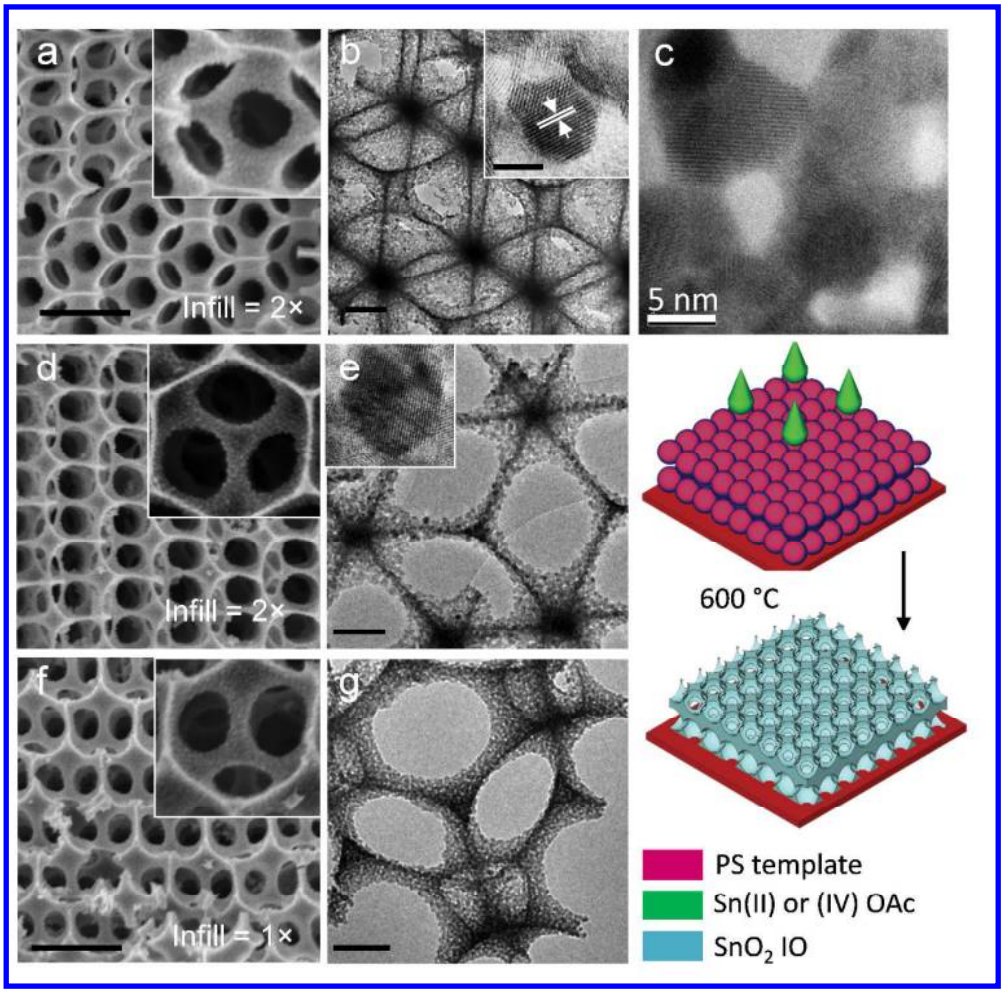
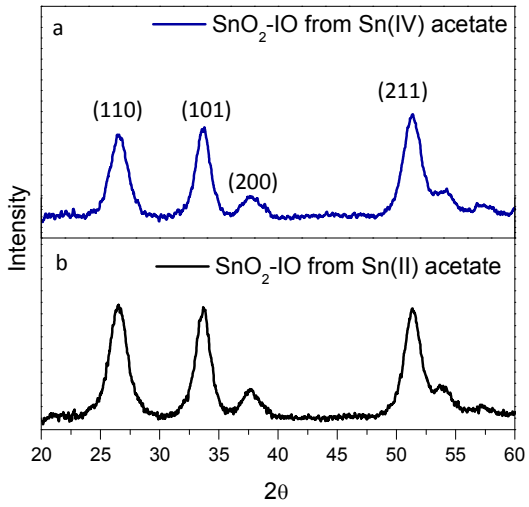


Figure 1. (a)-(b) SnO_2 IOs formed using double infiltration of a Sn(IV) acetate, calcined at 0.5 °C min. (c) HRTEM image of the fused arrangement of SnO_2 nanocrystals comprising the mesoporous walls of the IO. (d)-(e) SnO_2 IOs formed using double infiltration but crystallized at 5 °C min. A schematic to the right shows the arrangement of the original (111)-oriented fcc lattice of polymer sphere templates prepared by EPD and a definition of the hierarchical porosity for the porous 3D IOs. (f)-(g) SnO_2 IOs formed using single infiltration 0.5 °C min^{-1} using Sn(IV) acetate. Scale bar in main figure (a), (d) and (f) are 500 nm and the scale bar in in insets are 100 nm. Scale bar in figure (b), (e) and (g) are 100 nm.

