

Ultrafast dynamics of type-II GaSb/GaAs quantum dots

K. Komolibus, T. Piwonski, K. Gradkowski, C. J. Reyner, B. Liang, G. Huyet, D. L. Huffaker, and J. Houlihan

Citation: *Appl. Phys. Lett.* **106**, 031106 (2015); doi: 10.1063/1.4906106

View online: <http://dx.doi.org/10.1063/1.4906106>

View Table of Contents: <http://aip.scitation.org/toc/apl/106/3>

Published by the [American Institute of Physics](#)

Articles you may be interested in

[Hybrid type-I InAs/GaAs and type-II GaSb/GaAs quantum dot structure with enhanced photoluminescence](#)
Applied Physics Letters **106**, 103104 (2015); 10.1063/1.4914895

[Thermal emission in type-II GaSb/GaAs quantum dots and prospects for intermediate band solar energy conversion](#)
Journal of Applied Physics **111**, 074514 (2012); 10.1063/1.3703467

[Complex emission dynamics of type-II GaSb/GaAs quantum dots](#)
Applied Physics Letters **95**, 061102 (2009); 10.1063/1.3202419

[Band parameters for III–V compound semiconductors and their alloys](#)
Journal of Applied Physics **89**, 5815 (2001); 10.1063/1.1368156

[Improved room-temperature luminescence of core-shell InGaAs/GaAs nanopillars via lattice-matched passivation](#)
Applied Physics Letters **108**, 061104 (2016); 10.1063/1.4941435

[Radiative recombination in type-II GaSb/GaAs quantum dots](#)
Applied Physics Letters **67**, 656 (1998); 10.1063/1.115193



Ultrafast dynamics of type-II GaSb/GaAs quantum dots

K. Komolibus,^{1,2} T. Piwonski,^{1,2} K. Gradkowski,^{1,2} C. J. Reyner,³ B. Liang,³ G. Huyet,^{1,2,4} D. L. Huffaker,³ and J. Houlihan⁵

¹Centre for Advanced Photonics and Process Analysis, Cork Institute of Technology, Cork, Ireland

²Tyndall National Institute, University College Cork, Lee Maltings, Cork, Ireland

³Department of Electrical Engineering and California NanoSystems Institute, University of California - Los Angeles, Los Angeles, California 90095, USA

⁴National Research University of Information Technologies, Mechanics and Optics, Saint Petersburg, Russia

⁵School of Science, Waterford Institute of Technology, Waterford, Ireland

(Received 18 November 2014; accepted 6 January 2015; published online 20 January 2015)

In this paper, room temperature two-colour pump-probe spectroscopy is employed to study ultrafast carrier dynamics in type-II GaSb/GaAs quantum dots. Our results demonstrate a strong dependency of carrier capture/escape processes on applied reverse bias voltage, probing wavelength and number of injected carriers. The extracted timescales as a function of both forward and reverse bias may provide important information for the design of efficient solar cells and quantum dot memories based on this material. The first few picoseconds of the dynamics reveal a complex behaviour with an interesting feature, which does not appear in devices based on type-I materials, and hence is linked to the unique carrier capture/escape processes possible in type-II structures. © 2015 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4906106>]

Photonic materials based on type-II (staggered) band alignment such as GaSb/GaAs quantum dots (QDs) have proved promising in applications such as charge-based memories¹ and solar cells.² Many-particle interactions play a very important role in the dynamics of semiconductor nanostructures exhibiting such a band alignment.³ In these structures, spatial separation between unbound electrons and confined holes leads to greatly modified emission properties when compared with their type-I counterparts. Previous research on type-II GaSb/GaAs QDs utilizing time-resolved photoluminescence spectroscopy revealed a strong blue-shift of the emission energy accompanied by a significant reduction of the radiative lifetime at elevated carrier densities.⁴ However, a detailed analysis of the room temperature ultrafast carrier dynamics of these structures, which is essential for the improvement of the device performance, is still missing. Previously, two-colour pump-probe spectroscopy has proven to be very useful in revealing the carrier relaxation and escape processes of epitaxial and colloidal type-I QD structures.^{5,6} In this paper, we apply this technique to type-II based semiconductor optical amplifiers (SOAs) for the first time in both forward and reverse bias (RB) mode and discover an interesting ultrafast feature that seems to be unique to type-II structures.