The XRD characterization of crystal phase and purity of the IOs is shown in figure 2 (a)-(b). X-ray diffraction patterns showing peaks that are indexed to (110), (101), (200) and (211) reflections of tetragonal cassiterite phase SnO_2 (space group P42/*mnm*). Furthermore, the XRD pattern exhibits no additional peaks, indicating the absence of crystalline or heterophase impurities such as *e.g.* SnO. XPS analysis further confirms the presence of phase pure SnO_2. The survey spectra of the IOs (Supporting Information Figure S2) show only the presence of Sn, oxygen and small amounts of remnant carbon. The Sn 3d core level spectra of the IOs shown in figure 2 (c)-(d) consist of a doublet located a binding energies of 486.1 eV and 494.5 eV, which can be assigned to the Sn $3d_{5/2}$ and Sn $3d_{3/2}$ of Sn^{4+} in SnO_2, respectively.[39]



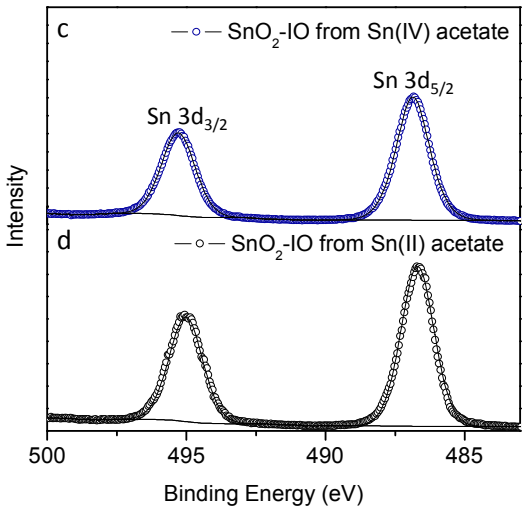


Figure 2. (a)-(b) XRD pattern of SnO_2-IO prepared from Sn(IV) acetate precursor and Sn(II) acetate precursor, respectively. The patterns can be indexed to JCPDS 00-041-1445. (c)-(d) Sn 3d core level XPS of SnO_2-IO from Sn(IV) acetate precursor and Sn(II) acetate precursor, respectively.

Palladium Immobilized SnO_2 IOs

Figure 3 illustrates the formation of the Pd NPs with an average diameter of 3.2 nm (standard dev. = 0.95) determined by size distribution analysis (Supporting Information, Figure S3). The XRD patterns (Supporting Information, Figure S4) for the Pd NPs shows the presence of 4 peaks at 40 °, 46°, 68 and 80°, indexed to the (111), (200), (220) and (311) diffraction peaks of Pd, respectively. The FTIR spectra of neat OA and OA-capped Pd NPs are (Supporting Information Figure S5) are similar, except for shifts in the absorption peaks due to the interactions of OA with the Pd NPs.[40] The band at 3343 cm^{-1} in the IR spectrum of OA assigned to the N–H stretching mode, is red-shifted to 3390 cm^{-1} in OA-capped Pd NPs due to absorption of amine groups onto the Pd surface. The N–H bending band at 1570 cm^{-1} in neat OA is also shifted to 1566 cm^{-1} in the Pd NPs spectrum. No evidence for the

presence of phosphine ligands is observed in the IR spectrum of the OA-Pd NPs and XPS analysis shown in figure 3 further confirms the absence of phosphorous in the survey spectra (P $3p_{3/2}$ ~130 eV) indicating that phosphine ligands are not bound to the Pd surface. The presence of TOP during NP synthesis was found to be important for improving the monodispersity of the Pd NP size and shape consistency, as shown in the TEM of the NPs in figure 4(a). The use of dual co-ordinating ligands during NP synthesis scan improve monodispersity due to their different binding affinity with the metal NP.[41] Nanoparticles synthesized without addition of TOP led to a mixture of small diameter (< 5 nm) nanoparticles and larger aggregates and illustrated by TEM (Supporting Information, Figure S6).

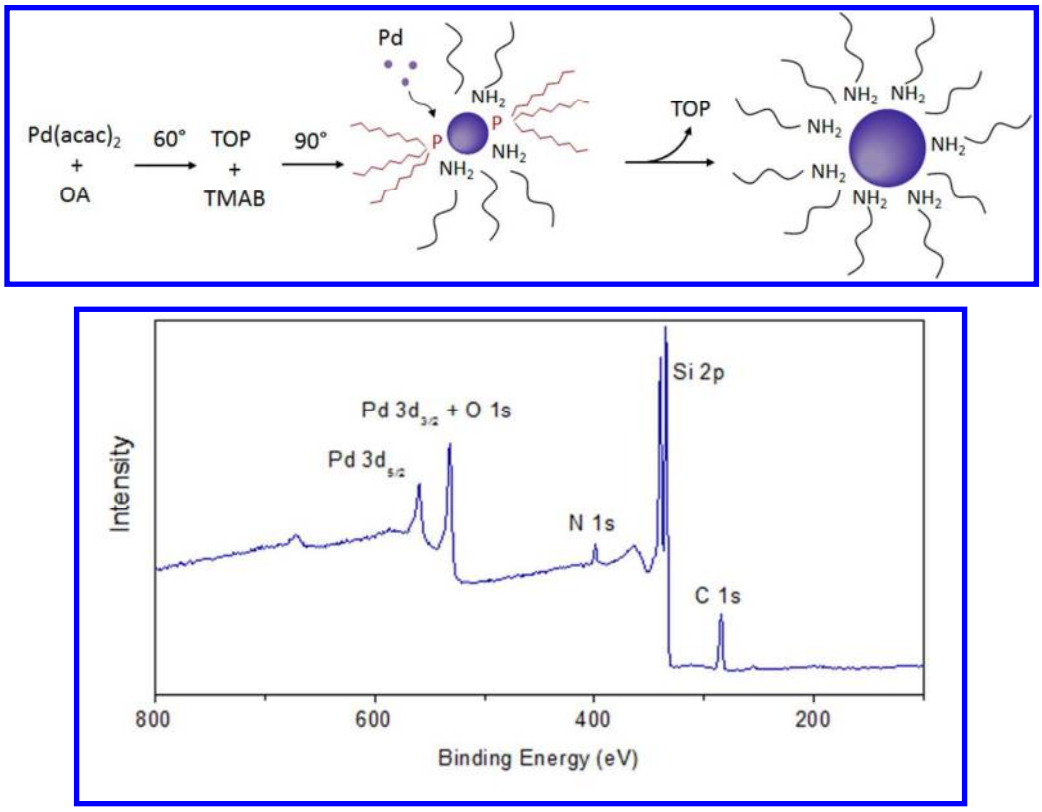


Figure 3. Schematic illustrating the steps in Pd NP formation in the presence of TOP and OA ligands. XPS survey spectra (100-800 eV) showing the absence of P capping ligands on the NPs. Note that the Si 2p peak arises from the Si wafer used to deposit the NPs for analysis.

Figure 4(b)-(c) shows highly dispersed Pd NPs immobilized on SnO_2 IO. High resolution TEM in figure 4(d) displays nanocrystals with an interplanar d-spacing of 0.33 nm and 0.26 nm, which can be indexed to SnO_2 (110) and SnO_2 (101), respectively [38]. The lattice fringes with a d-spacing of 0.23 nm correspond to those of Pd(111)[42]. From figure 4(d), the Pd NPs are found to be dispersed primarily around the SnO_2 nanocrystals; no dominant facet relationships between the crystalline Pd and SnO_2 at the NP attachment sites is found and the arrangement is random in distribution. The density of Pd NPs appears highest at either side of the walls (figure 4(b)), but we find that the distribution is relatively consistent spatially, *i.e.* that the density of Pd particles that form a randomly distributed assembly either side of the IO walls is similar to the density of Pd NPs covering the SnO_2 nanocrystals comprising the walls. TEM analysis of several regions consistently shows that the Pd NPs are generally well disperse and do not aggregate or overlap after immobilization in a low dielectric constant non-polar solvent (Supporting Information, Figure S7-S8). Furthermore, the Pd NPs do not entirely cover the available surface area of the IO structure for the concentration of NPs used in the dispersion. EDX elemental mapping of the Pd SnO_2 IO is shown in figure 5 (a)-(c) confirms the presence of Pd NPs immobilized on the SnO_2 IOs, and quantitatively confirm the Pd NP dispersion and coverage of the hierarchically porous SnO_2 IO structure, which does not suffer extensive cracking even for a thickness of 100 μm. The Pd concentration determined by quantitative EDX analysis is 10.4 wt%. Cross-sectional SEM was used to assess the distribution profile of the Pd NPs through the SnO_2 IO in a thick, multi-layered IO. Figure 5 (d) shows EDX line-scans for Pd and Sn through a 200 μm thick IO matrix. The presence of Pd NPs is clearly identified through the IO network, and follows the distribution of Sn from the SnO_2, i.e. while there is less Pd than SnO_2 present in the IO, the distribution within the interaction volume probed by the electron beam is

consistent with the variation in the density of the matrix, indicating relatively uniform solution based dispersion (discussed below) through a very think mesoporous matrix.

BET analysis of the SnO_2 IOs show a specific surface area of 35 m^2 g^{-1}, and a pore volume of 0.12 cm g^{-1}, consistent with reports of TiO_2 IOs.[43] A pore diameter of 13.5 nm indicates mesoporosity in the network.[44] After Pd NP deposition, the surface area increases to 48 m^2 g^{-1} and a small decrease in the pore diameter (12 nm) was observed due to presence of the Pd NPs within the IO network.