The structure comprises five layers of GaSb QDs grown on n-type GaAs substrate by molecular beam epitaxy. These QDs were formed by depositing 2.5 monolayers (MLs) of GaSb with a growth rate of 0.12 ML s⁻¹, followed by 25 nm thick GaAs spacer. This multi-layered structure is surrounded by 1.25 μm Al_{0.3}Ga_{0.7}As barriers and finally covered with 100 nm of GaAs. A piece of wafer was processed into 1 mm long SOA with 4 μm wide angled ridge. In order to provide good thermal conductivity, the SOA was then attached to a copper heat sink in a manner that allows light coupling to/from both facets.

Fig. 1 shows the amplified spontaneous emission (ASE) peak energy and a ratio of the total emission intensity to

applied current (termed the efficiency) as functions of increasing bias (in blue and red, respectively). Both quantities were examined over a wide range of currents exceeding 3 orders of magnitude and resulted in a total observed blue-shift of 200 meV. It is clear from the graph that there are two regimes of behaviour. For low values of injection current (<10 mA), the ASE exhibits a large blue-shift and the efficiency increases. For larger currents (>10 mA), there is a reduction of both the ASE efficiency and the rate of change of ASE peak energy per unit current.

It has been commonly accepted that the giant blue-shift of the emission energy that occurs with increasing excitation is a fingerprint of staggered band alignment.⁷ The reasons attributed to this behaviour are threefold and occur simultaneously after injection of the carriers: band bending,⁸ capacitive charging,^{3,9} and state filling.^{3,4} In addition, the increase in efficiency for low currents can be understood in terms of

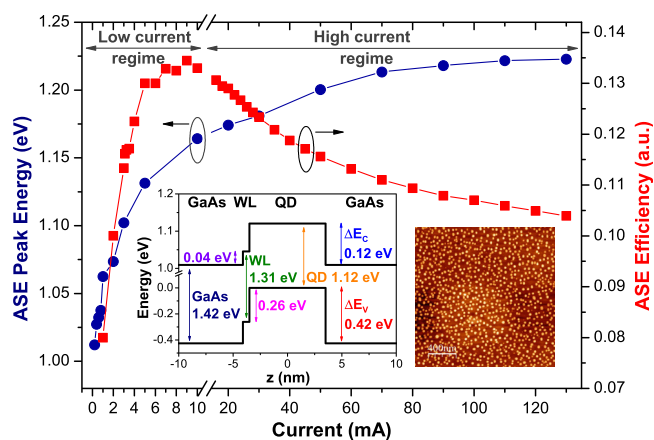


FIG. 1. Energy shift of ASE (blue curve) and ratio of ASE intensity to current (red curve) as a function of increasing current. Inset: calculated band alignment and AFM picture of examined QD structure.

an increased overlap between electron and hole wavefunctions which result in a peak efficiency at 10 mA.¹⁰

For higher currents, the reduction in blue-shift can be attributed to the lower confinement of higher energy hole wavefunctions which result in a lower charge density and hence a smaller contribution to the Coulomb potential,^{4,11} while the reduction in ASE efficiency can be attributed to the lower transfer rate of holes to the QD due to increased repulsion.¹² Some research groups have associated the complete saturation of blueshift to the possible formation of a confining potential for electrons in the QD due to band bending effects and therefore type-I transitions at high carrier densities.^{11,13} In order to gain better insight into the band structure of our sample, we employed an eight band $k \cdot p$ formalism to calculate the band alignment and potentials for both carrier species. The dimensions of the QD (32 nm \times 32 nm base size and 6.5 nm height) were obtained from atomic force microscopy (AFM) measurements and both a truncated pyramid shape and 24% of antimony content were assumed. The result of the calculation is presented in the inset of Fig. 1. The valence band offset and conduction band discontinuity were found to be 425 meV and 120 meV, respectively.