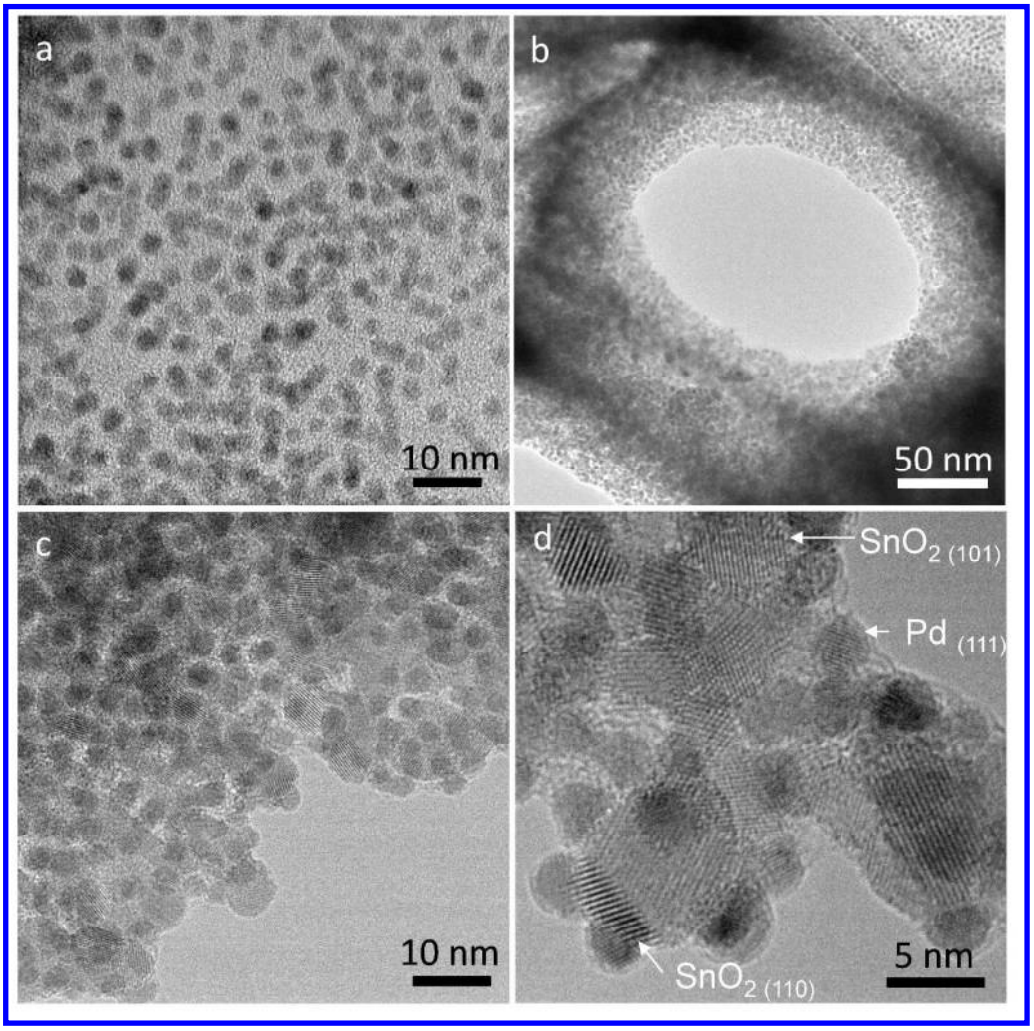


Figure 4. TEM images of (a) as-synthesized OA-capped Pd nanoparticles. (b)-(d) SnO_2 IOs after Pd nanoparticle immobilization.

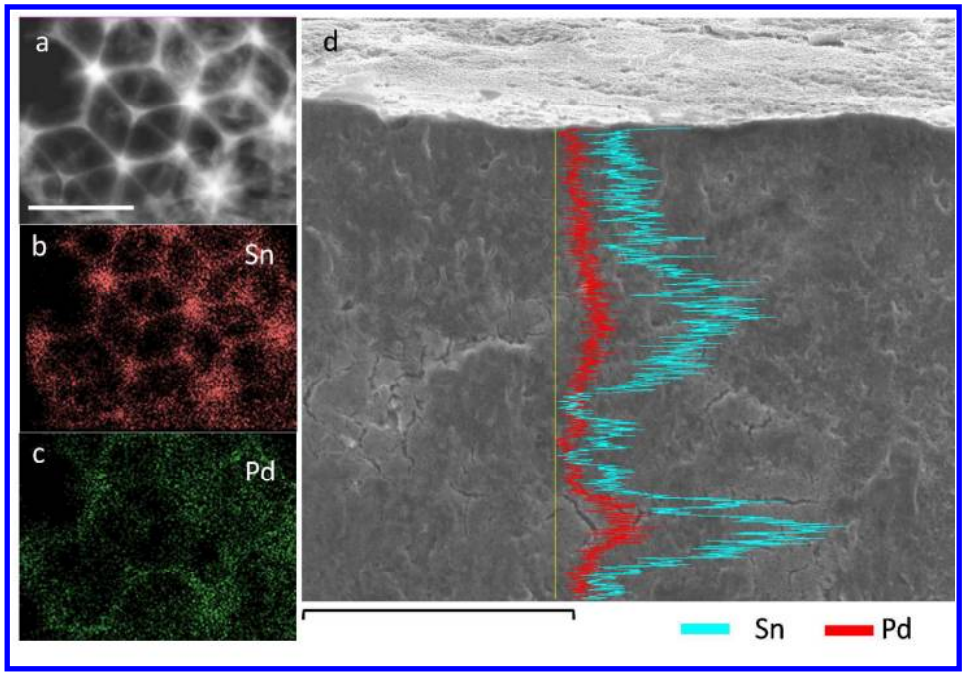


Figure 5. (a) STEM image of Pd deposited on SnO_2 IO at its (111) orientation. Scale bar = 500 nm. Corresponding elemental maps are shown for (b) Sn and (c) Pd. (d) Cross sectional SEM image of SnO_2-Pd IOs and EDX overlaying line scans for Sn and Pd. Scale bar in (d) is 100 μm.

Achieving good dispersion of colloidal nanoparticles on porous supports can be hindered by poor wetting, unfavourable ligand-support interactions and high surface tension solvents limiting penetration into porous supports. Gupta and co-workers[45] reported high loadings of colloidal FePt nanoparticles onto SiO_2 surfaces, which they attribute to the short range specific interactions between the metal and support, facilitated by the presence of weakly binding oleic acid and oleylamine ligands. This immobilization method relies on the repulsive interactions between the hydrophobic ligands and hydrophilic oxide being much weaker than those between the highly polarizable metal NP and oxide surface. To investigate the influence the capping ligand in the Pd-SnO_2 system, Pd NPs capped with polyvinylpyrrolidone (PVP) ligands in aqueous ethanol solutions were immobilized into

SnO_2 IOs using the same procedure as OA-Pd NPs (Supporting Information Figure S9). EDX analysis showed negligible coverage by Pd of the SnO_2 IO supports. While OA is a weakly binding ligand readily exchanged by organic and inorganic ligands[46, 47], PVP is a significantly bulkier capping ligand which screens the interaction between the NP and the surface, consequently, the NPs do not immobilize on the surface or are more easily removed during rinsing. Stucky *et al.*[48] demonstrated that a homogenous dispersion of hydrophobic dodecanethiol capped Au NPs in aprotic solvents can be achieved on oxide supports by this approach. While their procedure allowed for homogenous dispersion, calcination was still required to 'lock in' the dispersed NPs. Post-annealing treatments, which are typically > 300 °C, are not required for the immobilization of Pd NPs onto the SnO_2 IO, which were subject to heating at much lower temperature of 120 °C. The inherent roughness of the nanostructured SnO_2 surface, porosity and the comparable size of the Pd NPs and SnO_2 crystallites, as well as weak ligand interactions all contribute to the uniform dispersion and adhesion of the Pd NPs.

To further understand the immobilization of Pd on SnO_2 IOs, the nature of the Pd – SnO_2 interaction was probed by XPS analysis. Chemical compositional analysis of the as-synthesized and SnO_2 immobilized Pd NPs was determined by XPS, shown in figure 6. The Pd 3d core level spectrum of OA-Pd NPs show a doublet at a binding energy of 335.1 eV and 340.4 eV, which can be assigned to metallic Pd $3d_{5/2}$ and $3d_{3/2}$, respectively. There is also the presence of a shoulder peak at a binding energy of 336.3 eV which can be assigned to PdO[49]. Analysis of Pd 3d spectra for Pd NPs deposited on SnO_2 IOs also show the presence of Pd(0) and Pd(II) species, with an additional peak at a binding energy of 338 eV, associated with highly oxidized Pd^{4+} species such as PdO_2[50, 51], however, the presence of Pd^{4+} hydroxide species have also been reported in the range of 337.9-338.5 eV [52]. PdO_2 is unstable in its anhydrous form but has been observed on Pd NPs stabilized by a surrounding oxide matrix.