The dynamical properties of the SOA were examined at room temperature using two-colour differential pump-probe spectroscopy.⁵ Briefly, in this technique, femtosecond pulses of different wavelengths were used to pump and probe various transitions in the structure. In order to ensure independent wavelength tunability of the pump and probe pulses, the beam from a Ti:Sa laser was split into two parts; one part was used to pump an Optical Parametric Oscillator (OPO, 75%) and the other to pump a highly nonlinear Photonic Crystal Fibre (PCF, 25%). Thus, the OPO was the source of a tunable probe beam (<300 fs, bandwidth 20 nm), whereas optical filtering of the spectrally broadened PCF output provided the pump beam (<200 fs, bandwidth 50 nm). After applying a suitable delay between the beams and propagating through the SOA, low and high frequency lock-in amplifiers measured the relative ($\Delta R/R$) change of the probe intensity due to pump-induced gain change. The wavelength of the pump pulses was set to 950 nm providing enough energy (1.31 eV) to excite both wetting layer (WL) and QD e-h pairs, while the wavelength of probe pulses was varied from 1000 nm to 1100 nm (1.24 eV–1.12 eV), which corresponds to transitions between electrons in either the dot, WL, or GaAs and hole levels in the dot (inset of Fig. 2).

In Fig. 2, we show the normalized gain dynamics at 0 mA and various wavelengths for the same pumping conditions. After arrival of the pump pulse to the structure (at 0 ps delay), we observe initial absorption bleaching followed by a single exponential wavelength-dependent decay. This decay time is attributed to radiative recombination and varies from 970 ps down to 300 ps as the current increases from 0 mA to 120 mA (when probing at 1060 nm). While our values are somewhat lower than expected for type-II structures,¹⁰ we note that in our case, the wavelengths correspond to the higher energy side of the spectrum, which results in shorter timescales. The reduction of the decay time with current can be understood in terms of an increased overlap of electron and hole wavefunctions at moderate current densities and hence faster recombination processes.¹⁴

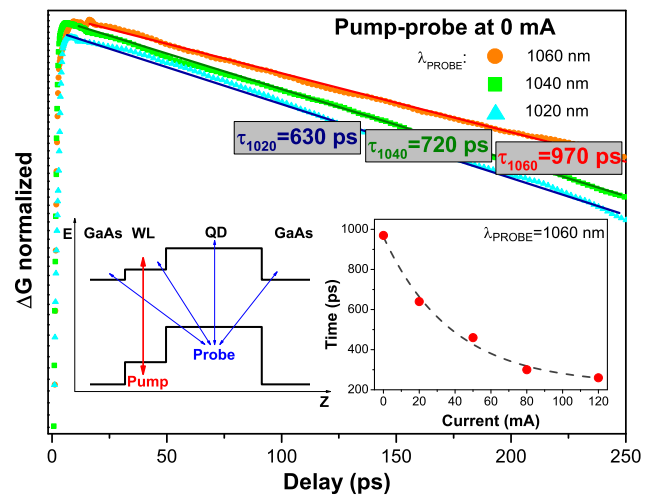


FIG. 2. Normalized change of the gain at different probe wavelengths at 0 mA bias. Inset: schematic of states pumped and probed during a single measurement and corresponding decay constant as a function of the bias current.

A more detailed examination of the initial few ps of the dynamics reveals an interesting feature, as shown in Fig. 3. Two dips are visible as the transmission increases due to the injection of carriers at higher energy. The first dip (at 0 ps) coincides with the arrival of the pump pulse and can be attributed to weak multiphoton absorption when pump and probe pulses overlap. A similar dip occurs in the type-I, two-colour case⁵ and is a much weaker version of the strong coherent effects that are observed in all single colour pump-probe measurements. The second dip occurs after ~ 1 ps and does not occur in similar measurements of type-I devices and so we attribute it to the type-II nature of the structure. As can be seen in Fig. 3, the properties of this dip do not show a clear dependence on bias current.

In contrast, Fig. 4 shows a strong dependence of the strength of the second dip on the pump power (number of injected carriers), while the first dip displays much less pump power dependence and thereby illustrates the difference in the underlying processes leading to each. Based on these observations, the appearance of a second dip can be understood in the following stages: 1. Following the first dip

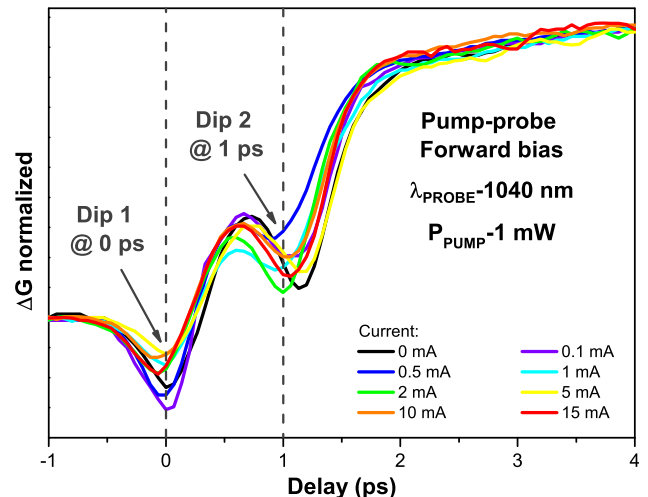


FIG. 3. Normalized first few ps of gain dynamics at various bias currents. Note the presence of dips at approximately 0 ps and 1 ps.

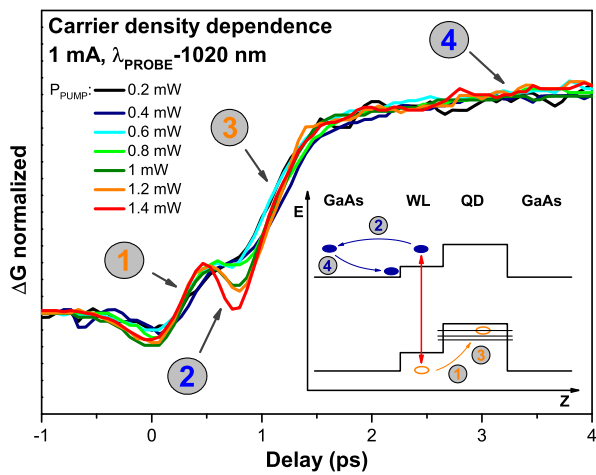


FIG. 4. Normalized first few ps of gain dynamics for increasing pump power level. Inset: schematic of various carrier transport/relaxation processes corresponding to gain changes labeled on main graph.

(which coincides with generation of electrons/holes in the WL), the probe gain increases due to relaxation of WL electrons/holes into the GaAs/QD, respectively. 2. After ~ 1 ps, the unconfined electrons start to escape and diffuse away into the GaAs and the gain reduces (second dip). 3. The gain then increases due to further relaxation of holes into the dot. 4. After ~ 1.5 ps, the final stage of the dynamics occurs with a much slower rise which we attribute to the Coulomb attraction of GaAs electrons to the vicinity of the dot by confined holes. Stages 1 and 3 are both predominantly related to hole capture dynamics and thus have similar slopes.