[51, 53, 54] Formation of PdO_2 may occur due to reaction with hydroxyl groups on the SnO_2 surface. Immobilization of the Pd NPs onto SnO_2 IOs gave rise to peak broadening, which has been associated with surface oxidation.[55] The fraction of Pd^0 and Pd^{2+} species of the as-synthesized OA-Pd NPs determined from the integrated Pd 3d peak intensities, are 0.74 and 0.26, respectively. After NP immobilization, the fraction of Pd species is 0.73 (Pd^0), 0.16 (Pd^{2+}) and 0.11 (Pd^{4+}) suggesting that the formation of Pd^{4+} is mainly at the expense of Pd^{2+} oxide species. Calculation of the oxide thickness from the combined Pd^{2+} and Pd^{4+} intensities, using Eqns 1 and 2, yields an effective oxide thickness of 0.91 nm averaged for all NPs. The curvature of nanoparticles however leads to an overestimate in the oxide thickness determined by XPS intensities and a geometric correction is required to account for the enhanced sensitivity of the oxide overlayer.[56] Using the correction factor for Pd nanoparticles derived by Van Devener *et al.*[57] the estimated oxide thickness of Pd NPs on SnO_2 IOs is 0.38 nm, essentially one monolayer of oxide.

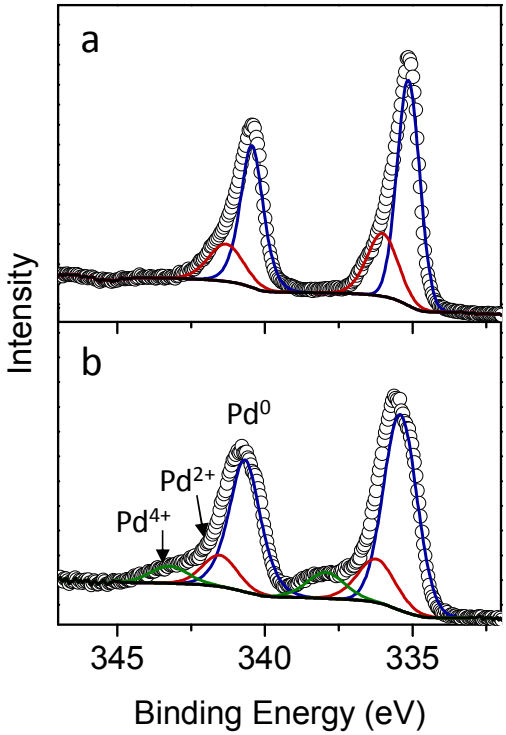


Figure 6. Pd 3d core-level spectra of (a) as-prepared OA-Pd NPs and (b) after immobilization onto SnO_2 IOs.

Analysis of the Sn 3d core level spectrum after Pd NP immobilization (Supporting Information Figure S10) displays a binding energy shift of +0.4 eV. This peak shift may be attributed to an interaction between Pd and SnO_2, manifest in shifts in core level binding energy. [58, 59] The observed peak shifts may also correspond to changes in the Sn^{n+} oxidation state .[60, 61] The electronic interaction between the SnO_2 and Pd NPs as well formation of oxide as detected by XPS may also contribute to anchoring Pd NPs to the surface.

In addition to changes in the Pd surface chemistry, immobilization of the NPs was also found to alter the chemistry of the oleylamine capping ligands. Figure 7 compares the N 1s core-level spectra of the as prepared OA-Pd NPs and is characterized by a single peak located at a binding energy of 399.8 eV. This energy is characteristic of alkylamine adsorption on Pd and is consistent with the FTIR analysis indicating that OA ligands absorb with the $–NH_2$ group intact.[62] After immobilization onto SnO_2 IOs, the N 1s spectrum shows the amine-associated peak at 399.8 eV and a second, lower intensity peak at a binding energy of 398.2 eV, which is associated with unsaturated N species.[63] The presence of imine (-N=C), nitriles (N≡C) and keteneimine (C=C=N) groups have been reported due to irradiation damage during XPS collection[63]. If this peak (398.2 eV) was induced by X-ray irradiation, its presence would also be expected in the pure OA-Pd NPs, which is not the case. Formation of nitriles and imines from aliphatic amines has been reported in the synthesis of Ag and Fe NPs.[40, 64] The weak binding strength of OA may result in migration of the amine from the NP surface to the surrounding support leading to reactions with the oxide surface.

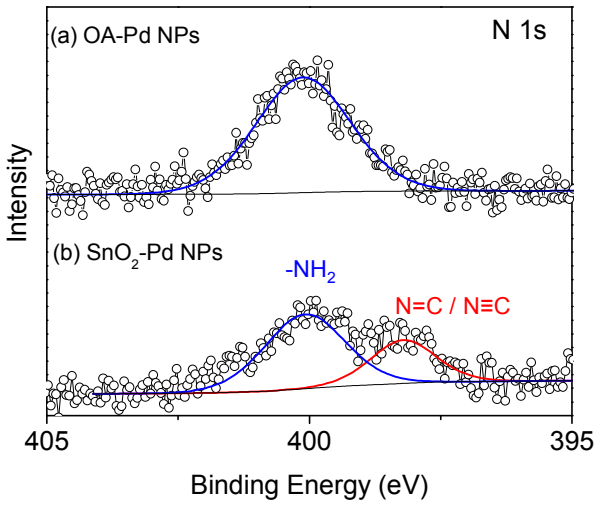


Figure 7. N 1s core level spectra of (a) OA-Pd NPs and (b) Pd-SnO_2 IOs.

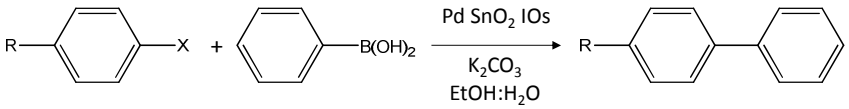
A further benefit of using a weakly binding capping ligand such as OA is the potential to remove the ligand by treatment with acetic acid.[31] XPS analysis of the SnO_2-Pd IOs refluxed in acetic acidic for 12 h show a 42% decrease in the Pd:N ratio indicating considerable loss of the capping ligands (Supporting Information, Figure S11). It is however worth noting that the use of OA as a capping ligand do not significantly impact on the catalytic performance.[41]

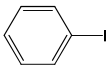
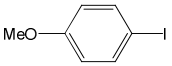
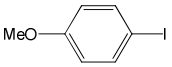
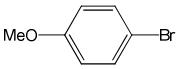
Catalytic Performance of SnO_2-Pd IOs

Heterogeneous catalysts with hierarchical structures are particularly beneficial for liquid phase reactions where mass transfer limitations can be problematic with the use of micro/nanoporous supports. The performance of SnO_2-Pd IOs as catalytic coatings were evaluated in Suzuki coupling reactions as shown in Table 1. The SnO_2-Pd IOs show excellent reactivity in cross coupling of aryl iodides and bromides with phenylboronic acid

obtaining yields > 90% under air and at room temperature. Furthermore, the catalyst does not require any separation procedure after the reaction expect form removal of the stainless steel substrate. The reusability of the catalyst was also evaluated in cross coupling of 4-methoxyiodobenzene and phenylboronic acid at room temperature. The catalyst showed excellent recyclability over 3 cycles showing no loss in catalytic activity with yields of 95% ± 3% (determined by GC using internal standard). Ellis *et al.*[65] found high reactivity associated with the edge and corner atoms of NPs in heterogeneously catalysed Suzuki coupling reactions, thus the presence of small diameter NPs as are immobilized on the IOs contain a higher proportion of these defect sites compared to larger diameter NPs. Furthermore, increased accessibility of these active sites may be facilitated by the hierarchical porous SnO_2 network. A major drawback of the use of unsupported colloidal NPs is that their solubility is influence by the nature of the capping ligands, therefore the unsupported NPs readily aggregate when added to the aqueous reaction due to the hydrophobic OA ligands (Supporting Information, Figure S12). To investigate the effect of the porosity, a non-porous SnO_2 film was prepared and subjected to the same Pd NP deposition. SEM of the SnO_2 (Supporting Information Figure S13) shows no obvious porosity in the film and EDX analysis (Figure S14) after Pd deposition reveals the presence of NPs on the surface of the film. The non-porous SnO_2 film gave a reasonably good yield (84%, determined by GC), although lower compared to the SnO_2 IOs, on the first cycle. The non-porous SnO_2 films showed poor recyclability and decreased catalytic performance after the first cycle, with the yield dropping to 43% in the second cycle. Suzuki coupling conditions have been reported to give rise to Pd leaching, thus the poor reusability is likely to be associated with loss of the active metal.[66] Furthermore, immobilization of NPs onto oxide support materials have been shown to increase instability of and leaching resistance of the catalyst.[67]

Table 1. Suzuki cross coupling reactions catalyzed by SnO_2 Pd IOs.



Aryl Halide	Time (h)	Temperature °C	Yield % [a]	TON	TOF^{tot} (h^{-1}) [b]
	3 h	20	93	11923	3974
	4 h	20	92	11795	2949
	1 h	80	95	12179	12179
	4 h	20	85	10897	2724

Conditions: aryl halide (2 mmol), phenylboronic acid (2.2 mmol), K_2CO_3 (2 equivalents), EtOH:H_2O (2:1). [a] Isolated yield. [b] The TOF calculated from the total mass of Pd estimated from the mass of SnO_2 IO thin film, assuming 10 wt% Pd.