To further investigate this feature, the dynamics were recorded under RB in Fig. 5(a) with the recovery times presented in the inset. The reduction of the recovery time as the RB increases is similar to that measured for type-I QDs and suggests a thermalisation process for carrier relaxation.¹⁵ In the top figure, both the strength of the dip and the strength of the stage 4 (not clearly visible due to short time scale) decrease as the RB voltage increases. Thus, we conclude that as the RB increases, the injected unconfined electrons are quickly swept out and do not contribute to the subsequent dynamics. It is important to note that in reverse bias, confinement of electrons at the QD-GaAs interface due to band bending becomes possible and will lead to type-I style transitions⁸ and thereby to similar dynamics as seen at increased RB. The effect of changing the probe wavelength is shown in Fig. 5(b) and allows us to somewhat distinguish these type-I/type-II transitions. A much stronger dip can be seen at lower energies (red curve), while type-I style dynamics occur for higher energies (blue curve). This behaviour can be understood if we apply a reverse bias to the band-edge diagram in Fig. 2, as shown in inset of Fig. 5(b). In this situation, QD holes can recombine via either type-II or type-I channels with the former dominating at lower energies, while the latter dominates at higher energies.

In summary, we have presented a study of the ultrafast carrier dynamics in type-II GaAs/GaSb QDs. Off-peak measurements using a two-colour pump-probe technique demonstrate a strong dependency of carrier escape processes on the applied reverse bias voltage, the probing wavelength, and the number of injected carriers. The first few picoseconds of the

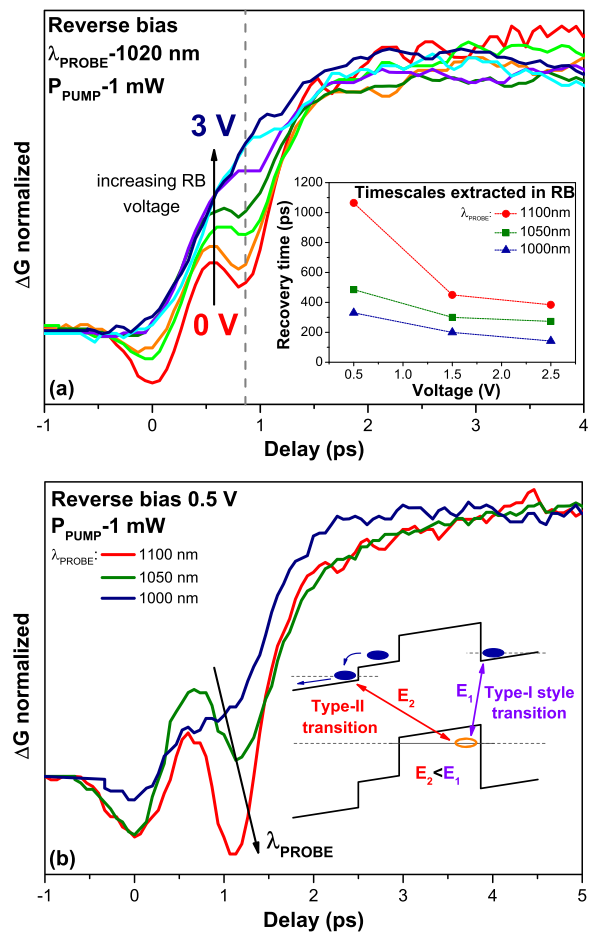


FIG. 5. (a) Normalized first few ps of gain dynamics in RB regime as RB voltage increases. Inset in (a) shows gain recovery timescales as a function of the probe wavelength/RB voltage. (b) Effect of changing probe wavelength in RB with schematic of possible shorter and longer wavelength processes. The longer wavelength process (E_2) involves unconfined electrons and is thus termed type-II, while the shorter wavelength one (E_1) involves confined electrons and is termed type-I.

dynamics reveal an interesting feature, which does not appear in devices based on type-I materials and is linked to the unique carrier processes possible in type-II structures. The onset of type-I transitions can also be seen for increasing RB voltages. In addition, the extracted decay times in forward and reverse bias may provide important information from an application point of view.