The electrocatalytic activity of the SnO_2-Pd IOs was also investigated for formic acid oxidation (FAO). Figure 8 shows the linear sweep voltammograms (LSVs) for SnO_2 Pd IOs and commercially available Pd/C in 2 M HCOOH + 0.1 M $HClO_4$. The SnO_2 Pd IOs display a well-defined anodic peak at 0.09 V, attributed to oxidation of formic acid catalyzed by the Pd NPs.[28] Furthermore, the onset potential of formic acid oxidation is ~ 180 mV lower in SnO_2 Pd IOs compared to commercial Pd/C which occurs at 0.2 V. The inset in figure 8 shows CVs for SnO_2 IOs in the absence of Pd NPs; it is clear that no oxidation of formic acid occurs and that SnO_2 support does not results in side reactions in the anodic response of the

polarization curve. It is well recognized that FAO is dependent on the nanoparticle size for a given density and dispersion.[68] TEM analysis of the commercial Pd/C shows Pd NPs with a diameter of 2-5 nm, however the presence of larger Pd aggregates is this composite is also evident (Supporting Information Figure S15). Better diameter control of the Pd NPs (consistency in shape, size and dispersion) supported on SnO_2 IOs facilitated by the assembly around the walls of the hierarchical morphology of the SnO_2 IO results in a greater proportion of electrochemically active Pd sites, while electrical contact required for electrochemical activity is maintained by their immobilization on to the interconnected SnO_2 nanocrystals.

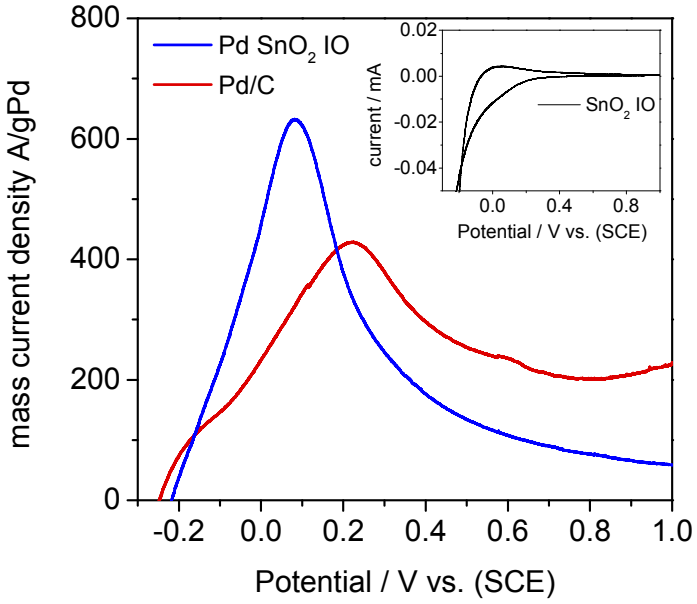


Figure 8. Linear sweep voltammogram of formic acid oxidation using Pd SnO_2 IOs. The inset shows the CV curve for blank SnO_2 IO before Pd NP immobilization.

Conclusions

We have demonstrated that a high surface area of interconnected phase-pure SnO_2 nanocrystals, formed as a continuous inverted opal network, is possible using tin acetate. The

structure exhibits hierarchical porosity on multiple lengths scales (cm down to nm). The thickness of the IO wall structure comprising nanocrystals of the oxide can be tuned by multiple infilling of the precursor. Uniform dispersion of highly uniform and dispersed assemblies of Pd NPs on the SnO_2 IO support is achieved by exploiting the relatively weak binding affinity between the OA capping ligand and the Pd NPs dispersed in a low surface tension solvent. The result is an ordered Pd NP assembly around the walls of the IO structure, accessible to liquids and gases. The catalytic performance of the SnO_2-Pd IOs was assessed in Suzuki cross coupling reactions and found to display excellent activity at room temperature with no loss of catalytic activity after 3 cycles. The SnO_2-Pd IOs also show potential for fuel cell applications and in electrocatalytic oxidation reactions, with improved formic acid oxidation demonstrated with the SnO_2-Pd IOs. The enhanced catalytic performance is attributed to the uniform dispersion of small diameter Pd NPs and the hierarchical porosity throughout the IO network, facilitating access to the catalytically active sites. The programmable porous order and the porosity possible at multiple length scale augers well for a range of catalytic process involving liquids and gases, and scope exists for the immobilization of a wide range of high index facet catalytic materials is programmable, ordered porous host structures. The high density of available catalytic sites in an order porous network has not been previously demonstrated for supported catalysts, which represent the primary catalyst in industrial oxidation and catalytic reactions.

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Supporting Information

This information is available free of charge via the Internet at http://pubs.acs.org/.

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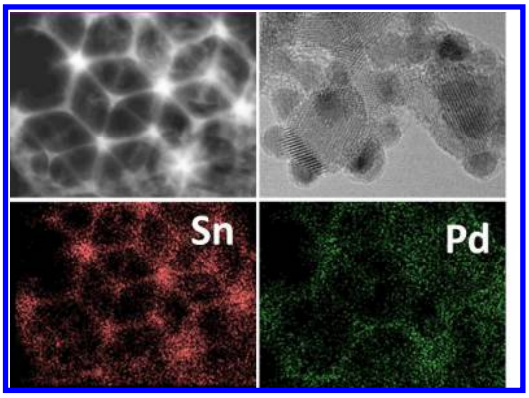
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