The authors would like to thank Dr. Liam Lewis for fabrication of SOAs and Dr. Bryan Kelleher for proof reading the manuscript. This work was conducted under the framework of the Irish Government's Programme for Research in Third Level Institutions Cycle 5, National Development Plan 2007-2013 with the assistance of the European Regional Development Fund. The authors acknowledge the support from the University of California Lab Fees Research Program (Grant No. 12-LR-238568).

¹M. Hayne, R. Young, E. Smakman, T. Nowozin, P. Hodgson, J. Garleff, P. Rambabu, P. Koenraad, A. Marent, L. Bonato *et al.*, *J. Phys. D: Appl. Phys.* **46**, 264001 (2013).

²J. Hwang, A. Martin, J. Millunchick, and J. Phillips, *J. Appl. Phys.* **111**, 074514 (2012).

- ³L. Müller-Kirsch, R. Heitz, A. Schliwa, O. Stier, D. Bimberg, H. Kirmse, and W. Neumann, *Appl. Phys. Lett.* **78**, 1418 (2001).
- ⁴K. Gradkowski, T. Ochalski, N. Pavarelli, H. Liu, J. Tatebayashi, D. Williams, D. Mowbray, G. Huyet, and D. Huffaker, *Phys. Rev. B* **85**, 035432 (2012).
- ⁵I. O'Driscoll, T. Piwonski, C. Schleussner, J. Houlihan, G. Huyet, and R. Manning, *Appl. Phys. Lett.* **91**, 071111 (2007).
- ⁶P. Kambhampati, *Acc. Chem. Res.* **44**, 1 (2010).
- ⁷F. Hatami, N. Ledentsov, M. Grundmann, J. Böhrer, F. Heinrichsdorff, M. Beer, D. Bimberg, S. Ruvimov, P. Werner, U. Gösele *et al.*, *Appl. Phys. Lett.* **67**, 656 (1995).
- ⁸D. Alonso-Álvarez, B. Alén, J. M. García, and J. M. Ripalda, *Appl. Phys. Lett.* **91**, 263103 (2007).
- ⁹P. Hodgson, R. Young, M. A. Kamarudin, P. Carrington, A. Krier, Q. Zhuang, E. Smakman, P. Koenraad, and M. Hayne, *J. Appl. Phys.* **114**, 073519 (2013).
- ¹⁰F. Hatami, M. Grundmann, N. Ledentsov, F. Heinrichsdorff, R. Heitz, J. Böhrer, D. Bimberg, S. Ruvimov, P. Werner, V. Ustinov *et al.*, *Phys. Rev. B* **57**, 4635 (1998).
- ¹¹J. Tatebayashi, A. Khoshakhlagh, S. Huang, G. Balakrishnan, L. Dawson, D. Huffaker, D. Bussian, H. Htoon, and V. Klimov, *Appl. Phys. Lett.* **90**, 261115 (2007).
- ¹²C.-C. Tseng, W.-H. Lin, S.-Y. Wu, S.-H. Chen, and S.-Y. Lin, *J. Cryst. Growth* **323**, 466 (2011).
- ¹³S. Morozov, D. Kryzhkov, A. Yablonsky, A. Antonov, D. Kuritsin, D. Gaponova, Y. G. Sadofyev, N. Samal, V. Gavrilenko, and Z. Krasilnik, *J. Appl. Phys.* **113**, 163107 (2013).
- ¹⁴K. Gradkowski, N. Pavarelli, T. Ochalski, D. Williams, J. Tatebayashi, G. Huyet, E. O'Reilly, and D. Huffaker, *Appl. Phys. Lett.* **95**, 061102 (2009).
- ¹⁵T. Piwonski, J. Pulka, G. Madden, G. Huyet, J. Houlihan, E. A. Viktorov, T. Erneux, and P. Mandel, *Appl. Phys. Lett.* **94**, 123504 (2009).